# Synthesis, Characterization and Electrochemical Performance of Manganese Dioxide in a Quaternary Microemulsion: the Role of the Co-surfactant and Water

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Manganese dioxide nanoparticles were synthesized in a quaternary microemulsion using surfactant hexadecyl-trimethylammonium bromide. The structure and morphology of the as-prepared manganese dioxide powders were characterized by means of powder X-ray diffraction, scanning electron microscopy and transmission electron microscopy. The average size of nanoparticles, as well as the crystalline structure of manganese dioxide, can be controlled by the content of co-surfactant n-pentanol and water. The particle size decreased with adding the amount of n-pentanol or reducing water, and simultaneously manganese dioxide nanosheets were generated. Their electrochemical properties were investigated using cyclic voltammetry in 0.1 M Na<sub>2</sub>SO<sub>4</sub> electrolyte. The specific capacitance of the as-prepared manganese dioxide powder with the surface area of 231.5 m<sup>2</sup> g<sup>-1</sup> was 140.5 F g<sup>-1</sup> at scanning rate of 2 mV s<sup>-1</sup>.

Keywords: Manganese dioxide, supercapacitor, microemulsion, co-surfactant.

# **1. INTRODUCTION**

In recent years, supercapacitors have attracted more and more attentions because they can offer higher powder density and have the excellent cyclicity [1]. Based on the charge-storage mechanism, supercapacitors are divided into two types: electrical double-layer capacitors (EDLCs) and Faradic pseudocapacitors. EDLCs are based on the double-layer effect of carbon materials with high surface area and Faradic pseudocapacitors are caused by a faradic charge exchange reaction. Some noble or transition metal oxides show pseudo-capacitive behavior. For example, amorphous hydrous ruthenium dioxide (RuO<sub>2</sub>) has a high specific capacitance (760 F g<sup>-1</sup>) in H<sub>2</sub>SO<sub>4</sub> solution [2]. Compared to RuO<sub>2</sub>

with high cost, Manganese dioxide  $(MnO_2)$  is inexpensive, environmentally friendly, and abundant in raw materials [3]. Since Lee and Good-enough found its ideal capacitive behavior,  $MnO_2$  has attracted more interest due to its possible application in future.

Various methods have been reported to synthesize  $MnO_2$ , including thermal decomposition [4], template method [5], hydrothermal reaction [6], and microemulsion methods [7-9], etc. Microemulsion-based synthesis has been proved to be a useful method to synthesize a variety of nanostructured nanoparticles. This method enables researchers to synthesize nanomaterials with varied size and shape (such as nanosphere, nanotube, nanowire, etc) by altering various components involved in the formation of microemulsion [10-12]. "Water-in-oil" microemulsion is composed of aqueous nano droplets and continuous oil medium, which are stabilized by surfactant and co-surfactant located at the water/oil interface. Devarj and Munichandraiah [8] synthesized  $MnO_2$  particles in the spherical/hexagonal shape with about 50 nm size by microemulsion method. Xu *et al.* [9] reported a self-reacting microemulsion method by which  $MnO_2$  nanosphere with the diameter of 4 nm was synthesized. However, there are few reports concerning the changes of the size and crystalline structure of  $MnO_2$  by altering the content of co-surfactant and water in the microemulsion system. The nano water droplets can be seen as nanoreactors in a microemulsion system, and the size or the stability of nanoparticles could be adjusted during the reaction [13-17] by altering the content of the co-surfactant and the water.

In this paper, we chose n-hexane (NH) as the oil phase, hexadecyl-trimethylammonium bromide (CTAB) as the surfactant, n-pentanol (NP) as the co-surfactant. As far as the quaternary microemulsion is concerned, there are three factors that determine the formation of microemulsion: the molar ratio between water and surfactant  $(W_0)$ , that molar ratio between co-surfactant and surfactant  $(P_0)$ , and the molar concentration of the surfactant (S). The aqueous phase of the reverse micelles plays an important role in determining the size of the final products [18]. Nano water droplets surrounding the reactants, are the places where the reaction occurs, and the particle size is determinate by the size of droplets which is a function of  $W_0$ . Pileni's group [11] pioneered the studies about the controlling of nanoparticles' size and shape by varying  $W_0$ , and showed that the size of copper nanoparticles increases with increasing water content. The role of a co-surfactant is to lower the interfacial tension between the oil and water for the spontaneous formation of surfactant aggregates [14]. Curri et al reported that Cadmium sulfide (CdS) nanoparticles were synthesized in a quaternary CTAB/NP/NH/water microemulsion with NP as the co-surfactant, and the average size of the CdS nanoparticles was also controlled by the NP content [19]. Therefore, here we studied intensively the role of the co-surfactant NP and water as potentially useful parameters in controlling the size and crystalline structure of MnO<sub>2</sub>.

