

## Effects of nanosized Bi<sub>2</sub>O<sub>3</sub> addition on the superconducting properties of Bi<sub>1.6</sub>Pb<sub>0.4</sub>Sr<sub>2</sub>Ca<sub>2</sub>Cu<sub>3</sub>O<sub>10</sub>

Nabil A. A. Yahya<sup>1</sup>, R. Abd-Shukor<sup>2,\*</sup>

<sup>1</sup> Department of Physics, Tamar University, Tamar, Yemen

<sup>2</sup> School of Applied Physics, Universiti Kebangsaan Malaysia, 43600 Bangi, Selangor Malaysia

\*E-mail: [ras@ukm.edu.my](mailto:ras@ukm.edu.my)

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The effects of nanosized Bi<sub>2</sub>O<sub>3</sub> (150 nm) addition on Bi<sub>1.6</sub>Pb<sub>0.4</sub>Sr<sub>2</sub>Ca<sub>2</sub>Cu<sub>3</sub>O<sub>10</sub> ((Bi,Pb)-2223) superconductor have been investigated. Samples with nominal starting composition (Bi,Pb)-2223(Bi<sub>2</sub>O<sub>3</sub>)<sub>x</sub> with  $x = 0-0.15$  wt.% were prepared using the co-precipitation method. The structure and microstructure were examined by using powder X-ray diffraction (XRD) method and scanning electron microscopy (SEM), respectively. The onset-temperature ( $T_{c-onset}$ ), zero-resistance-temperature ( $T_{c-zero}$ ), and transport critical current density ( $J_c$ ) were determined by using the four-probe technique. The XRD patterns confirmed the presence of Bi,Pb-2223 phase.  $J_c$  of all Bi<sub>2</sub>O<sub>3</sub> added samples were higher than the non-added sample. The highest  $J_c$ ,  $T_{c-zero}$  and hole concentration ( $p$ ) was observed in the  $x = 0.01$  sample. At 77 K,  $J_c$  of  $x = 0.01$  wt. % was about 42 times higher than the non-added sample. This could be due to the enhanced flux pinning as the size of Bi<sub>2</sub>O<sub>3</sub> was between the coherence length and penetration depth of (Bi,Pb)-2223.

**Keywords:** nanosized Bi<sub>2</sub>O<sub>3</sub>; flux pinning centers; transport critical current density

### 1. INTRODUCTION

The Bi<sub>1.6</sub>Pb<sub>0.4</sub>Sr<sub>2</sub>Ca<sub>2</sub>Cu<sub>3</sub>O<sub>10</sub> ((Bi,Pb)-2223) high- $T_c$  superconductor is one of the most promising materials for applications as tapes or wires. However, the transport critical current density ( $J_c$ ) is strongly suppressed by weak links and weak pinning of magnetic flux lines [1-3]. The coherence length ( $\xi$ ), penetration depth ( $\lambda$ ) and magnetic flux size in a superconductor is in the nanometer range. The weak pinning of magnetic flux lines decreases  $J_c$  with increasing temperature and magnetic field, due to the motion of the vortices.

The penetration depth,  $\lambda$  of (Bi,Pb)-2223 superconductor is about 1000 nm and the coherence length  $\xi$  is 2.9 nm. It is expected that the interaction between the pinning center (e.g. nanoparticles) and magnetic flux lines will be strong for a particle with size  $d$  where  $\xi < d < \lambda$  [4]. The introduction of

nanosized particles as defects into (Bi,Pb)-2223 can be an effective method to improve flux pinning without destroying the superconductivity and thus enhance  $J_c$  [e.g. 5,6]. The size and type of nanoparticles as the magnetic flux pinning center in  $\text{YBa}_2\text{Cu}_3\text{O}_7$  and (Bi,Pb)-2223 are important parameters [7-13]. In a previous study, we have investigated the effect of nanosized PbO (10 – 30 nm) on  $J_c$  of  $\text{Bi}_{1.6}\text{Pb}_{0.4}\text{Sr}_2\text{Ca}_2\text{Cu}_3\text{O}_{10}$  tapes [5].

