

# Constructing Fe<sub>3</sub>O<sub>4</sub>/Nitrogen-doped Graphene Composites with Rich Pyrrolic Nitrogen for Excellent Supercapacitor Performance

Jugong Zheng, Ting Yang, Jianfa Chen, Jianping Xu\*, Tianxiang Jin\*

Jiangxi Engineering Research Center of Process and Equipment for New Energy, East China University of Technology, Nanchang 330013, P.R. China

\*E-mail: [201660027@ecut.edu.cn](mailto:201660027@ecut.edu.cn) (T. X. Jin), [xujp@ecut.edu.cn](mailto:xujp@ecut.edu.cn) (J. P. Xu)

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Nitrogen-doped graphene (NG)-supported Fe<sub>3</sub>O<sub>4</sub> nanoparticle composites were prepared via a facile solvothermal and calcination process. The experimental results showed that the Fe<sub>3</sub>O<sub>4</sub> nanoparticles were evenly embedded in the surface of NG without agglomeration and that pyrrolic N was the main N binding form in NG. The as-prepared Fe<sub>3</sub>O<sub>4</sub>@NG composites displayed a good specific capacitance of 244.3F/g at a current density of 1A/g. Furthermore, the Fe<sub>3</sub>O<sub>4</sub>/NG composites exhibited a low capacitance loss after 1000 cycles.

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**Keywords:** Nitrogen-doped graphene, Fe<sub>3</sub>O<sub>4</sub> nanoparticles, supercapacitor, electrode material

## 1. INTRODUCTION

Generally electrochemical methods are sensitive, stable, and accurate [1-3] and have been widely used in applications, such as electrochemical biosensors [4-6], supercapacitors [7], fuel cells [8, 9]. As one of the most indispensable energy storage devices, supercapacitors have many notable features such as multifunctionality, excellent cycle stability, and low cost. [10, 11] Nevertheless, the low energy density of supercapacitors limits their further application. Therefore, it is significant to find effective strategies for enhancing the energy density of supercapacitors without reducing their power density and cycle stability. [12, 13] Facts show that the electrode materials have a significant impact on the performance of supercapacitors. Many studies have been conducted on the use of transition metal oxides as electrode materials because of their large pseudocapacitance. [14, 15] However, pseudocapacitor materials have poor rate capacities and cyclic stabilities during charging and discharging; specifically, the cyclic stability is far less stable than that of the double layer capacitance. [15] Therefore, a hybrid material that is composed of carbon-based materials and metal oxides could be a high-performance electrode material because it has the advantages of integrating an electronic double-layer capacitor

(EDLC) and a pseudocapacitor. [16, 17] Many composites that involve the combination of metal oxides (such as  $\text{Co}_3\text{O}_4$ , [18]  $\text{MnO}_2$ , [19, 20]  $\text{MoO}_2$ , [21]  $\text{V}_2\text{O}_5$ , [22]  $\text{SnO}_2$  [23]) and carbon-based materials generally exhibit excellent electrochemical energy storage performance. [24, 25]

$\text{Fe}_3\text{O}_4$  is an attractive material for pseudocapacitor electrodes, and has the advantages of a wide range for the operational potential window ( $-1.2$  to  $0.25\text{V}$ ), low cost, large theoretical specific capacitance ( $2299\text{F/g}$ ), and excellent conductivity ( $10^2$ - $10^3\text{S/cm}$ ). [26-28] The surface morphology, [29] crystallinity, [30] and particle distribution [31] of  $\text{Fe}_3\text{O}_4$  largely influence the supercapacitive performance. Khoh synthesized multilayer films made of  $\text{Fe}_3\text{O}_4$  nanoparticles and reduced graphene oxide (RGO) using a layer-by-layer progressive approach. The as-synthesized multilayer films exhibited excellent capacitive performances with  $151\text{ F/g}$  at  $0.9\text{ A/g}$ . [32] Meng synthesized porous  $\text{Fe}_3\text{O}_4$ /carbon composites via calcination of an iron-based metal organic framework. This material displayed a specific capacitance of  $139\text{ F/g}$  at  $0.5\text{ A/g}$ . [33] Ghasemi synthesized an RGO/ $\text{Fe}_3\text{O}_4$  composite via a simple electrophoretic deposition and electrochemical reduction method. This nanocomposite displayed a specific capacitance of  $154\text{ F/g}$  at  $1\text{ A/g}$ . [34]