## 2. EXPERIMENTAL DETAILS

## 2.1 Preparation of $MnO_2$ with different $P_0$

The  $MnO_2$  powders were synthesized by a microemulsion route, as shown in Fig.1. The quaternary microemulsion system was composed of CTAB, water, NP, and NH. The oil phase was

NH, and CTAB was chosen as the surfactant and NP as the co-surfactant. The concentration of CTAB in oil phase was 0.1M.  $W_0$  was 10, and  $P_0$  was 5, 10, 15. Only when the value of  $W_0$  is larger than or equal to 5, CTAB could dissolve in water and the water-in-oil system could be formed [20]. Therefore, 4.5 ml water was first added to form a water-in-oil solution. Denoted the two water-in-oil systems as A and B, added 4.5 ml water and 4.5 ml of 0.2 M KMnO<sub>4</sub> solution into system A, and 4.5 ml water and 4.5ml of 0.3 M Mn(C<sub>2</sub>H<sub>3</sub>O<sub>2</sub>)<sub>2</sub> solution into B, to form nano water droplets containing reactants. Then two systems were mixed and stirred for 1 h, and the MnO<sub>2</sub> powder suspension was formed. After filtered with deionized water and dried for 24 h at 85 °C, the samples were obtained. The as-prepared powder was denoted as MnO<sub>2</sub>- $P_0$ -X where X represents the value of  $P_0$ .



Figure 1. Process procedure for preparing MnO<sub>2</sub> using a microemulsion route.

## 2.2 Preparation of MnO<sub>2</sub> with different W<sub>0</sub>

The synthesis route was the same as the above. The concentration of CTAB in oil phase was 0.1 M. Among the above all of as-prepared  $MnO_2-P_0-X$  powders, the electrochemical performance of sample  $MnO_2-P_0-10$  is the best and thus  $P_0$  value was fixed at 10, and  $W_0$  was set as 10, 20, 30. In the same way, the  $MnO_2$  powders with different  $W_0$  were obtained, and the as-prepared powder was denoted as  $MnO_2-W_0-Y$  where Y represents the value of  $W_0$ . Because  $W_0$  was set as 10 when preparing sample  $MnO_2-P_0-10$ ,  $MnO_2-P_0-10$  and  $MnO_2-W_0-10$  were the same, then added samples  $MnO_2-W_0-20$ , 30.

## 2.3 Characterization

Powder X-ray diffraction (XRD) patterns of the samples were obtained by using a TW3040/60 diffractometer (Tanalygical Company, Holland) in which Cu-Ka ( $\lambda$ =0.154 nm) was used as the radiation source. The morphologies were examined by Hitachi S-4800 scanning electron microscopy

(SEM) and JEM-2100F transmission electron microscopy (TEM). The surface area and the pore size distribution were determined by  $N_2$  adsorption at 77 K with an automated adsorption apparatus (Micromeritics ASAP 2020). Based on the Brunauer-Emmett-Tellr (BET) equation, the surface area was calculated. The pore size distribution was obtained by the BJH method [21] from the desorption isotherm.

#### 2.4 Electrochemical measurement

The electrode was prepared by mixing 70 wt.% of  $MnO_2$  powder with 20 wt.% acetylene black and 10 wt.% polytetrafluoroethylene (PTFE). The detailed procedure was as follows: 35 mg of  $MnO_2$ powder and 10 mg of acetylene black were first mixed and dispersed in ethanol by sonication for 30 min. Then the ink was dried at 80 °C for 12 h and 100 mg of PTFE aqueous solution (5 wt.%) was added. After being dried at 80 °C, a few drops of 1-methy-2-pyrrolidinone (NMP) were added to get the syrup. The syrup was rolled into thick film. The film was cut into the 1cm × 1cm pieces of about 3 mg, and then hot-pressed at 80 °C under 40 Mpa onto the titanium foil as the collector.