Bismuth oxide has been useful in enhancing the superconducting properties of the cuprates such as the bismuth- and thallium-based high temperature superconductors. When  $\text{Bi}_2\text{O}_3$  (50-80 nm) was added to (Bi,Pb)-2223, the samples with composition (Bi,Pb)-2223( $\text{Bi}_2\text{O}_3$ )<sub>0.06</sub> showed the highest  $J_c$  [14]. It would be interesting to investigate the effects of  $\text{Bi}_2\text{O}_3$  with other size on the (Bi,Pb)-2223 phase. In this work,  $\text{Bi}_2\text{O}_3$  with size 150 nm was added into (Bi,Pb)-2223. This size was chosen because it is between  $\xi$  and  $\lambda$ . The onset temperature  $T_{c\text{-onset}}$ , zero resistance temperature  $T_{c\text{-zero}}$  and  $J_c$  of (Bi,Pb)-2223( $\text{Bi}_2\text{O}_3$ )<sub>x</sub> for  $x = 0$  to 0.15 wt. % were measured. The structural and microstructural properties of (Bi, Pb)-2223 were also investigated.

## 2. EXPERIMENTAL DETAILS

The pellets were prepared from high purity powders (> 99%) of  $\text{Bi}(\text{CH}_3\text{CO}_2)_3$ ,  $\text{Pb}(\text{CH}_3\text{CO}_2)_3 \cdot 3\text{H}_2\text{O}$ ,  $\text{Sr}(\text{CH}_3\text{CO}_2)_2 \cdot 1/2\text{H}_2\text{O}$ ,  $\text{Ca}(\text{CH}_3\text{CO}_2)_2 \cdot \text{H}_2\text{O}$ , and  $\text{Cu}(\text{CH}_3\text{CO}_2)_2 \cdot \text{H}_2\text{O}$  with nominal starting composition  $\text{Bi}_{1.6}\text{Pb}_{0.4}\text{Sr}_2\text{Ca}_2\text{Cu}_3\text{O}_{10}$ . The powders were prepared by the acetate co-precipitation technique, where the filtered precipitate was calcined in a tube furnace for 12 h at 730 °C to remove the volatile materials. An additional calcination was performed for 24 h at 845 °C to start the formation of (Bi,Pb)-2223 superconducting phase. The size of the precursor powder was a few micrometers.  $\text{Bi}_2\text{O}_3$  with size 150 nm (US-nano, 99+% purity) was added into (Bi,Pb)-2223 with different concentrations ( $x = 0, 0.01, 0.05, 0.10$  and 0.15 wt.%). The mixed powders were ground and then pressed into pellets of ~ 12 mm diameter and ~ 2 mm thickness and sintered for 48 h at 845 °C. The heating and cooling rate was 2 °C/min.

The phase and structure of the samples were examined by a D8 Advance X-ray diffractometer (XRD) from Bruker AXS with a  $\text{CuK}\alpha$  source ( $\lambda = 0.15406$  nm). The volume fraction of Bi-2223 (high- $T_c$  phase) and Bi-2212 (low- $T_c$  phase) was estimated from the total intensities of these phases using the following equations [15,16]:

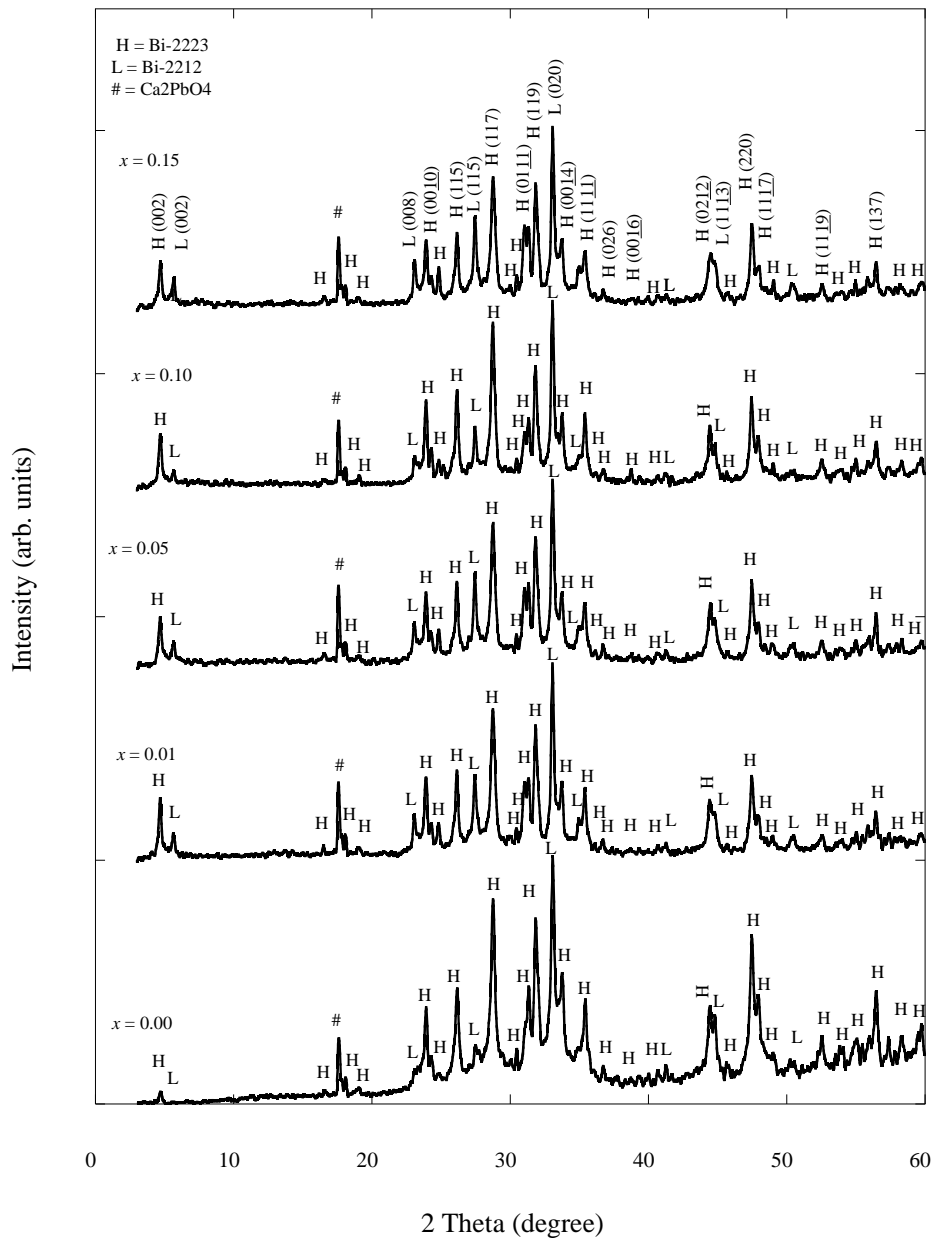
$$\text{Bi-2223 \%} = \frac{\sum I_{2223}}{\sum I_{2223} + \sum I_{2212}} \times 100\%$$

$$\text{Bi-2212 \%} = \frac{\sum I_{2212}}{\sum I_{2223} + \sum I_{2212}} \times 100\%$$