Although all the above composites have good energy storage properties, the synthesis methods of these composites are extremely sophisticated, and it is difficult to control the final microstructure of  $\text{Fe}_3\text{O}_4$ . It is known that metal oxide particles that have large specific surface area generally have a high specific capacitance. [35] However, magnetic nano- $\text{Fe}_3\text{O}_4$  particles have a tendency to agglomerate during the preparation process, and this leads to larger sized particles. [28] Therefore, if the specific surface area of  $\text{Fe}_3\text{O}_4$  on the RGO surface can be increased, the material can, in theory, exhibit better electrochemical performance. [36]

Graphene is a new type of carbon material and has the characteristics of good conductivity and large specific surface area, which make it suitable for energy storage devices [37]. Moreover, graphene can also act as a support network for embedding  $\text{Fe}_3\text{O}_4$  nanoparticles to prevent  $\text{Fe}_3\text{O}_4$  from agglomerating. [28] However, graphene has low polarity, and it usually has poor interaction with metal oxides, and this is not conducive for the immobilization and dispersion of  $\text{Fe}_3\text{O}_4$  magnetic nanoparticles. To solve the above problem, we attempted to introduce nitrogen atoms into the graphene lattice to enhance the interaction between graphene and  $\text{Fe}_3\text{O}_4$ . [38, 39]

Thus, we developed a simple hydrothermal and annealing approach for preparing  $\text{Fe}_3\text{O}_4$ /N-doped graphene ( $\text{Fe}_3\text{O}_4@\text{NG}$ ) nanocomposites. We found that the pyrrolic N was the major sort of N in the NG that was synthesized via this method. Pyrrolic N has good electron-donor characteristics that can efficiently improve the electrochemical performance of electrode materials. Electrochemical testing was used to test the composites, and the as-obtained  $\text{Fe}_3\text{O}_4@\text{NG}$  composite-based supercapacitors exhibited excellent performance.

