

*Mini Review*

## **Progress in the Structural Design of a Titanium Dioxide Membrane and its Photocatalytic Degradation Properties**

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A great deal of attention has recently been paid to degrading polluted water using titanium dioxide because it has the characteristics of low price, nontoxicity and extensive wastewater. However, its wide energy gap, high composite rate of photoelectrons and holes, and difficult separation from wastewater limit its wide application. To improve the photocatalytic property of titanium dioxide and simplify the wastewater treatment process, recent progress in the effect of ion/atomic doping on the photocatalytic activity of titanium dioxide and the applications of solidification technology in the preparation of titanium dioxide members have been discussed, and urgent issues for future research and development are proposed.

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**Keywords:** Titanium dioxide, photocatalyst, structural design, solidification technology, doping modification, degradation

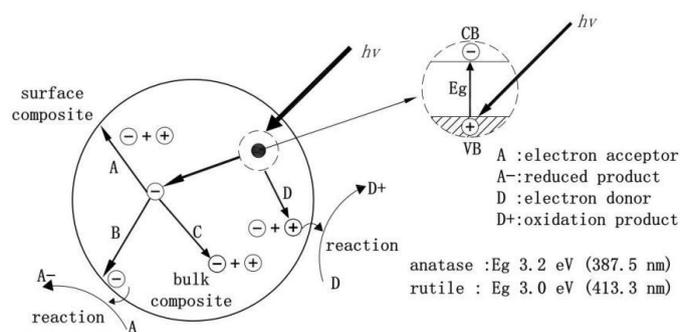
### **1. INTRODUCTION**

With the continuous intensification of anthropogenic industrial activities, environmental problems have become increasingly serious. Since Fujishima and Honda [1] first discovered in 1972 that titanium dioxide ( $\text{TiO}_2$ ) could undergo redox reaction with water under light conditions,  $\text{TiO}_2$  has attracted extensive attention from numerous scientists owing to its advantages, such as nontoxicity, no secondary pollution, low price and theoretically unlimited use. These scientists attempted to use it in the advanced treatment of drinking water and dye wastewater [2-4]. The study of  $\text{TiO}_2$  degradation of pollutants is shown in Table 1.  $\text{TiO}_2$  has been proven to be an efficient photocatalyst for the selective elimination of environmental pollutants. However, it responds primarily to UV light; for technoeconomic reasons, it is important that visible-light-driven photocatalysts with high efficiency be developed.

**Table 1.** Effect of TiO<sub>2</sub> on the photocatalytic degradation of organic pollutants

Operating condition	Pollutants	Degradation rate	Reference
3 h natural light irradiation	Methyl Blue	>90%	[5]
200 min UV-light irradiation	Methyl Orange	>80%	[6]
2 h UV-light irradiation	Methyl Blue	>99%	[7]
240 min visible light irradiation	Rhodamine B	>60%	[8]
1 h UV light irradiation	Organic Vapors (Benzene, toluene, ethyl benzene, and xylene)	>90%	[9]
40 min UV-light irradiation	Phenol	>66.8%	[10]
20 min visible-light irradiation	Gaseous Acetone	>99%	[11]
3 h UV-light irradiation	Organic Acid	>68.6%	[12]
20 min UV light	Caffeine	>99%	[13]
50 min UV-light irradiation	Acetaldehyde	>85%	[14]
12 h UV-light irradiation	Methyl Orange	>99.9%	[15]
1 h solar light	Gaseous Butane	>90%	[16]
40 min Visible-light irradiation	Escherichia coli K12	>65.9%	[17]
12 h UV-light irradiation	Diphenhydramine	>90%	[18]
5 h Visible-light	Sulfate-Reducing Bacteria	>56.3%	[19]
1 h Visible-light with 1000 W Xenon lamp	Trichlorophenol	87.8%	[20]

In Fig. 1, the light-excited photocatalytic reaction process of TiO<sub>2</sub> is displayed. Given the wide band gap of the TiO<sub>2</sub> photocatalyst ( $E_g = 3.2$  eV,  $\lambda = 387$  nm), it can only absorb ultraviolet (UV) light with wavelengths less than 387 nm, which account for only 4–6% of the sunlight wavelengths [21]. In early research concerning photocatalytic wastewater treatment with TiO<sub>2</sub>, suspension systems were mainly used, which led to problems such as the easy loss of TiO<sub>2</sub> and difficult separation. Moreover, the light absorption and blocking by suspended TiO<sub>2</sub> particles in wastewater also affected the radiation depth of light, thereby restricting the further extensive application of TiO<sub>2</sub> photocatalysts. To broaden the photoresponse range of TiO<sub>2</sub>, enhance its photocatalytic efficiency and simplify the treatment process, numerous researchers around the world have carried out a series of studies on the composition regulation of TiO<sub>2</sub> and the preparation of TiO<sub>2</sub> membranes, which are reviewed below.

