Influence of Calcination Temperature on the Structure and Porosity of Nanocrystalline SnO₂ Synthesized by a Conventional Precipitation method

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Tin dioxide nanocrystalline were synthesized by adding ammonia solution (NH₄ OH) to a solution containing tin chloride pentahydrate (SnCl₄.5H₂O). The obtained products have been characterized by x-ray diffraction (XRD), scanning electron microscopy (SEM) and high resolution transmission electron microscopy (HRTEM). The results indicate that the crystallite size increases from 3.45 to 23.5 nm as the calcination temperature rises from 300°C to 1050°C. The XRD analysis exhibits a rutile tetragonal phase of the SnO₂ nanocrystals. The crystallinity is developed due to a decrease of lattice distortion ratio and dislocation density as the calcination temperature is increases. Surface area and porosity of SnO₂ nanoparticles are measured. Specific surface area which is related to pore volume and decreases from 140 m²/g at 100°C to 8.3 m²/g at 1050°C.

Keywords: nanocrystalline SnO₂, XRD, Specific surface area, HRTEM

1. INTRODUCTION

Tin dioxide SnO_2 is a very stable material which has tetragonal rutile structure [1, 2]. SnO_2 in nanometer scale have many advantages compared to its bulk materials. Tin oxide has many technological applications among them as catalysts for oxidation of organic compounds, solid state sensors, biomedicine, ceramics and transparent conductors [3, 4]. In the field of gas sensors, SnO_2 nanoparticles are the most important material used in gas sensing applications. SnO_2 performance, such as stability, sensitivity, selectivity and response time, depend on the reduction of particles size and high surface area of the prepared material [5-7]. On the other hand, the associated challenge with sol-gel and chemical precipitation methods is; when the product is calcined at high temperatures the crystallite size increases and the surface area decreases due to the particle growth [8, 9]. SnO₂

nanoparticles is prepared by several methods as sonochemical method [10], laser decomposition reaction [11], sol- gel [12], two-step solid state reaction [13], spray pyrolysis [14], precipitation [15, 16], etc.

Precipitation method is the most common methods used for synthesis of tin dioxide nanparticles. The main factors, which affect the preparation of SnO_2 nanoparticles by precipitation method in terms of smaller and more dispersed particles, are temperature and pH [2].

The aim of this article is to characterize the nanocrystalline tin dioxide synthesized by the conventional precipitation method. The effect of calcination temperature on the crystallite size, lattice distortion ratio, and dislocation density has been characterized. Specific surface area, pore size distribution, and activation energy of sintering of SnO_2 nanoparticles calcined at various temperatures have been evaluated by N_2 adsorption using Brunauer, Emmet and Teller (BET); Barrer, Joiyner and Halenda (BJH) and t-plot methods.

2. EXPERIMENTAL PROCEDURE

2.1. SnO₂ sample preparation

Nanocrystalline tin oxide (SnO₂) was prepared by conventional precipitation method, by adding ammonia solution NH₃.H₂O 30 % drop wise to 0.05M aqueous solution of SnCl₄.5H₂O at 40°C. The pH of the mixed solution was adjusted at 8 to obtained white precipitate of tin hydroxide. The product was separated using a centrifuge operating at 3000 rpm for 20 min. The white gel was washed with deionized water and ethanol several times to remove chlorine ions. The resulting product was dried for 24 h at 100°C in drying oven. The dried powder has been ground in a mortar and finally calcined at different temperatures in a muffle furnace for 2 h in the temperature range 300⁻ 1050 °C.