Electrochemical tests were performed with the battery system VMP3 (Bio-Logic Corp., France). A piece of platinum gauze and a saturated calomel electrode (SCE) were assembled as the counter and reference electrode. 0.1 M Na<sub>2</sub>SO<sub>4</sub> solution was used as the electrolyte.

## **3. RESULTS AND DISCUSSION**

## 3.1 Characteristics of samples $MnO_2$ with different $P_0$ or $W_0$

Fig.2 shows the XRD patterns of samples MnO<sub>2</sub>- $P_0$ -5, 10, 15 when other parameters  $W_0$  and S of microemulsion system are the constant. The presence of a few broad peaks is related to the poor crystallinity and nanoscale nanoparticles. The profiles of a few broad peaks around  $2\theta$ =37.1 ° and 65.6 ° seem to be consistent with the peaks of  $\gamma$ -MnO<sub>2</sub> (JCPDS NO.14-0644). The peak around  $2\theta$ =21 ° shows the existence of  $\gamma$ -MnO<sub>2</sub> when  $P_0$ =5, and then disappears when  $P_0$ =10. By adding more NP, the appearance of a broad peak around  $2\theta$ =12.2 ° shows the existence of Birnessite-type MnO<sub>2</sub> (JCPDS NO.18-0802).

Among the above all of as-prepared MnO<sub>2</sub>- $P_0$ -X powders, the electrochemical performance of sample MnO<sub>2</sub>- $P_0$ -10 is the best and  $P_0$  value was fixed at 10. Actually, MnO<sub>2</sub>- $P_0$ -10 and MnO<sub>2</sub>- $W_0$ -10 are the same. Fig.3 shows the XRD patterns of samples MnO<sub>2</sub>- $W_0$ -10, 20, 30. As discussed above, When  $W_0$ =10, 20, 30, a few broad peaks around 2 $\theta$  =37.1° and 65.6° are the same, the profile of these peaks seems to be consistent with the peaks of  $\gamma$ -MnO<sub>2</sub>. By adding more water, the appearance of a broad peak around 2 $\theta$ =12.2° shows the existence of Birnessite-type MnO<sub>2</sub>.



Figure 2. XRD patterns of samples  $MnO_2-P_0-5$ , 10, 15 prepared with different  $P_0$  when other parameters  $W_0$  and S of microemulsion systems are the constant.



**Figure 3.** XRD patterns of samples  $MnO_2$ - $W_0$ -10, 20, 30 with different  $W_0$  when other parameters  $P_0$  and S of microemulsion systems are the constant,  $P_0$ =10.

As discussed above, it can be concluded that the crystalline structure of  $MnO_2$  can be controlled by changing the content of co-surfactant NP and water. With increasing the content of NP or

water, Birnessite-type MnO<sub>2</sub> is generated gradually among the as-prepared  $\gamma$ -MnO<sub>2</sub>, which can be proved by the appearance of a broad peak around 20=12.2 °.

The morphology of samples  $MnO_2-P_0-5$ , 10, 15 and  $MnO_2-W_0-10$ , 20, 30 is shown as the SEM images in Fig.4, and  $MnO_2-P_0-15$  and  $MnO_2-W_0-20$ , 30 TEM images are shown in Fig.5. It shows that  $MnO_2-P_0-5$ , 10 spherical nanoparticles agglomerate together, and  $MnO_2-P_0-15$  has mainly spherical nanoparticles besides  $MnO_2$  nanosheets observed by TEM image in Fig.5 (a). It can be seen that the size of spherical  $MnO_2$  nanoparticles decreases significantly by adding more NP, because the interfacial tension between NH and water decreases gradually, the higher fluidity of the interfacial film and higher mean curvature of radius in turn make the intermicellar exchange rate higher, which implies the higher consumption of reactants at the nucleation stage, thus reduces the effective concentration for further growth, and results in smaller  $MnO_2$  nanoparticles.