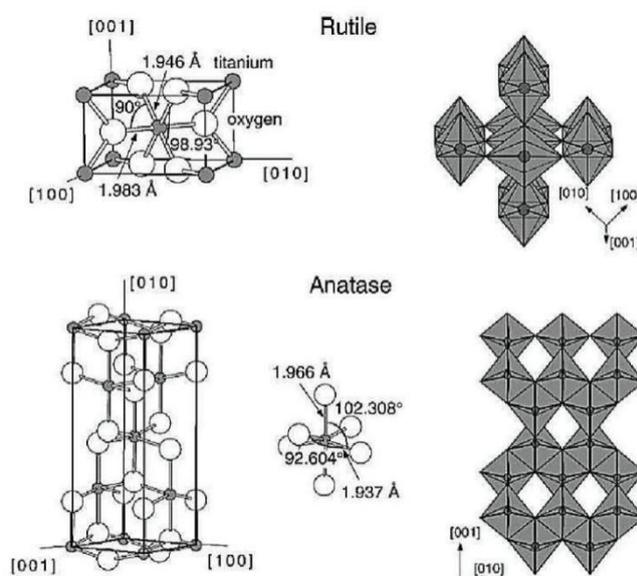
**Figure 1.** Mechanism of the photocatalytic reaction of TiO<sub>2</sub> excited by light[22]

## 2. COMPOSITION REGULATION OF TiO<sub>2</sub>

### 2.1. Crystal structure of TiO<sub>2</sub>

The TiO<sub>2</sub> used in photocatalysis is often in the rutile and anatase forms, whose crystal structures are displayed in Fig. 2. Although both are composed of titanium–oxygen octahedrons, anatase TiO<sub>2</sub> has a lower density (3.894 g/cm<sup>3</sup> vs. 4.250 g/cm<sup>3</sup>) and a wider band gap (3.2 eV vs. 3.0 eV) than rutile TiO<sub>2</sub> since it is connected by common edges and common points [23]. Additionally, anatase TiO<sub>2</sub> has rhombic tunnels in its crystal structure along the [010] and [100] directions, which thus has a macroporous structure on its surface.

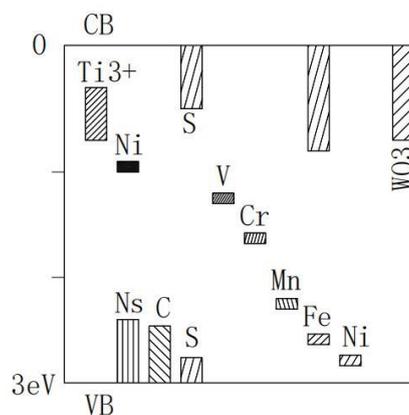
Moreover, its crystal structure is asymmetric, thus better facilitating the generation and separation of photogenerated electrons. So it is photocatalytically more active than rutile TiO<sub>2</sub> [24]. Nevertheless, anatase TiO<sub>2</sub> has a slightly worse catalytic stability than rutile TiO<sub>2</sub> due to its metastable structure. To improve the photoresponse range of TiO<sub>2</sub>, prolong its life cycle and enhance its catalytic efficiency, measures such as ion doping [25], noble metal deposition [26], semiconductor photosensitization and semiconductor compounding are currently adopted to regulate its composition.



**Figure 2.** (a) Rutile and (b) anatase structures of titanium dioxide[24]

### 2.2. Ion doping

By physically or chemically introducing ions into the target semiconductor lattice, the distortions and defects of the TiO<sub>2</sub> lattice can be promoted. Fig. 3 displays the changes in the energy band structure of TiO<sub>2</sub> doped with different ions.



**Figure 3.** Energy gap of titanium dioxide doped with different ions[27]

It is clear that the alteration of the  $\text{TiO}_2$  energy band structure by introduced foreign ions is the most deeply investigated and most widely used method for improving the photocatalytic performance of  $\text{TiO}_2$  [28-30]. Depending on the mode of implementation, ion doping can be classified into during treatment and after treatment. During treatment, ion doping is implemented simultaneously with  $\text{TiO}_2$  formation, while after treatment, ion doping is implemented after the formation of the  $\text{TiO}_2$  catalyst. Ion doping can also be classified into single doping and codoping depending on the number of doped ions and into  $M^{x+}$  doping and nonmetal ion doping depending on the type of ions. Generally, with metal ion doping,  $\text{Ti}^{4+}$  ions in the lattice are replaced to form  $M\text{-O-Ti}$  bonds, while with nonmetal ion doping,  $N_y^-$  replaces  $\text{O}^{2-}$  ions in the lattice to form  $\text{Ti-N}_y\text{-Ti}$  bonds or exists in the  $\text{TiO}_2$  lattice gaps as interstitial molecules/ions. The introduction of doped ions facilitates the generation of  $\text{TiO}_2$  oxygen vacancies and other defects, thereby promoting carrier separation.