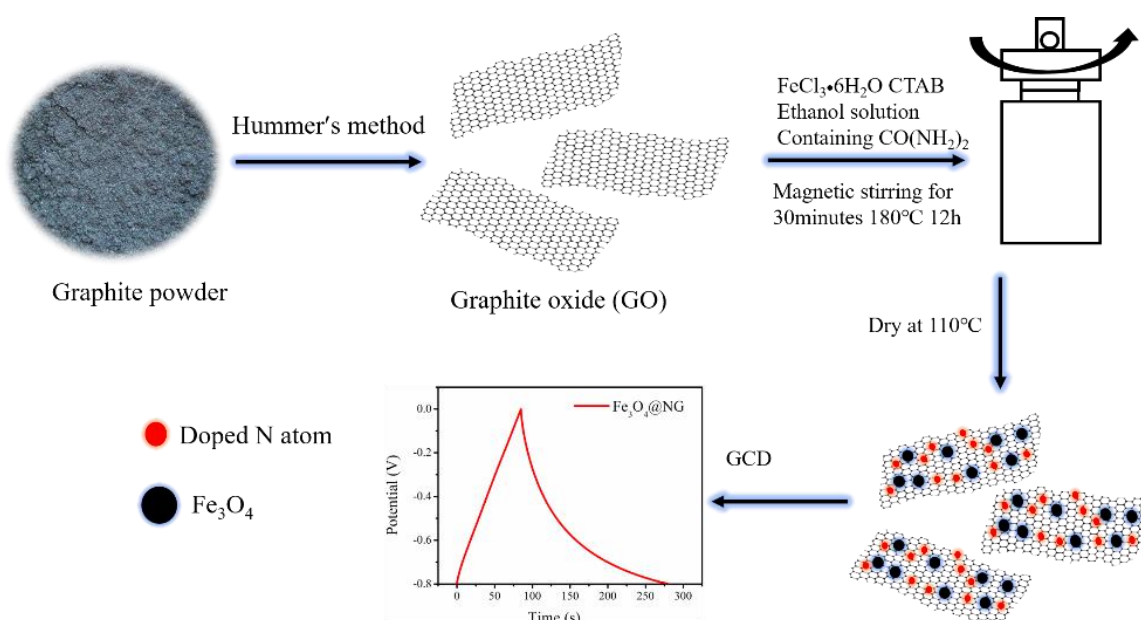
## 2. EXPERIMENTAL

### 2.1. Chemicals

$\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$ , cetyltrimethylammonium bromide (CTAB), urea, and ethylene glycol (EG) were obtained from Sigma-Aldrich.  $\text{KMnO}_4$ ,  $\text{KOH}$ ,  $\text{NaNO}_3$ , 98%  $\text{H}_2\text{SO}_4$ , 30%  $\text{H}_2\text{O}_2$ , and graphite powder were obtained from China Medicine Co. All the reagents were of analytical grade.

## 2.2 Preparation of $Fe_3O_4$ @ NG composites

GO was synthesized from graphite powder via a modified improved Hummer method [40]. Ultrasonication was used for 30min to disperse GO in aqueous solution (2mg/mL, 50mL). Subsequently, 1.35g  $FeCl_3 \cdot 6H_2O$ , 0.5g CTAB, and 0.5mL EG were introduced in sequence and stirred for 0.5h. Then, 3g of urea that was dissolved in 25mL ethanol solution was added dropwise into the system, which was stirred for 0.5h. After that, the reaction solution was transferred into a 100mL Teflon-lined stainless steel autoclave that was maintained at  $180^\circ C$  for 12h. After the autoclave was cooled to room temperature, the product was washed several times with DI water and anhydrous ethanol. Finally, the product was dried under vacuum at  $80^\circ C$  for 24h. For comparison, NG was prepared without the added of  $FeCl_3 \cdot 6H_2O$  using the same procedure. The synthetic route of the  $Fe_3O_4$ @NG composite is shown in Figure 1.



**Figure 1.** Schematic diagram of the synthesis process for  $Fe_3O_4$ @NG composites.

## 2.3 Characterization of materials

TEM observations were conducted using a JEM-2100F transmission electron microscope. XPS analyses were performed using a Thermo Fisher X-ray photoelectron spectrometer (Al  $K\alpha$  radiation).

## 2.4 Electrochemical Characterization

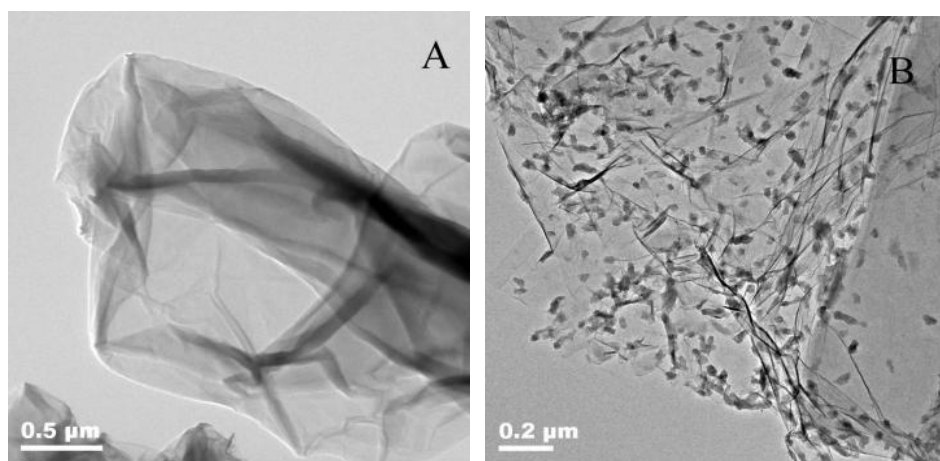
The working electrodes were fabricated using electroactive material, acetylene black, and polytetrafluoroethylene (PTFE) binder in a weight ratio of 80:15:5. These electrodes were pressed onto Ni-grids under a pressure of 15MPa. The area of the electrode was  $1cm^2$ . Electrochemical performances were assessed using cyclic voltammetry (CV) and galvanostatic charge-discharge (GCD) tests. All of the tests were conducted on a CHI 660E electrochemical cell equipped with a 3-electrode system in 1M