Figure 4. The morphology of samples  $MnO_2-P_0-5$ , 10, 15 (a-c),  $MnO_2-W_0-20$ , 30 (d-e),  $MnO_2-P_0-10$  and  $MnO_2-W_0-10$  are the same (b).



Figure 5. TEM images of samples  $MnO_2-P_0-15$  and  $MnO_2-W_0-20$ , 30,  $MnO_2$  nanosheets were observed in these three samples.

 $MnO_2-P_0-10$  or  $MnO_2-W_0-10$  is shown in Fig.4 (b), and  $MnO_2-W_0-20$ , 30 have mainly spherical nanoparticles besides  $MnO_2$  nanosheets observed by TEM images in Fig.5 (b-c). It can be seen that the size of the spherical  $MnO_2$  nanoparticles increases by adding more water in the system. Due to the increasing of the water content, the number of nano water droplets in the unit volume increases, greatly promoting the possibilities of collision between droplets and the exchange rate of the reactants, which implies more rapid nucleation and growth, and finally results in the larger  $MnO_2$  nanoparticles.

Some researchers synthesized  $MnO_2$  by microemulsion-based methods using different surfactants [8, 9, 22, 23], the results show that the size, shape and crystalline structure of  $MnO_2$  could be changed by using different surfactants. However, the effect of co-surfactant or water on  $MnO_2$  in a quaternary microemulsion has not been studied. In this work, we investigated this issue initially, and pointed out that the average size and the crystalline structure of  $MnO_2$  could be controlled by altering the content of the co-surfactant NP and water as discussed above. It can be concluded that the particle size decreases when adding the amount of NP or reducing water, and  $MnO_2$  nanosheets are generated simultaneously.

The nitrogen adsorption/desorption isotherms and the pore size distributions of samples  $MnO_2$ - $P_0$ -5, 10, 15 are shown in Fig.6 and Fig.7, respectively. It shows that samples  $MnO_2$ - $P_0$ -5, 10, 15 are typically mesoporous materials with different pore size distributions piled up by the nano powders,

respectively. The specific surface area of  $MnO_2-P_0-5$  is 84.9 m<sup>2</sup> g<sup>-1</sup>, and  $MnO_2-P_0-10$  has a higher specific surface area of 231.5 m<sup>2</sup> g<sup>-1</sup>, because the average size of  $MnO_2$  nanoparticles is smaller and the spherical nanoparticles agglomerate together more easily, resulting that mesopores could be piled up by the nanoparticles (Fig.7). The specific surface area of  $MnO_2-P_0-15$  decreases to 21.6 m<sup>2</sup> g<sup>-1</sup> probably because MnO<sub>2</sub> nanosheets are generated, which may lead to the smaller surface area.

The nitrogen adsorption/desorption isotherms and the pore size distributions of samples  $MnO_2$ - $W_0$ -10, 20, 30 are shown in Fig.8 and Fig.9, respectively. As is mentioned above, when  $P_0$ =10 and  $W_0$ =10, the as-prepared  $MnO_2$  is typically a mesoporous material with the specific surface area of 231.5 m<sup>2</sup> g<sup>-1</sup>.  $MnO_2$ - $W_0$ -20, 30 are mesoporous materials with the specific surface area of 15.3, 14.5 m<sup>2</sup> g<sup>-1</sup> probably because  $MnO_2$  nanosheets generated have smaller surface area.

The nanoparticles size and the crystalline structure of  $MnO_2$  which can be controlled by the content of co-surfactant NP and water, both have an influence on the pore size distributions of  $MnO_2$  samples, which is not studied elsewhere. When  $P_0=10$  and  $W_0=10$ , the as-prepared  $MnO_2$  has the highest specific surface area.



Figure 6. Nitrogen adsorption/desorption isotherms at 77K for samples  $MnO_2$ - $P_0$ -5, 10, 15.



**Figure 7.** The pore size distribution of samples  $MnO_2$ - $P_0$ - 5, 10, 15 calculated by BJH method.



Figure 8. Nitrogen adsorption/desorption isotherms at 77K for MnO<sub>2</sub>-W<sub>0</sub>-10, 20, 30.