### 2.2.1. Metal ion doping

Transitional or rare earth element ions are normally used in metal ion doping. Doping of transitional metal ions into the  $\text{TiO}_2$  lattice promotes an increase in the number of  $\text{TiO}_2$  defects. Given their multiple valence states, the transition metal ions doped into  $\text{TiO}_2$  can act as traps for photogenerated carriers. When the traps are shallow, the separation of carriers is facilitated, thereby promoting the improvement of  $\text{TiO}_2$  catalytic efficiency. When the traps are deep, they become the recombination center of carriers, thereby leading to the degraded photocatalytic performance of  $\text{TiO}_2$  [31]. When doped metal ions are 3d transition elements represented by Cr, Co, Ni, etc., new energy levels can be generated in the original forbidden band of  $\text{TiO}_2$ , resulting in a redshift of the  $\text{TiO}_2$  absorption wavelength [32]. Hoffman [33] prepared metal ion-doped  $\text{TiO}_2$  by the sol-gel process and systematically investigated the effects of various (21 types) transitional metal doping treatments on the catalytic performance of  $\text{TiO}_2$ . Their results showed that the metal ions represented by  $\text{Fe}^{3+}$  and  $\text{Ru}^{2+}$  were conducive to improving the  $\text{TiO}_2$  catalytic performance. The metal ions represented by  $\text{Cr}^{3+}$  and  $\text{Ni}^{2+}$  reduced the photocatalytic performance of  $\text{TiO}_2$ , and the transition metal ions represented by  $\text{Ga}^{3+}$  and  $\text{Zn}^{2+}$  had little impact on the photocatalytic performance of  $\text{TiO}_2$ . Notably, at excessively high doping concentrations, the transition element ions became the recombination center of carriers, which

aggravated the carrier separation to reduce the catalytic performance of TiO<sub>2</sub> instead.

Rare earth elements feature good photocatalytic performance owing to their incompletely occupied 4f orbitals and empty 5d orbitals, which are thus widely used in the photocatalytic modification and regulation of semiconductor catalysts [34]. Doping of rare earth elements can elevate the phase transition temperature of anatase TiO<sub>2</sub> and enhance the thermal stability of the catalyst. It can also alter the ligand field of TiO<sub>2</sub> and narrow its forbidden bandwidth. In addition, rare earth elements can improve the number of Lewis base centers on the TiO<sub>2</sub> surface as well, which facilitates an increase in the number of TiO<sub>2</sub> adsorption centers. Xu [35] used rare earth precursors as a doping source to prepare rare earth-doped TiO<sub>2</sub> nanoparticles by a sol–gel process, which greatly increased the photoresponse range of TiO<sub>2</sub>. Furthermore, they analyzed the redshift after doping TiO<sub>2</sub> with rare earth ions and discovered the following rule: Gd<sup>3+</sup> > Nd<sup>3+</sup> > La<sup>3+</sup> > Pr<sup>3+</sup> ≈ Er<sup>3+</sup> > Ce<sup>3+</sup> > Sm<sup>3+</sup>, which was in the opposite order of the rare earth ion radii, suggesting that the ionic radius is the major factor influencing the doping doses of rare earth ions in the TiO<sub>2</sub> lattice. After rare earth doping of TiO<sub>2</sub>, the catalytic degradation of nitrite can be achieved within the visible light range.

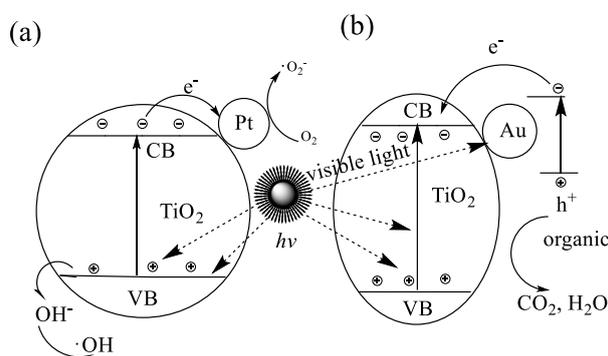
### 2.2.2. Nonmetal ion doping

Nonmetal ion doping can significantly reduce the forbidden bandwidth of TiO<sub>2</sub> and increase its range of light wave absorption. Currently usable nonmetals include N, C, P, S, B and F, of which N doping is the most investigated. Although nonmetal ion doping has been widely accepted since it can reduce the forbidden bandwidth of TiO<sub>2</sub> and improve its photocatalytic response range [36], there remains no consensus on the mechanism whereby nonmetal ions broaden the photoresponse range of TiO<sub>2</sub>. Rather extensive existing mechanism models include the fixed energy level theory represented by Nakamura [37] (holding that the doped ions form new impurity energy levels that are characterized by isolation and fixedness), the localized energy level theory represented by Di valentin [38] (believing that at high concentrations of nonmetal dopant ions, new localized energy levels can be formed above the O2p level of TiO<sub>2</sub>, which can well explain Batzill's [39] experimental results), and the hybrid energy level theory represented by Asahi [40] (through density functional theory, Asahi explored the orbital energy variations after doping of nonmetal elements into the TiO<sub>2</sub> lattice, finding that the reduction of TiO<sub>2</sub>'s forbidden band width was caused by hybridization of its O2p orbital with the p orbital in the nonmetal and that the position of the hybrid orbital depended on the concentration of doped ions). The higher the doped ion concentration is, the higher the energy level of the hybrid orbital, the smaller the forbidden band width, the larger the absorption cutoff wavelength, and the more significant the redshift of TiO<sub>2</sub>. The oxygen vacancy theory is represented by Ihara [41] (holding that the doped ions occupy the oxygen vacancies in the TiO<sub>2</sub> lattice to prevent their reoxidation, thereby leading to the redshift of TiO<sub>2</sub>). The impurity sensitization theory is represented by Sato [42] (holding that the doped impurities endow TiO<sub>2</sub> with visible light catalytic properties). Although nonmetal doping can enhance the photocatalytic activity of TiO<sub>2</sub>, the treatment process is mostly implemented under vacuum and high temperature conditions, which are thus complicated and expensive. In addition, the catalyst stability can hardly be ensured.