Na<sub>2</sub>SO<sub>4</sub> solution at room temperature. The specific capacitance (Cs) was calculated from the GCD curves using equation (1).

$$Cs = \frac{I \cdot t}{m \cdot \Delta U} \quad (1)$$

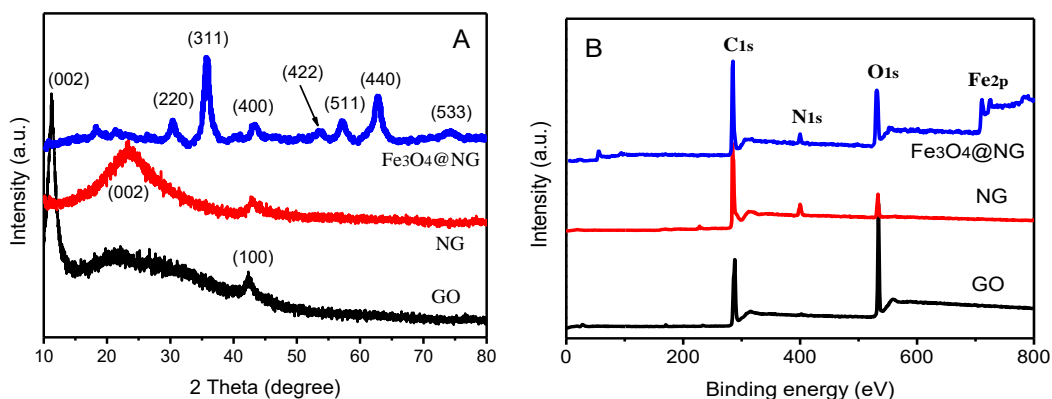
where *I* represents the discharge current, *t* is to the discharge time, *m* denotes the mass of the composites, and  $\Delta V$  represents the voltage window.

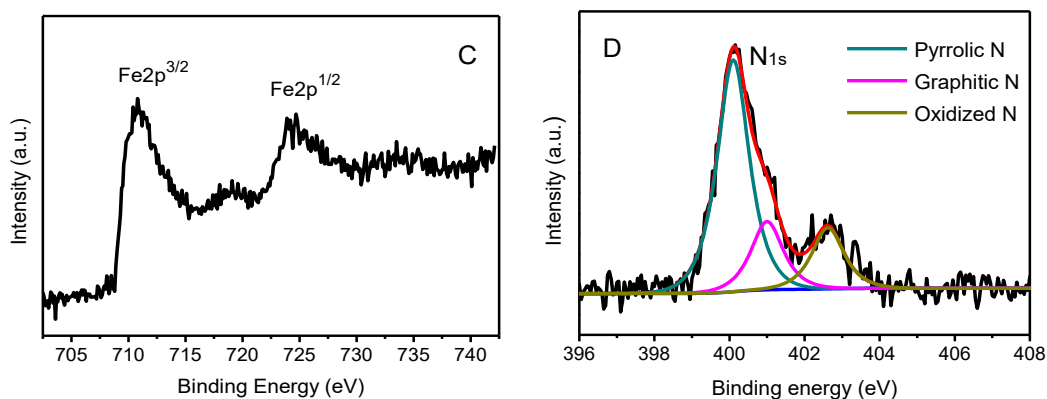
### 3. RESULT AND DISCUSSION



**Figure 2.** TEM images of (A) NG and (B) Fe<sub>3</sub>O<sub>4</sub>@NG.

The morphology of NG and Fe<sub>3</sub>O<sub>4</sub>@NG was assessed via TEM observations. As seen in Fig. 2A, NG exhibited a crumpled silk veil morphology. An image of as-prepared Fe<sub>3</sub>O<sub>4</sub>@NGs is displayed in Fig. 2B. Fe<sub>3</sub>O<sub>4</sub> nanoparticles with a small size (of about 20 nm) were uniformly embedded in the surface of NG without agglomeration. This result may be attributed to the strong interaction between Fe<sub>3</sub>O<sub>4</sub> and NG, which restrains the agglomeration of Fe<sub>3</sub>O<sub>4</sub> nanoparticles.