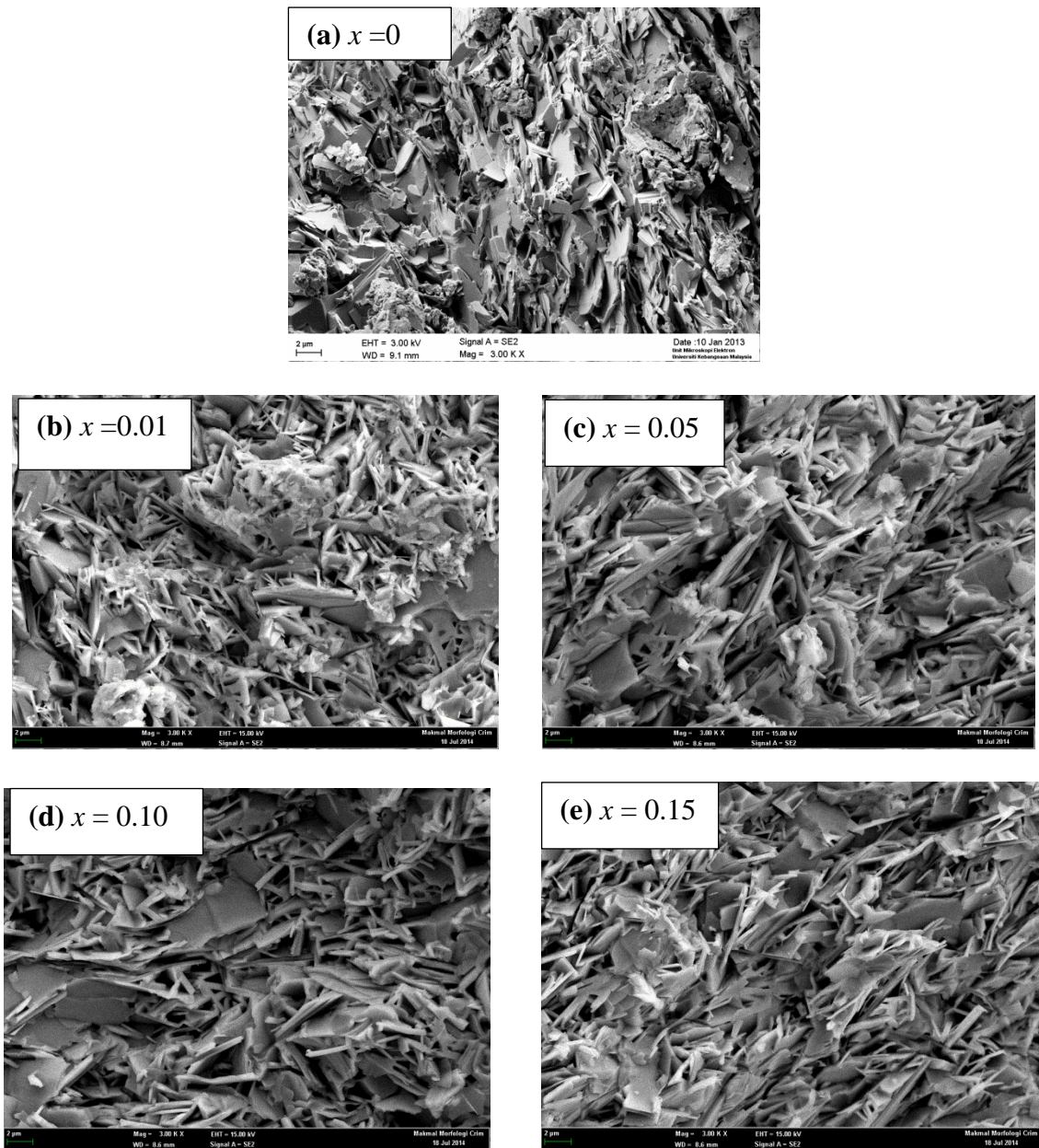
The microstructure of the samples was recorded using a Zeis VPSEM (Leo 1450). The size of  $\text{Bi}_2\text{O}_3$  was determined by transmission electron microscope (HRTEM, JEOL JEM- 2100F). The four-probe technique was used to determine  $T_{c\text{-onset}}$  and  $T_{c\text{-zero}}$  of the pellets. The effect of nanosized  $\text{Bi}_2\text{O}_3$

on  $J_c$  was determined by using four-probe technique from 30 to 77 K in self-fields. The 1- $\mu$ V/cm criterion was used to determine  $J_c$ .