### 2.2.3. Rare earth ion–nonmetal ion codoping

Since rare earth element ions can promote the separation of photogenerated electrons and improve the photon quantum yield, whereas nonmetal ions can broaden the photoabsorption capacity of TiO<sub>2</sub>, codoping of appropriate rare earth elements and nonmetal ions into TiO<sub>2</sub> is conducive to enhancing the photocatalytic performance of TiO<sub>2</sub> through the synergistic effect between the two dopants [43][44]. Through a sol-gel process, Sakatani [45] doped TiO<sub>2</sub> powder simultaneously with La and N and found that codoped TiO<sub>2</sub> achieved photocatalytic degradation of acetaldehyde under visible light. Wu [46] compared the methyl orange-degrading effects between Ce–N codoped and N-doped TiO<sub>2</sub>, which revealed a higher photocatalytic efficiency of the codoped TiO<sub>2</sub>. Given the limited source of rare earth raw materials and the substantial wastewater treatment, it is difficult to actually apply codoping to industrial production.

### 2.3. Noble metal deposition



**Figure 4.** Mechanism of electron-hole separation on the surface of titanium dioxide: (a) Pt/TiO<sub>2</sub> and (b) Au/TiO<sub>2</sub>[53]

Fig. 4 describes the mechanism whereby noble metals promote electron–hole separation on the TiO<sub>2</sub> surface. Since noble metals (Ag [47], Au [48], Ru [49], Pt [50] and Pd [51]) have lower Fermi energy than TiO<sub>2</sub>, deposition of noble metal nanoparticles onto the TiO<sub>2</sub> surface can facilitate the migration of electrons in TiO<sub>2</sub> toward the noble metal particles, thereby promoting effective electron–hole separation in TiO<sub>2</sub> to improve TiO<sub>2</sub>'s photocatalytic activity. Common techniques for noble metal deposition include surface sputtering [52], impregnation reduction [53], electrodeposition [54] and photoassisted induction [55]. Apart from decreasing the specific surface area of catalysts, noble metal deposition can also affect the selectivity of TiO<sub>2</sub> photocatalysts. The deposition amount and particle size distribution of noble metals are important factors influencing catalyst performance. Kong [56] fabricated Ag-loaded TiO<sub>2</sub> nanorods by bipotential codeposition. Their results showed that after Ag loading, charge carrier recombination by the catalyst was evidently inhibited, which could improve the photocatalytic reduction efficiency of TiO<sub>2</sub> by 5 times. Nevertheless, the high price and low surface area of noble metals limit their wide application in wastewater treatment.

#### 2.4. Semiconductor photosensitization

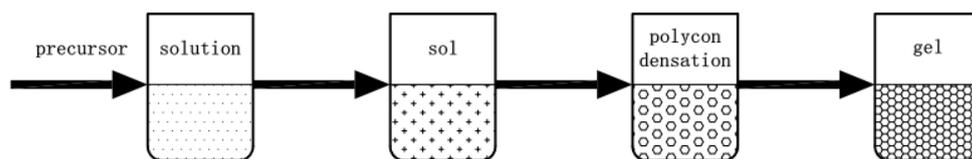
By adsorbing a photosensitizer on the TiO<sub>2</sub> surface, a photosensitized TiO<sub>2</sub> photocatalyst can be prepared. When the excited state potential of the photosensitizer is lower than the conduction band potential of the semiconductor, the photogenerated electrons on the photosensitizer surface are possibly injected into the conduction band of TiO<sub>2</sub>, which then migrate onto the catalyst surface, thereby broadening the photoabsorption wavelength range of the TiO<sub>2</sub> catalyst. Photosensitizers can be classified into two types: organic and inorganic. Organic photosensitizers mainly include chlorophyllin [57] and rose bengal [58], which have a good photosensitization effect but a poor binding force with TiO<sub>2</sub>. Meanwhile, inorganic photosensitizers are primarily complexes of various noble metals [59], which have limited application due to their complicated preparation process and high price. Regardless of whether the photosensitizer is organic or inorganic, it has a competitive adsorption relationship with TiO<sub>2</sub> and organic matter to be degraded, which thus affects the number of TiO<sub>2</sub> active centers and reduces the active specific on the surface area.

### 3. PREPARATION OF TiO<sub>2</sub> MEMBRANES

Since powdered TiO<sub>2</sub> was mostly used for the TiO<sub>2</sub> photocatalytic degradation of polluted wastewater in the early days, while the research on the loading method of active catalyst components is rather mature, many researchers have loaded TiO<sub>2</sub> powders on carriers to increase the active center number of TiO<sub>2</sub> and prolong its life cycle. Currently, carriers such as alumina [60], zeolite [61], silica gel [62], stainless steel wire mesh [63], optical fiber [64] and plexiglass [65] are mainly used around the world. Meanwhile, the sources of immobilized TiO<sub>2</sub> are primarily TiO<sub>2</sub> powder [66], TiO<sub>2</sub> precursor [67] or titanium metal [68], and the common techniques for TiO<sub>2</sub> immobilization include the sol-gel process, electrophoretic deposition, hydrothermal synthesis, chemical vapor deposition and anodization.