2.2. Samples Characterization

The crystal structure of all samples characterized at room temperature using a Philips PW1700 X-ray diffractometer (operated at 40 kV and current of 30 mA) and samples were scanned between 20° and 90° at a scanning speed of 0.06 °C/s using Cu K α radiation with $\lambda = 1.5418$ Å. (SEM) (JEOL JSM-5400 LV) used to examine the surface morphology of particles. (HRTEM) and selected area electron diffraction (SAED) performed using a JEOL JEM-2100 (operating voltage = 200KV and beam current = 102µA) were used to examine the size and shape of particles. Porosity and specific surface area of SnO₂ nanoparticles calcined at various temperatures measured by N₂ adsorption using (BET), (BJH) and t-plot methods. Using a computer interfaced Nova 3200 series instrument (Quantachrgme, USA), all samples are thoroughly outgassed for 3hr at 250°C.

3. RESULTS AND DISCUSSIONS

3.1. Dependence of crystallite size, crystal lattice distortion and dislocation density on the calcination temperature:

The analysis of the structure and size determination of SnO_2 nanocrystals were carried out by using modified Scherrer equation in the form [17]

$$D = \frac{k\lambda}{\beta cos\theta} \tag{1}$$

where β is defined as $\beta = (\beta^2_{observed} - \beta^2_{instrumental})^{1/2}$ (in radians), k is the shape factor equals to 0.89, λ is the X-ray wave length for Cu k α radiation ($\lambda = 1.5418$ Å), θ is the Bragg diffraction angle (in degrees) and β is the full width at half maximum (FWHM) of the observed peak which is corrected by considering the instrumental broadening of the standard Si.



Figure 1. Shows the XRD charts of the calcined samples at different temperatures,

The representative XRD charts, Fig1, of dried sample (100°C) and samples calcined in the range of 300°C to 1050°C. Miller indices provided in the figure and all peaks determine transformation of dried powder to SnO_2 crystallites with tetragonal rutile crystal structure. In addition, the number of reflection increases with increasing of the calcination temperature but the degree of crystallinity does not change after 600°C indicating fully crystalline SnO_2 nanopowders. Table 1, shows the XRD parameters of SnO_2 nanopowder at various crystalline orientations.

Table 1. The crystallite size, lattice parameter, dislocation density and number of cells in SnO_2 nanoparticles.

Temp. (°C)	100°C	300°C	450°C	600°C	750°C	900°C	1050°C
Crystallite size (nm)	2.9	3.45	5.4	13.3	14.4	18.4	23.54
Miller indices (hkl)	$110 \\ 101 \\ 211$	110 101 211	110 101 211	$110 \\ 101 \\ 211$	110 101 211	110 101 211	$110 \\ 101 \\ 211$
d-spacing	3.391 2.667 1.779	3.383 2.667 1.756	3.348 2.656 1.763	3.354 2.645 1.764	3.353 2.645 1.764	3.356 2.647 1.765	3.353 2.646 1.764
a (Å)	4.795	4.709	4.734	4.743	4.74	4.74	4.746
c (Å)	3.22	3.24	3.21	3.19	3.187	3.187	3.188
Unit volume $(Å^3)$	74.04	71.84	71.95	71.86	71.62	71.62	71.84
Lattice distortion	0.013	0.0114	0.0079	0.0032	0.0029	0.0025	0.0018
Dislocation density(lines/m ²) $\times 10^{15}$	122	89.1	35.6	5.65	4.82	2.95	1.8
Number of unit cell $(\times 10^3)$	0.165	0.3	1.08	17	21.8	45.5	90



Figure 2. Dependence of the FWHM of (110), (101), (211) and (200) planes on the calcination temperature



Figure 3. Dependence of the crystallites growth on the calcination temperature.

Fig. 2, shows the relation between the broadening of highest intensity planes (110), (101), (211) and (200) and calcination temperatures used to estimate the average crystallite size.

On the other hand, it's noticed that the width of peaks decreases with increasing of the calcination temperatures due to growth of crystals and construction to larger clusters. The crystallite size increases with increasing the calcination temperature. Fig. 3, describes the slow increase of crystallite size in the early stage of calcination from 100°C - 450°C related to presence of water molecules and hydroxyl groups which is confirmed by TGA and DTA results (presented elsewhere) prevent rapid growth of crystals. Above 450°C the crystal growth increases rapidly with increasing in the calcination temperature which can be ascribed by more agglomerated crystals.