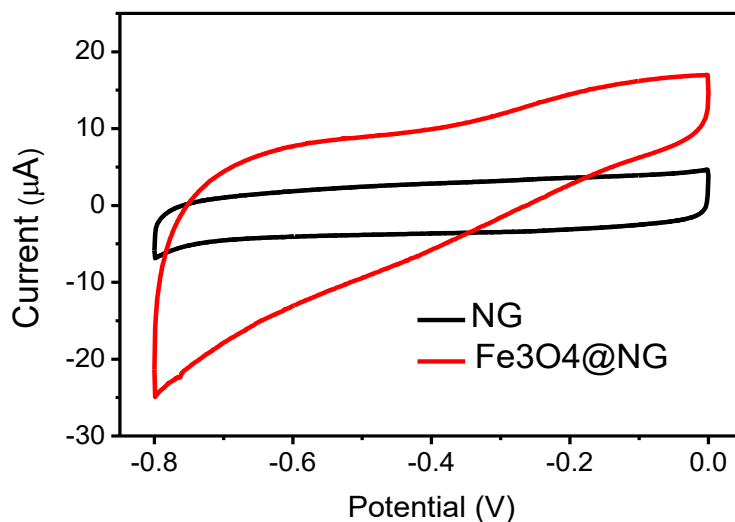




**Figure 3.** (A) XRD patterns and (B) XPS survey spectra of GO, NG, and Fe<sub>3</sub>O<sub>4</sub>@NG and high resolution XPS spectra of (C) Fe2p and (D) N1s.

Figure. 3A shows XRD patterns, and as seen, a sharp diffraction peak of GO appeared at  $2\theta = 11.2^\circ$ , corresponding to the (002) crystal plane of carbon. This peak disappeared after reduction. Instead, a broad peak that corresponds to the (002) crystal plane of carbon appeared at  $23.5^\circ$  in NG, and this indicates poorly ordered graphene sheets. The diffraction peaks of Fe<sub>3</sub>O<sub>4</sub>@NG are at  $2\theta = 30.5^\circ, 35.8^\circ, 43.6^\circ, 57.2^\circ, 62.9^\circ, \text{ and } 73.9^\circ$  and can be perfectly assigned to the diffraction planes (220), (311), (400), (422), (511) and (440), respectively. These results are in accord with the standard data (JCPDS no.19-0629). In particular, a sharp diffraction peak appeared at  $35.8^\circ$ , and this proves that Fe<sub>3</sub>O<sub>4</sub> nanoparticles were loaded on the surface of NG and had a better crystal form without impurities.