#### 3.1. Sol-gel process

TiO<sub>2</sub> sol is prepared through reactions (e.g., hydrolysis and condensation) of titanium alkoxides or their inorganic salts (as the raw material) under acidic conditions, and Fig. 5 illustrates the relevant preparation process. During the gelation process by aging, TiO<sub>2</sub> sol is loaded onto the carrier by means of impregnation, dip coating or spraying. In this way, TiO<sub>2</sub> precursor-loaded photocatalysts can be prepared, which is currently the most common method of preparing immobilized TiO<sub>2</sub> photocatalysts in experimental research. Exploiting this technique, Yu [69] et al. prepared TiO<sub>2</sub> photocatalysts on glass, which could effectively degrade dimethyl dichloro-vinyl phosphate under sunlight. However, there are numerous factors affecting the preparation of TiO<sub>2</sub> membranes by this technique, such as the wide particle size distribution and long production cycle of TiO<sub>2</sub>, as well as the poor bonding force between TiO<sub>2</sub> and the substrate, which thus remains in the laboratory research stage.



**Figure 5.** Preparation process of titanium dioxide by the sol-gel method[10]

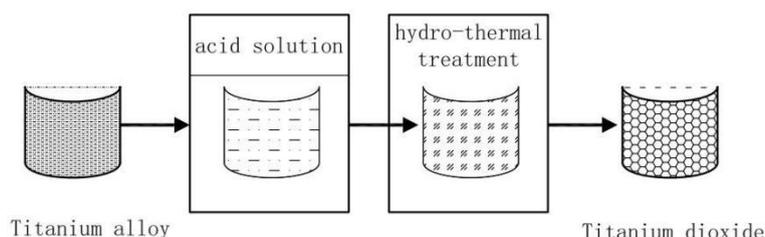
### 3.2. Electrophoretic deposition

During the immobilization of  $\text{TiO}_2$  with Pt [70], C [71], conductive glass [72] and stainless steel [73] (as the substrate material), the highly dispersed  $\text{TiO}_2$  nanopowder can be immobilized on the substrate under electric field force by using the substrate material as the cathode. Since the  $\text{TiO}_2$  membrane is prepared under electrochemical action, the quality of the membrane is closely linked to the power supply parameters. Limited by the power supply conditions, it is difficult to prepare a large-area  $\text{TiO}_2$  membrane by this technique. In addition, to ensure the continuity of the current, the substrate material needs to be conductive, thus narrowing the material choice. Mostly, electrophoretic deposition is used for the preparation of photovoltaic cell electrodes or electrically assisted photocatalysts.

### 3.3. Chemical vapor deposition

With this method, titanium-containing alkoxide or inorganic salt is used as the titanium source [74], which is vaporized by heating and deposited on the carrier surface in an inert gas. Under thermal action, the decomposition of titanium salt is promoted to form  $\text{TiO}_2$  membranes. The bonding force between the  $\text{TiO}_2$  membrane (prepared by chemical vapor deposition) and substrate is low and decreases as the coating thickness increases. Moreover, the preparation of  $\text{TiO}_2$  membranes by this technique needs to be implemented at high temperature, the process is complicated, and the cost is high.  $\text{TiO}_2$  prepared by chemical vapor deposition also has low photocatalytic activity.

### 3.4. Hydrothermal synthesis



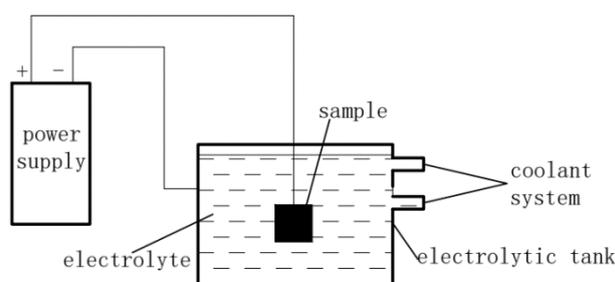
**Figure 6.** Preparation process of titanium dioxide by hydrothermal method[76]

The hydrothermal synthesis for preparing  $\text{TiO}_2$  photocatalysts is described in Fig. 6. With this

technique, the substrate material and the TiO<sub>2</sub> precursor-containing aqueous solution are placed under high temperature and pressure to facilitate TiO<sub>2</sub> decomposition and precipitation on the substrate material surface. The crystal structure and surface morphology of TiO<sub>2</sub> can be controlled by introducing hydrothermal temperature [75], reaction time [76], precursor concentration [77] and surfactant [78], thereby improving the catalytic performance of photocatalysts. Fig. 7 illustrates the conventional process of preparing TiO<sub>2</sub> photocatalysts by hydrothermal synthesis. Exploiting this technique, Aydil [79] performed orientation growth on FTO glass to obtain single-crystal rutile TiO<sub>2</sub>. Although the structure and crystal form of TiO<sub>2</sub> membranes can be regulated by hydrothermal synthesis, problems such as insufficient bonding force with the substrate and a wide pore size distribution of the membrane exist.