The crystal lattice distortion ratio of SnO_2 nanoparticles at different calcination temperatures calculated by using the following equation [18, 19]

$$\beta^2 \cos^2 \theta = \frac{4\lambda^2}{\pi^2 D^2} + 32\langle \varepsilon^2 \rangle \sin^2 \theta, \tag{2}$$

The parameters θ , β and λ are defined earlier, $(\epsilon^2)^{1/2}$ is the crystal lattice distortion ratio. According to modified Scherrer equation (1) and equation (2), it can be deduced the lattice distortion ratio using equation (3) in the form [19].

$$\sqrt{\langle \varepsilon^2 \rangle} = \frac{1}{D} \frac{1}{\sin\theta} \frac{\pi}{\lambda} \sqrt{\frac{\pi^2 k^2 - 4}{32}}$$
(3)

Fig.4. describes the relation between lattice distortion ratios determined using (110) plane from XRD line broadening and calcination temperatures.



Figure 4. Relations between crystal lattice distortion ratio and calcination temperature of SnO₂ nanoparticles.

In this figure it can be noticed a decrease in lattice distortion ratio with increasing the calcination temperature. This result can be explained as follows; when the temperature increases, it leads to a decrease in defect concentration by decreasing in the proportion of surface atoms. It also improves the degree of crystallization in SnO_2 nanoparticles [18-20]. The dislocation density (δ) and number of unit cells (n) of SnO_2 nanoparticles is calculated and listed in table (1) using the relation [21-23].

$\delta = 1 / D^2$	(4)
$n = \pi D^3 / 6 V$	(5)

where D is the crystallite size and V is the unit volume. Dislocation density decreases and the number of unit cells increases with increasing in the calcination temperatures that leads to increase the crystal growth and decreasing the defects in crystallites.

3.2. Scanning electron microscope (SEM)

The SEM micrographs of the as- dried powder and samples calcined at 750°C, 900°C and 1050°C, are shown in Fig (5). All images indicate that the particles have non-uniform size with high degree of agglomeration. The as- dried powder Fig (5-a) has a non- uniformed size of particles with

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wide size distribution. After calcination from 750°C - 1050°C, Fig (5-b, c and d), the morphology of the specimen surface changes obviously with increasing the average particles size and more agglomeration occurs among particles. On the other hand, all particles exhibit a spherical shape with a high degree of agglomeration among fine particles. Therefore, one cannot measure the particles size.



Figure 5. SEM micrograph of the surface of SnO₂ nanocrystals at different temperature a) as prepared powder, b) T=750 °C, c) T=900 °C and d)T=1050 °C respectively.

3.3. High resolution transmission electron microscope (HRTEM)

The micro- structure of the SnO₂ nanoparticles calcined at 450 °C, 600 °C, and 750 °C, are investigated by HRTEM. The micrographs in Fig (6) show the nanoparticles appear as spheres with uniform size as the calcination temperature increased. The average particle size increases as a function of temperature. Furthermore, the HRTEM results confirm the XRD results that show the nanoparticle size is dependent on the calcination temperature. It is clearly shown that the SnO₂ particles has a high degree of agglomeration caused by mutual interaction between particles which arises from some forces such as van der – walls forces, capillary forces and electrostatic forces [24, 25]. Furthermore, adding ammonia solution directly leads to complete the nucleation and growth in a few seconds that affects on agglomeration of particles. The agglomerated particles have regular distribution and the shapes of particles are, to some extent, similar and homogenous.



Figure 6. HRTEM images and SAED of SnO₂ nanocrystals calcined at different temperatures a, b) T = 450 °C, c, d) T = 600 °C, e, f) T = 750 °C.