### 3. RESULTS AND DISCUSSION

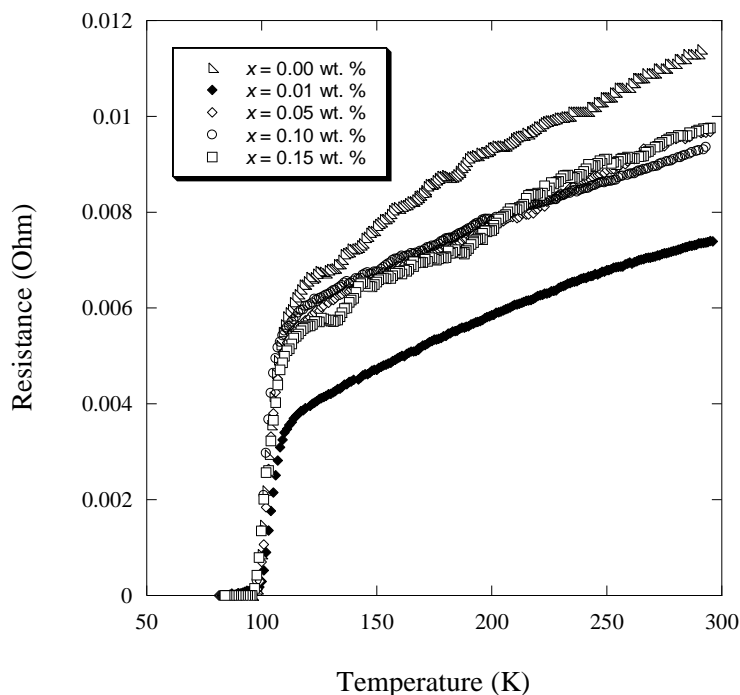


**Figure 1.** XRD patterns of  $(\text{Bi,Pb})\text{-}2223(\text{Bi}_2\text{O}_3)_x$  for  $x = 0, 0.01, 0.05, 0.10,$  and  $0.15$  wt. %. (*H*) indicates the high- $T_c$  phase and (*L*) indicates the low- $T_c$  phase



**Figure 2.** SEM micrographs of (Bi,Pb)-2223(Bi<sub>2</sub>O<sub>3</sub>)<sub>x</sub>; (a)  $x = 0$  wt. %, (b)  $x = 0.01$  wt. %, (c)  $x = 0.05$  wt. %, (d)  $x = 0.10$  wt. %, and (e)  $x = 0.15$  wt. %

Figure 1 shows the XRD patterns of (Bi,Pb)-2223(Bi<sub>2</sub>O<sub>3</sub>)<sub>x</sub> for  $x = 0, 0.01, 0.05, 0.01,$  and  $0.15$  wt.%. The XRD patterns showed that a small amount of nanosized Bi<sub>2</sub>O<sub>3</sub> did not hinder the formation of (Bi, Pb)-2223 phase. Most of the peaks corresponded mainly to the Bi-2223 (high- $T_c$  phase). Minor peaks due to the Bi-2212 (low- $T_c$ ) phase was also present in the samples. In addition, a small peak signifying the Ca<sub>2</sub>PbO<sub>4</sub> phase was observed at  $2\theta \approx 17.6^\circ$  in all samples. The lattice parameters  $a$ ,  $b$ , and  $c$  of the non-added sample are 5.415, 5.407, and 37.12 Å, respectively. The lattice parameters in Bi<sub>2</sub>O<sub>3</sub> added samples were almost the same as those of the non-added sample. It is therefore likely that a small amount of nanosized Bi<sub>2</sub>O<sub>3</sub> did not affect the (Bi,Pb)-2223 crystal structure. SEM micrographs for (Bi, Pb)-2223(Bi<sub>2</sub>O<sub>3</sub>)<sub>x</sub> ( $x = 0, 0.01, 0.05, 0.01,$  and  $0.15$  wt.%) are shown in Figures 2(a-e). SEM of all the samples showed plate-like grain structure of the high- $T_c$  phase (Bi-2223).