### 3.5. Anodizing

Fig. 7 displays a schematic of the equipment for preparing TiO<sub>2</sub> photocatalysts by anodization. Since titanium is a valve metal, when it is used as an anode and placed in an electrolyte, a TiO<sub>2</sub> membrane can be grown on its surface driven by an external electric field force. Since the anodizing preparation of TiO<sub>2</sub> nanotubes on titanium sheets by Zwilling [80] in 1999, the anodization technique has attracted extensive attention from scientists owing to the good adhesion to the substrate, simple operation, uniform texture and self-corrosion resistance of the prepared TiO<sub>2</sub> membranes. Ruan et al. [81] explored the photoelectric conversion efficiency of titanium nanotubes in 1 M KOH and borate electrolytes and found the values to be 6.8% and 7.9%, respectively. Schmuki [82] prepared a TiO<sub>2</sub> anodic oxide film in ferrous ammonium sulfate solution, where the microstructure and nanotube length of the TiO<sub>2</sub> photocatalyst were regulated by controlling the electrolyte composition and oxidation time. Therefore, TiO<sub>2</sub> photocatalysts with excellent performance were obtained. Zhang [83] prepared a TiO<sub>2</sub> membrane on a titanium metal surface in a NaOH electrolyte, thereby achieving regulation of the fiber network structure on the TiO<sub>2</sub> membrane surface, which exhibited a strong ability to degrade methylene blue. However, TiO<sub>2</sub> prepared by anodization techniques has problems such as a small absorption range of visible light and simple phase composition of catalysts, which result in insufficient efficiency of catalyst degradation.



**Figure 7.** Preparation process of titanium dioxide by anodizing method[82]

#### 4. CONCLUSION AND OUTLOOK

Conclusively, despite the rapid development of TiO<sub>2</sub> photocatalytic technology, it is still at an experimental and exploratory stage. Given the low utilization rate of sunlight by TiO<sub>2</sub>, it can hardly treat high-concentration polluted wastewater. Traditionally, TiO<sub>2</sub> is mostly prepared as a powder, which is thus difficult to recycle. Although it can be loaded onto the carrier by immobilization, the effective catalytic area of catalysts is also reduced. Moreover, the treatment process is complicated, and the life cycle is short. Hence, broadening the photoabsorption range of TiO<sub>2</sub> and preparing highly active immobilized TiO<sub>2</sub> are the two major challenges. With ion doping techniques, seeking a reasonable doping measure can broaden the photoabsorption range of TiO<sub>2</sub>, while reducing the photon-hole recombination rate is the key to improving the photocatalytic performance of TiO<sub>2</sub>. With electrochemical techniques, the microstructure and phase composition of TiO<sub>2</sub> membranes are regulated during membrane preparation, which can greatly improve the catalytic performance of catalysts and simplify the wastewater treatment process.

#### References

1. A. Fujishima and K. Honda. *Nature*, 238 (1972) 37.
2. M. Choi, J. Lim, W. Choi, W. Kim and K.J. Yong. *ACS Appl. Mater. Interfaces*, 9 (2017) 16252.
3. K.Z. Qi, B. Cheng, J.G. Yu and W. Ho. *Chin. J. Catal.*, 38 (2017) 1936.
4. H.F. Yin, Y. Cao, T.L. Fan, B. Qiu and S. Chen. *J. Alloys Compd.*, 824 (2020) 153915.
5. X. Li, J. Sun, G. He, G. Jiang, Y. Tan and B. Xue. *J. Colloid Interface Sci.*, 411 (2013) 34.
6. C. Zhang, Q.L. Li and J.Q. Li. *Synth. Met.*, 160 (2010) 1699.
7. Q. Luo, X. Li, D. Wang and J. An. *J. Mater. Sci.*, 46 (2011) 1646.
8. F. Duan, Q. Zhang, D. Shi and M. Chen. *Appl. Surf. Sci.*, 268 (2013) 129.
9. A.F. Ghanem, A.A. Badawy, N. Ismail, Z.R. Tian, M.H.A. Rehim and A. Rabia. *Appl. Catal. A*, 472 (2014) 191.
10. M.R. Golobostanfard and H. Abdizadeh. *Microporous Mesoporous Mater.*, 183 (2014) 74.
11. X. Nie, G. Li, P. Wong and H. Zhao, *Catal. Today*, 230 (2014) 67.
12. L.M. Pastrana-Martínez, S. Morales-Torres, S.K. Papageorgiou, F.K. Katsaros, G.E. Romanos, J.L. Figueiredo, J.L. Faria, P. Falaras and A.M.T. Silva. *Appl. Catal. B-Environ.*, 142-143 (2013) 101.
13. P. Gao, Z. Liu, M. Tai, D.D. Sun and W. Ng. *Appl. Catal. B-Environ.*, 138-139 (2013) 17.
14. B. Long, J. Lin and X. Wang. *J. Mater. Chem. A*, 2 (2014) 2942.
15. H. Lee, B.J. Kim, Y.K. Park, J. Kim and S. Jung. *Catal. Today*, 355 (2020) 435.
16. C.S. Tsao, C.M. Chuang, C.Y. Chen, Y.C. Huang, H.C. Cha, F.H. Hsu, C.Y. Chen, Y.C. Tu and W.F. Su. *J. Phys. Chem. C*, 118 (2014) 26332.
17. R. Singh and S. Dutta. *Fuel*, 220 (2018) 15 607.
18. M. Gharagozlou and R. Bayati. *Ceram. Int.*, 40 (2018) 10247.
19. Y. Li, J. Wang, J.Y. Zhang, D. He, Q. An and G. Gao. *J. Hazard. Mater.*, 292 (2015) 79.
20. S. Leong, A. Razmjou, K. Wang, K. Hapgood, X. Zhang and H. Wang. *J. Membr. Sci.*, 472 (2014) 167.
21. J.B. Joo, Q. Zhang, M. Dahl, F. Zaera and Y. Yin. *J. Mater. Res.*, 28 (2013) 362.
22. W.Y. Choi, A. Terimn and M.R. Hoffmann. *J. phys. Chem. C*, 98 (1994) 13669.
23. N. Fajrina and M. Tahir. *Int. J. Hydrogen Energ.*, 44 (2019) 540.
24. A. Imran, S. mohd, Z.A. Alothman and A. Abdulrahman. *RSC Adv.*, 8 (2018) 30125.
25. V. Kurnaravel, S. Mathew, J. Bartlett and S.C.Pillai. *Appl. Catal. B*, 244 (2019) 1021.