The selected area electron diffraction (SAED) patterns of tin oxide nanocrystals calcined at different temperatures characterized by diffraction rings with discrete spots which indicate that tin oxide is fully crystalline with polycrystalline nature which is in good agreement with the XRD patterns of tin dioxide nanocrystals

3.4. Porosity and surface area of tin dioxide (SnO₂) nanoparticles

3.4.1. N_2 adsorption / desorption for tin dioxide (SnO₂) nanoparticles:



Figure 7. Nitrogen adsorption/ desorption isotherms of SnO₂ nanoparticles

 N_2 adsorption / desorption for tin dioxide nanoparticles synthesized by conventional precipitation method has been carried out. Figure 7, shows the isotherm curves for dried sample and samples calcined in the temperature range 300°C- 1050°C. In the dried sample A1 and sample A2 calcined at 300°C, the curves illustrate the samples have similar type I adsorption isotherm behaviors with H4 type of hysteresis loops according to the IUPAC classifications [26, 27]. This type of adsorption isotherm is related to microporous solids, and the hysteresis loop is often associated with narrow slits like pores as a result of agglomerates or aggregates of particles forming slit pores with uniform size and shape [26, 28].

The samples A4, A5, A6 and A7 calcined at temperatures 600°C, 750°C, 900°C and 1050°C, respectively. The isotherm curves of these samples are corresponding to atypical type II isotherm. This

isotherm determined that the samples are free of micropores with a high energy of adsorption. In addition, the hysteresis loops for these samples are H4 which means that the material is often associated with narrow slit- like pores as in the previous two samples A1, A2. In contrast, the sample A3 calcined at 450°C shows the similar IV isotherm classification with H4 of hysteresis loop [26, 28]. This type is characteristic of mesoporosity material with uniform size and shape of pores. The results indicate decreases of the adsorbed volume with increasing the calcination temperature, which leads to decreasing in specific surface area.





Figure 8. Changes in the specific surface area and total pore volume for the samples calcined at various temperatures.

The relation between specific surface area of the samples calcined at different temperatures and total pore volume illustrated in Fig 8. It's notable that the total pore volume (t.p.v) and specific surface area (S_{BET}) for samples decrease with increasing in the calcination temperature due to increasing the particle size and degree of agglomeration by the effect of sintering occurring in the material [3]. Furthermore, this result shows a sharp decrease in specific surface area for samples in the temperature range 450°C to 750°C due to elimination of ammonia and chemically bonded water during development tin hydroxide to pure tin oxide [3, 16, 29-30]. A bove 750°C, S_{BET} decreases gradually. The specific surface area of the samples decreases from 112.4 m²/g at 300°C to 8.3 m²/g at 1050°C. The high surface area of SnO₂ at low calcination temperatures lower than 450°C suggests small size of particles.

Assuming that the particles have spherical – shape and uniform size, the average particle size can be estimated by (BET) equation in the form [12].

$$\mathbf{D} = \frac{6000}{\rho \, \mathrm{A}} \tag{6}$$

where D is the average particle size in nm, ρ is the theoretical density of tin dioxide nanoparticles (6.95 gm/cm³) and A is the specific surface area at corresponding temperature in m²/g.



Figure 9. Variations of crystallite size calculated by XRD and particle size calculated from BET as a function of calcination temperature.



Figure 10. Variation of log S_{BET} versus 1/T for SnO₂ nanoparticles calcined at different temperatures.

In order to compare between the average crystallite size estimated by XRD and the average particle size estimated by BET, given in Fig 9, the results are in fair agreement in the low temperatures (lower than 450°C) due to high surface area and smaller particle size. But in contrary, the average particle size calculated by BET is larger than the crystallite size calculated by XRD when the calcination temperature rises from 450°C to 1050°C. The results are inconsistent because aggregates and/ or agglomerates of crystals due to increasing the calcination temperature. These results indicate that the particles contain several crystallites [31].