**Figure 3.** Temperature dependence of electrical resistance of (Bi, Pb)-2223(Bi<sub>2</sub>O<sub>3</sub>)<sub>x</sub> for  $x = 0, 0.01, 0.05, 0.10,$  and  $0.15$  wt. %

The temperature dependence of electrical resistance for all samples exhibited metallic normal state behavior (Figure 3).  $T_{c\text{-onset}}$  and  $T_{c\text{-zero}}$  for the non-added sample is 110 K and 98 K, respectively. The sample with  $x = 0.01$  wt. % showed the highest  $T_{c\text{-zero}}$  (99 K) among the samples and the same  $T_{c\text{-onset}}$  compared with non-added sample (Table 1). It is probable that very small addition of nanosized Bi<sub>2</sub>O<sub>3</sub> ( $x = 0.01$  wt. %) slightly improved the intergranular links[17].

**Table 1.** Lattice parameters, volume fraction,  $T_{c\text{-onset}}$ ,  $T_{c\text{-zero}}$ ,  $J_c$ , (at 30 and 77 K) and hole concentration ( $p$ ) of (Bi,Pb)-2223(Bi<sub>2</sub>O<sub>3</sub>)<sub>x</sub> for  $x = 0 - 0.15$  wt. %

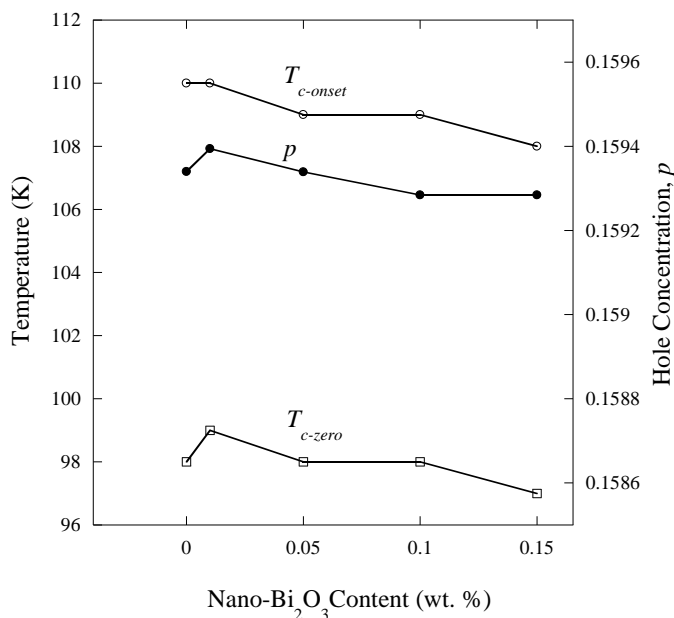
$x$	$a / \text{\AA}$	$b / \text{\AA}$	$c / \text{\AA}$	BiPb-2223 / %	BiPb-2212 / %	$T_{c\text{-onset}}$ / K	$T_{c\text{-zero}}$ / K	$p$ hole conc.	$J_c$ (30 K) / mA cm <sup>-2</sup>	$J_c$ (77 K) / mA cm <sup>-2</sup>
0.00	5.415	5.407	37.12	74	26	110	98	0.15934	344±10	59±2
0.01	5.403	5.417	37.11	72	28	110	99	0.15939	4620±90	2500±50
0.05	5.416	5.409	37.11	71	29	109	98	0.15934	2310±70	1640±50
0.10	5.414	5.409	37.12	74	26	108	97	0.15928	2250±40	1020±20
0.15	5.413	5.411	37.10	71	29	108	97	0.15928	1840±60	690±20