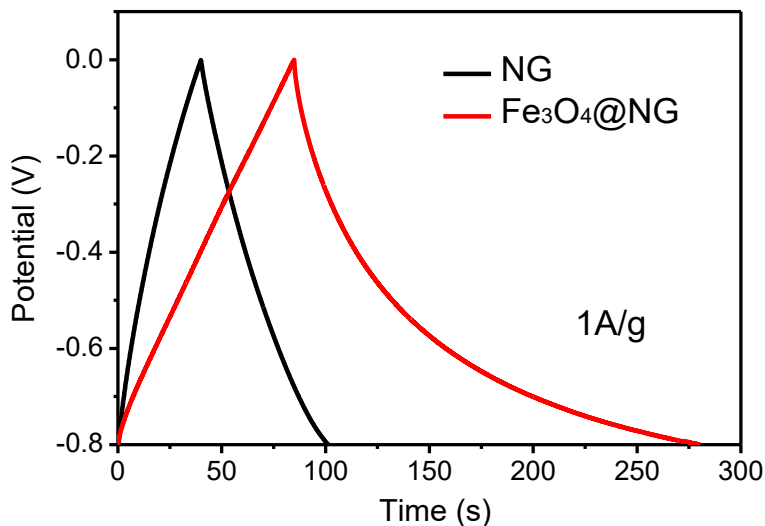
X-ray photoelectron spectroscopy (XPS) was used to analyze the chemical composition of GO, NG, and Fe<sub>3</sub>O<sub>4</sub>@NG. As seen in Fig. 3B, C1s and O1s characteristic peaks appeared in the XPS spectrum of GO. After the nitrogen doping process, a new peak that corresponds to N1s appeared at a binding energy (BE) of 400 eV. The appearance of this peak confirms the successful incorporation of nitrogen into the carbon framework. Moreover, compared with the spectrum of GO, the O1s peak of NG at 532.2 eV was significantly weaker, and this indicates that GO was reduced during the doping process. Fig.2B also shows the characteristic peaks of the C 1s, N 1s, O 1s, and Fe 2p in the XPS spectra, proving that carbon, oxygen, nitrogen, and iron elements were present in the Fe<sub>3</sub>O<sub>4</sub>@NG composites. The results clearly indicate the successful incorporation of Fe<sub>3</sub>O<sub>4</sub> nanoparticles into the carbon nanosheets. Compared with NG, Fe<sub>3</sub>O<sub>4</sub>@NG showed a significant increase in the O1s peak. These results are consistent with previous reports and further verify that Fe<sub>3</sub>O<sub>4</sub> particles were successfully loaded on the surface of graphene. In the high resolution XPS Fe 2p spectrum (Fig.3C), there are two strong peaks at 711.8eV and 724.2 eV that correspond to Fe2p<sub>3/2</sub> and Fe2p<sub>1/2</sub> for Fe<sub>3</sub>O<sub>4</sub>. As seen in Fig.3D, the N1s peak can be deconvoluted into three components at 399.9 eV, 401.2 eV and 402.6 eV, which correspond to pyrrolic N, graphitic N, and oxidized N, respectively. Pyrrolic N is the dominant species in Fe<sub>3</sub>O<sub>4</sub>@NG composites. Pyrrolic N has a good electron donor property, and can effectively improve electrochemical performance. [41,42]



**Figure 4.** CV curves of NG and Fe<sub>3</sub>O<sub>4</sub>@NG composite modified electrodes at a scan rate of 20 m V/s in 1.0 M Na<sub>2</sub>SO<sub>4</sub> solution.

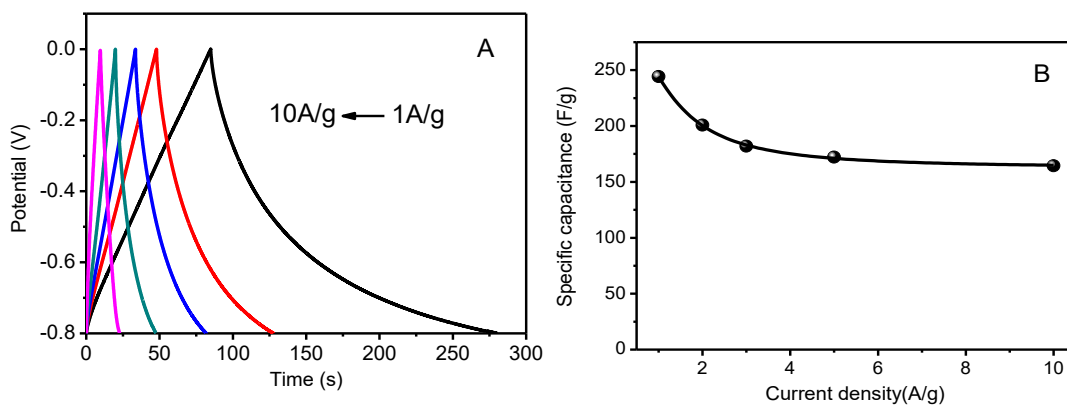
CV curves of NG and Fe<sub>3</sub>O<sub>4</sub>@NG were recorded at a scan rate of 20mV/s at a three-electrode system with 1.0 M Na<sub>2</sub>SO<sub>4</sub> as the electrolyte solution. As seen in Fig.4, the CV curve of Fe<sub>3</sub>O<sub>4</sub>@NG exhibited a near-rectangular shape with obvious redox peaks, and this indicates that the capacitance of Fe<sub>3</sub>O<sub>4</sub>@NG was the combination of the electric double-layer capacitance of NG and the Faraday pseudocapacitance from both NG and the Fe<sub>3</sub>O<sub>4</sub> nanoparticles. It is known that the specific capacitance of an electrode is proportional to the area of the detected CV curve. Obviously, the enclosed area of the Fe<sub>3</sub>O<sub>4</sub>@NG composite was evidently larger than that of the NG electrode at the same scan rates, and this indicates that the Fe<sub>3</sub>O<sub>4</sub>@NG composite electrode has excellent capacitance.