Fig. 10, shows the determination of the activation energy of sintering (ΔE_s) for samples calcined from 600°C to 1050°C. ΔE_s can be deduced from the data of S_{BET} as a function of calcination temperature. The relation between ΔE_s and S_{BET} given by the equation [32]

 $S_{BET} = A \exp \left(\Delta E_s / RT\right)$ (7)

where A is a constant and ΔE_s is the approximate activation energy of sintering determined by plotting log S_{BET} versus 1/T. The activation energy of sintering calculated from the slope of the straight line equals to 19.9 kJ/mol. The dried sample and samples calcined at temperatures lower than 600°C are containing water molecules and hydroxyl groups that lead to prevent the increase in the rate growth and sintering of the particles.

3.4.3. Nature of porosity in SnO₂ nanoparticles

Temp.(°C)	$S_{BET}(m^2/g)$	APD (nm)	TPV(nm)	D _{BET} (nm)
100	140.17	2.243	0.0786	6.2
300	112.38	2.617	0.0735	7.7
450	66.73	4.215	0.0703	13
600	21.19	6.157	0.0326	41
750	15.43	5.012	0.0193	56.4
900	11.06	5.129	0.0142	78.6
1050	8.32	4.967	0.0103	103.7

Table 2. Specific surface area, pore size distribution and particle size of SnO₂ nanoparticles calcined at the indicated temperatures.

 S_{BET} (m²/g): specific surface area. APD (nm): average pore diameter, TPV (nm): total pore volume, D_{BET} (nm): particle size calculation by BET equation

Fig. 11, presents the pore size distribution of the samples obtained by BJH method at different temperatures. It is noticed that the mean peaks of all samples are in the range between 2 nm to 4.5 nm. These peaks indicate that the pore size distribution is very narrow with micropores and mesopores in samples. The total number of pores decreases as a result of sintering while the amount of larger pore size increases. The values of total pore volume and average pore diameter calculated and cited in table (2). The increase in the average pore diameter is resulted from agglomeration among particles that leads to sintering of pores to small ones with decreasing the homogeneity of the dimensions and number of pores [3, 33].



Figure 11. Pore size distribution of SnO₂ nanoparticles calcined at different temperatures.



Figure 12. t-plots of SnO₂ nanoparticles calcined at different temperatures.

On the other hand, by using the shape of the V- t plot it can be determined the type of existing porosity. Downward deviation determines microporosity while the upward deviation indicates the existence of mesoporosity [33].

From Fig 12, all samples exhibit downward deviations which indicate a micropores nature of pores. These results indicate that the calcination temperature for the samples and the preparation method of tin dioxide nanparticles have no major effect on the nature of pores. The mentioned above indicate that the BJH method and t- plot method determine the nature of pores in our materials as micropores with little mesopores.

4. CONCLUSIONS

From the study of SnO₂ nanoparticles the following conclusions can be drawn:

- 1. The crystallite size is calcination temperature dependent. The crystallite size of the samples increased from 3.45 to 23.5 nm when the calcination temperature increases from 300 to 1050°C respectively.
- 2. The crystallinity of SnO_2 nanoparticles might be improved by decreasing the lattice distortion and dislocation density with increasing calcination temperature.
- 3. The SEM examinations reveal that when the calcination temperature increases the agglomeration of non uniform particles increases.
- 4. HRTEM indicates a dependence of particle size and the shape on the calcination temperatures. The particle size increases and has a higher degree of agglomeration as the calcination temperature increase.
- 5. Specific surface area of SnO_2 nanoparticles lays in the range 140.17 m²/g to 8.32 m²/g as the calcination temperature rises from 100°C to 1050°C.
- 6. Specific surface area of SnO_2 nanoparticles is related to pore volume and decreased by sintering of pores to smaller pores.
- 7. Pore size distribution measured by BJH and t-plot methods indicates that the nature of pores in SnO_2 nanoparticles is a collection of micropores with a very little mesopores.

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