26. V.M. Menéndez-Flores and T. Ohno. *Catal. Today*, 230 (2014) 214.
27. O. Carp, C.L. Huisman and A. Reller. *Prog. Solid State Ch.*, 32 (2004) 33.
28. V. Etacheri, C.D. Valentin, J. Schneider, D. Bahnemann and S.C. Pillai. *J. Photochem. Photobiol. C*, 25 (2015) 1.
29. C. Acar, I. Dincer and C. Zamfirescu. *Int. J. Energ. Res.*, 38 (2014) 1903.
30. R. Nawaz, C.F. Kait, H.Y. Chia, M.H. Isa and L.W. Huei. *Environ. Technol.*, 19 (2020) 101007.
31. A. Houas, H. Lachheb, M. Ksibi, E. Elaloui, C. Guillard and J. Herrmann. *Appl. Catal. B*, 31 (2001) 145.
32. X.P. Wang and T.Lim. *Appl. Catal. B*, 100 (2010) 355.
33. D.Z. Li, Z.X. Chen, Y.L. Chen, W.J. Li, H.J. Huang, Y.H. He and X.Z. Fu. *Environ. Sci. Technol.*, 42 (2008) 2130.
34. Z.S. Guo, H. Wu, M. Li, T. Tang, J. Wen and X. Li. *Appl. Surf. Sci.*, 526 (2020) 146724.
35. H. Li, Y.B. Hao, H.Q. Lu, L.P. Liang, Y.Y. Wang, J.H. Qiu, X.C. Shi, Y. Wang and J.F. Yao. *Appl. Surf. Sci.*, 344 (2015) 112.
36. S.Y. Fang, Z.X. Sun and Y.H. Hu. *ACS Catal.*, 9 (2019) 5047.
37. H. Park, Y. Park, W. Kim and W. Choi. *J. Photochem. Photobiol. C*, 15 (2013) 1.
38. P. Mazierski, M. Nischk, M. Gołkowska, W. Lisowski, M. Gazda, W. Micha, T. Klimczuk and A. Zaleska-Medynska. *Appl. Catal. B*, 196 (2016) 77.
39. D. Li, H. Haneda, N. Labhsetwar, S. Hishita and N. Ohashi. *Chem. Phys. Lett.*, 401 (2005) 579.
40. J.Q. Yan, T. Wang, G.J. Wu, W. Dai, N. Guan, L. Li and J. Gong. *Adv. Mater.*, 27 (2015) 1580.
41. W. Wang, X.M. Xu, W. Zhou and Z.P. Shao. *Adv. Sci.*, 4 (2017) 1600371.
42. Y. Cao, Z.P. Xing, Y.C. Shen, Z. Li, X. Wu, X. Yan, J. Zou, S. Yang and Z. Wei. *Chem. Eng. J.*, 325 (2017) 199.
43. N. Nursam, X. Wang, J. Tan and R.A. Caruso. *ACS Appl. Mater. Interfaces*, 8 (1944) 17194.
44. N. Hirabuki, Y. Watanabe, T. Mano, N. Fujita, H. Tanaka, T. Ueguchi and H. Nakamura. *Am. J. Neuroradiol.*, 21 (2020) 1497.
45. C.D. Valentin, E. Finazzi, G. Pacchioni, A. Selloni and E. Giamello. *Chem. Mater.*, 20 (2008) 3706.
46. X.Q. Gong, A.A. Selloni, M. Batzill and U. Diebold. *Nat. Mater.*, 5 (2006) 665.
47. R. Asahi, T. Morikawa, T. Ohwaki, K. Aoki and Y. Taga. *Science*, 293 (2001) 269.
48. I. Nakamura, N. Negishi, S. Kutsuna, T. Ihara, S. Sugihara and K. Takeuchi. *J. mol. Catal. A-Chem.*, 161 (2000) 205.
49. S. Sato. *Chem. Phys. Lett.*, 123 (1986) 126.
50. X. Zhang, K. Udagawa, Z. Liu, S. Nishimoto, C. Xu, Y. Liu, H. Sakai, M. Abe, T. Murakami and A. Fujishima. *J. Photoch. Photobio. A.*, 202 (2009) 39.
51. F. Meng, S.K. Cushing, J. Li, S. Hao and N. Wu. *ACS Catal.*, 5 (2015) 1949.
52. Y. Ide, N. Inami, H. Hattori, K. Saito, M. Sohmiya, N. Tsunoji, K. Komaguchi, T. Sano, Y. Bando and D. Golberg. *Angew. Chem. Int. Edit.*, 128 (2016) 3664.
53. C. Pan, T. Takata, M. Nakabayashi, T. Matsumoto and K. Domen. *Angew. Chem. Int. Edit.*, 127 (2015) 2998.
54. N. Tohge, K. Tadanaga, H. Sakatani and T. Minami. *J. Mater. Sci.*, 15 (1996) 1517.
55. X.J. Yu, L.L. Xiong, G.P. Ma, Y. Liang and K. Liu. *J. Rare Earth.*, 33 (2014) 26.
56. P. Karishma, C. Bahrim, S. Twagirayezu and T.J. Benson. *Adv. Catal.*, 66 (2020) 109.
57. J.M. Du, J.L. Zhang, Z.M. Liu, B.X. Han and Y. Huang. *Langmuir*, 22 (2006) 1307.
58. S.A. Ansari, M.M. Khan, M.O. Ansari and M.H. Cho. *New J. Chem.*, 39 (2015) 4708.
59. P.S. Awati, S.V. Awate, P.P. Shah and V. Ramaswamy. *Catal. Commun.*, 4 (2003) 393.
60. Q.H. Zhang, W.D. Han, Y.J. Hong and J.G. Yu. *Catal. Today*, 148 (2009) 335.
61. C. Belver, M.J. Lopez-Muoz, J.M. Coronado and J. Soria. *Appl. Catal., B-environ.*, 46 (2003) 497.
62. C.K. Jung, S.H. Cho, S.B. Lee, T.K. Kim, M.N. Lee and J.H. Boo. *Surf. Rev. Lett.*, 10 (2008) 635.
63. S. Kurajica, J. Macan, V. Mandic, M. Galjer, K. Mužina and J.R. Plaisier. *Mater. Res. Bull.*, 105 (2018) 142.