GCD technology was used to further detect the capacitance performance of the NG electrode and the Fe<sub>3</sub>O<sub>4</sub> @ NG composite electrode. GCD curves of the electrodes were recorded at a current density of 1 A/g within a potential range of -0.8–0 V. As seen in Fig.5, both the NG electrode and the Fe<sub>3</sub>O<sub>4</sub>@NG electrode displayed shapes that were quasi-isosceles triangles. The calculated specific capacitance of Fe<sub>3</sub>O<sub>4</sub>@NG is 244.3 F/g, which higher than that of NG (120.1 F/g). The larger specific capacitance of Fe<sub>3</sub>O<sub>4</sub>@NG is mainly attributed to the synergistic effect between Fe<sub>3</sub>O<sub>4</sub> and NG. On one hand, NG provides a high specific surface area and enhances the dispersion of Fe<sub>3</sub>O<sub>4</sub> nanoparticles. On the other hand, Fe<sub>3</sub>O<sub>4</sub> can prevent agglomeration of NG. Because of this effect, both NG and the Fe<sub>3</sub>O<sub>4</sub> nanoparticles had a large available surface; this can facilitate electron transport and promotes redox reactions across the electrode. [43] Also, the pyrrolic-N-rich structure of NG plays an important role in promoting the electrochemical performance.

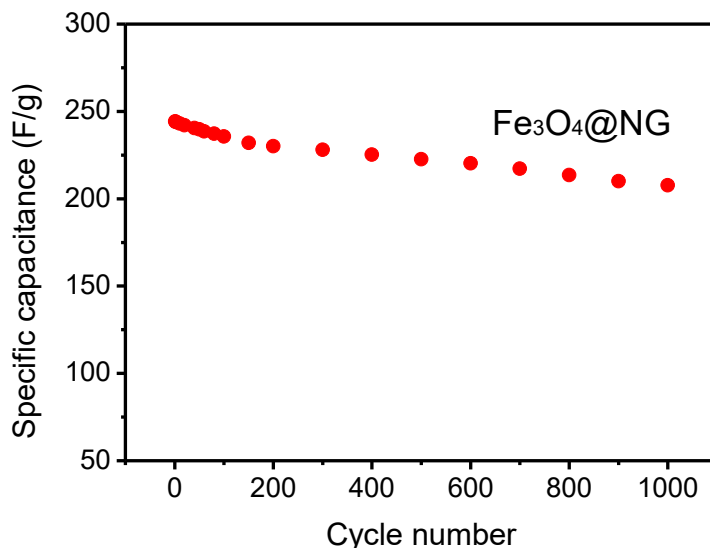


**Figure 5.** Constant current charge-discharge curves of NG and Fe<sub>3</sub>O<sub>4</sub>@NG

The specific capacitances of Fe<sub>3</sub>O<sub>4</sub>@NG under different current densities (1, 2, 3, 5, and 10 A/g) are shown in Fig. 6. The specific capacitances were calculated to be 244.3F /g, 198.7F /g, 182.5F /g, 168.8 F /g, and 161.2F /g at 1, 2, 3, 5,10 A/g, respectively. The corresponding specific capacitance retention rates are 100%, 81.33%, 74.70%, 69.09%, and 65.98%, respectively. These results prove that the Fe<sub>3</sub>O<sub>4</sub>@NG composites have good capacitance ratios at high current densities.