Figure 9. The pore size distribution of samples  $MnO_2$ - $W_0$ -10, 20, 30 calculated by BJH method.

3.2 Electrochemical properties of samples  $MnO_2$  with different  $P_0$  or  $W_0$ 

Typical cyclic voltammetry (CV) measurements were conducted for as-prepared samples in 0.1 M  $Na_2SO_4$  solution. The potential range was 0-0.8 V versus SCE, the measurements were taken at different scanning rates from 2 to 50 mV s<sup>-1</sup>. The average specific capacitance (SC) was calculated by the equation.

 $C = Q/(\Delta E \times m) \tag{1}$ 

Where *C* is the specific capacitance, *Q* is the half charge obtained after integrating the voltammogram, *m* is the total mess of MnO<sub>2</sub> material in the electrode, and  $\Delta E$  is the potential window.



Figure 10. Cyclic voltammograms at different scanning rates in 0.1 M Na<sub>2</sub>SO<sub>4</sub> electrolyte, MnO<sub>2</sub>- $P_0$ -5, 10, 15 (a-c), MnO<sub>2</sub>- $W_0$ -20, 30 (d-e), MnO<sub>2</sub>- $P_0$ -10 and MnO<sub>2</sub>- $W_0$ -10 are the same (b).

The CV curves of  $MnO_2$  samples at different scanning rates of 2-50 mV s<sup>-1</sup> are shown in Fig. 10. All the curves at the scanning rate of 2 mV s<sup>-1</sup> are relatively rectangular, and the current response

to the voltage reversal exhibits near mirror-image, indicating a reversible reaction and ideal capacitive behavior. According to Eq. (1), the SC of  $MnO_2-P_0-5$ , 10, 15 calculated from CV of 2 mV s<sup>-1</sup> is 117.0, 140.5 and 125.3 F g<sup>-1</sup>, respectively. And the SC of  $MnO_2-W_0-20$ , 30 calculated from CV of 2 mV s<sup>-1</sup> is 120.8 and 105.2 F g<sup>-1</sup>, respectively. When the scanning rate increases from 2 to 50 mV s<sup>-1</sup>, CV curves of  $MnO_2-P_0-5$ , 15 became distorted more dramatically than  $MnO_2-P_0-10$  as shown in Figs.10 and 11, and  $MnO_2-P_0-10$  has a better electrochemical performance. At the scanning rate 50 mV s<sup>-1</sup>, CV curves of  $MnO_2-W_0-10$  became distorted more dramatically than  $MnO_2-W_0-20$ , 30, the SC decreases drastically from 140.5 to 41.3 F g<sup>-1</sup>.



Figure 11. Dependence of the specific capacitance on the scanning rate for  $MnO_2-P_0-5$ , 10, 15,  $MnO_2-W_0-20$ , 30, and  $MnO_2-P_0-10$  and  $MnO_2-W_0-10$  are the same.

Compared with the previous work [8, 9, 22, 23], the two parameters in the microemulsion system were optimized in the present work: the content of the co-surfactant NP and water, and the relationship between the microstructure and the electrochemical property of MnO<sub>2</sub> has been studied intensively. When  $P_0=10$  and  $W_0=10$ , the as-prepared  $\gamma$ -MnO<sub>2</sub> with the surface area of 231.5 m<sup>2</sup> g<sup>-1</sup> had excellent electrochemical performance.

## **4. CONCLUSIONS**

In a quaternary microemulsion composed of CTAB, water, NP, and NH, the average size of  $MnO_2$  nanoparticles could be controlled by both the content of NP and water, and the crystalline structure also changed simultaneously. The average size of the spherical  $MnO_2$  nanoparticles decreased due to the increasing of NP content and increased accordingly to the increasing of water content. Meanwhile,  $MnO_2$  nanosheets were generated gradually by adding more NP or water. The specific capacitance of the as-prepared  $MnO_2$ - $P_0$ -10 was up to 140.5 F g<sup>-1</sup> at scanning rate of 2 mV s<sup>-1</sup>, but its poor conductivity lead to the capacitance's decay at high scanning rate. Therefore, the further studies should be focused on the introduction of conductive additive to improve the electrochemical performance of the  $MnO_2$  materials at high scanning rate.

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