64. B. Cangul, L.C. Zhang, M. Aindow and C. Erkey. *J. Supercrit. Fluids*, 50 (2009) 82.
65. R.Y. Chen, V. Trieu, B. Schley, H. Natter, J. Kintrup, A. Bulan, R. Weber and R. Hempelmann. *Z. Phys. Chem.*, 227 (2013) 651.
66. L. Guo, L.B. Huang, W.J. Jiang and Z.D. Wang. *J. Mater. Chem. A*, 5 (2000) 9014.
67. M.C. Vebber, A.C.R. Faria, N. DalAcqua, L.L. Beal, G. Fetter, G. Machado, M. Giovanela and J.S. Crespo. *Int. J. Hydrogen Energy*, 41 (2016) 17995.
68. L.L. Li, B. Cheng, Y.X. Wang and J. Yu. *J. Colloid Interface Sci.*, 449 (2015) 115.
69. S. Obregon, M.J. Munoz-Batista, M. Fernandez-Garcia, A. Kubacka and G. Colón. *Appl. Catal. B*, 179 (2015) 468.
70. P. Srivastava and L. Bahadur. *Int. J. Hydrogen Energy*, 37 (2012) 4863.
71. M. Kitano and M. Hara. *J. Mater. Chem.*, 20 (2010) 627.
72. J. Liu, L. Yang, N. Liu, Y. Han, Z. Xing, H. Huang, Y. Lifshitz, S.T. Lee, J. Zhong and Z. Kang. *Science*, 347 (2015) 970.
73. S. Ghosh, N.A. Kouame, L. Ramos, S. Remita, A. Dazzi, A. Deniset-Besseau, P. Beaunier, F. Goubard, P.H. Aubert and H. Remita. *Nat. Mater.*, 14 (2015) 505.
74. J.B. Fei and J.B. Li. *Adv. Mater.*, 27 (2014) 314.
75. R. Fagan, D.E. McCormack, S. Hinder, S. Hinder and S.C. Pillai. *Mater. Design*, 96 (2016) 44.
76. A.A. Ismail and D.W. Bahnemann. *Sol. Energy Mater. Sol. C.*, 128 (2014) 85.
77. M.T. Chao and A.R. Mohamed. *J. Alloys Compd.*, 509 (2011) 1648.
78. R. Jaiswal, J. Bharambe, N. Patel, A. Dashora, D.C. Kothari and A. Miotello. *Appl. Catal. B*, 168 (2015) 333.
79. S. Kuriakose, B. Satpati and S. Mohapatra. *Phys. Chem. Chem. Phys.*, 17 (2015) 2517225181.
80. R. Abe. *J. Photoch. Photobio. C*, 11 (2010) 179.
81. J. Yuan, M.X. Chen, J.W. Shi and W.F. Shangguan. *Int. J. Hydrogen Energy*, 31 (2006) 1326.
82. M.N. Chong, B. Jin, C.W.K. Chow and C. *Water Res.*, 44 (2010) 2997.
83. Y. Zhang, Z.Y. Zhao, J.R. Chen, L. Cheng, J. Chang, W.C. Sheng, C.Y. Hu and S.S. Cao. *Appl. Catal. B*, 165 (2015) 715.