**Figure 6.** (A) Charge-discharge curves of the Fe<sub>3</sub>O<sub>4</sub>@NG electrode at current densities of 1, 2, 3, 5, and 10A/g. (B) Specific capacitance curves of Fe<sub>3</sub>O<sub>4</sub>@NG at different current densities.



**Figure 7.** Changes in  $C_s$  for  $\text{Fe}_3\text{O}_4@\text{NG}$  composites with the number of cycles at a current density of 1 A/g.

The cycling stabilities of the NG and  $\text{Fe}_3\text{O}_4@\text{NG}$  electrodes were tested up to 1000 cycles, as displayed in Fig.7. The specific capacity of the  $\text{Fe}_3\text{O}_4@\text{NG}$  composite electrode decreased from an initial value of 244.3F/g to a value of 207.7F/g after 1000 cycles, and the specific capacitance still remained 85.02% of the initial capacity. These results prove that  $\text{Fe}_3\text{O}_4@\text{NG}$  has an advantageous long-term electrochemical stability. This excellent performance may be attributed to the strong interaction between nitrogen-doped graphene and  $\text{Fe}_3\text{O}_4$  nanoparticles, which inhibited the destruction of the microstructure during redox reaction.

**Table 1.** Comparison of specific capacitance of different Fe-based composite electrode materials

Electrode Materials	Electrolyte	$C_s(\text{F/g})$
CNT/ $\text{Fe}_3\text{O}_4$ [44]	6 M KOH	117.2
$\text{Fe}_3\text{O}_4/\text{rGO}$ [45]	1 M KOH	220.1
$\text{Fe}_2\text{O}_3$ -5wt% CNT [46]	2 M KCl	54
$\text{Fe}_3\text{O}_4/\text{carbon nanosheets}$ [47]	1 M $\text{Na}_2\text{SO}_4$	163.4
$\text{Fe}_3\text{O}_4/\text{CNF}$ [48]	1 M $\text{Na}_2\text{SO}_4$	127
$\text{Fe}_3\text{O}_4/\text{rGO}$ [43]	1 M KOH	241
$\text{Fe}_3\text{O}_4/\text{carbon nanofiber}$ [49]	3M KOH	225
$\text{Fe}_3\text{O}_4@\text{NG}$ (This work)	1M $\text{Na}_2\text{SO}_4$	244.3



To evaluate the electrochemical performance of Fe<sub>3</sub>O<sub>4</sub>@NG, the specific capacitance measured in this work was compared with that of reported in several previous studies of Fe-based electrodes. As seen in Table 1, the Fe<sub>3</sub>O<sub>4</sub>@NG electrode showed a competitive electrochemical performance, and this means that the Fe<sub>3</sub>O<sub>4</sub>@NG is an ideal electrode material.

#### 4. CONCLUSIONS

In summary, the Fe<sub>3</sub>O<sub>4</sub>@NG composite material that was rich in pyrrolic N was successfully synthesized via a hydrothermal and calcination route. Fe<sub>3</sub>O<sub>4</sub> nanoparticles that had a large specific surface area were uniformly embedded in NG sheets. Benefiting from the pyrrolic-N-rich structure of NG and the synergetic effect between NG and Fe<sub>3</sub>O<sub>4</sub>, the Fe<sub>3</sub>O<sub>4</sub>@NG composite electrode showed a high specific capacity of 244.3F/g at a current density of 1 A/g. Furthermore, the specific capacitance still remained 85 % of the initial capacity after 1000 cycles. Therefore, we have determined that the Fe<sub>3</sub>O<sub>4</sub>@NG composite has good electrochemical performance and is an excellent material for high-performance supercapacitors.

#### ACKNOWLEDGEMENT

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