The charge carriers,  $p$  (hole concentration) in the samples can be calculated according to the following equation [16, 18]:

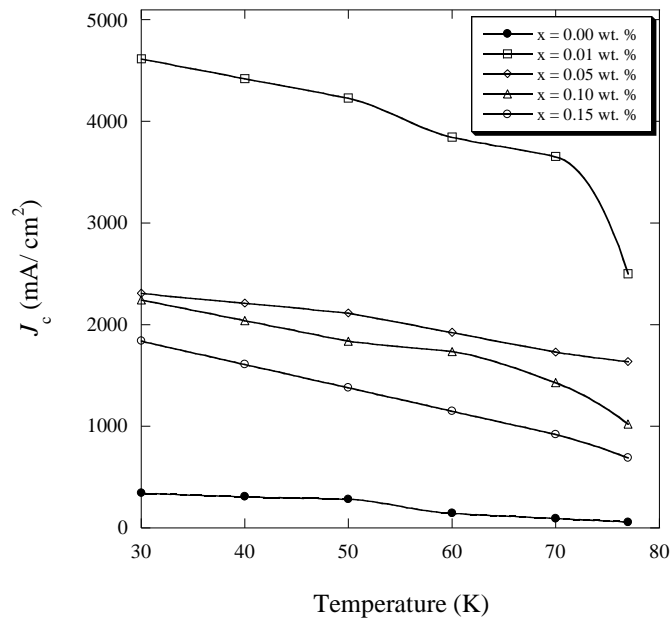
$$p = 0.16 - \left( \frac{\left( \frac{1 - T_c}{T_c^{\max}} \right)}{82.6} \right)^{1/2}$$

where  $T_c^{\max}$  for (Bi, Pb)-2223 superconductor was taken at 110 K.

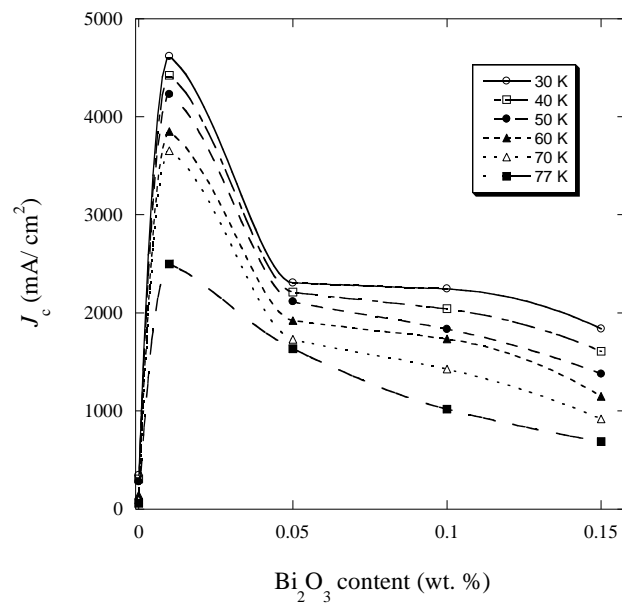
The Bi<sub>2</sub>O<sub>3</sub> content dependence of hole concentration,  $p$ ,  $T_{c-onset}$  and  $T_{c-zero}$  are shown in Figure 4. The hole concentration,  $p$  for  $x = 0.01$  wt. % sample showed the highest value. The  $x = 0.05$  wt. % sample showed similar hole concentration,  $p$  compared to the non-added sample. The hole concentration,  $T_{c-onset}$  and  $T_{c-zero}$  for  $x > 0.05$  wt. % samples were slightly lower than those of the non-added sample. It is clear that  $T_{c-onset}$  and  $T_{c-zero}$  with  $x > 0.05$  wt. % decreased slightly due to the decrease of the hole concentration,  $p$ .



**Figure 4.** Nanosized Bi<sub>2</sub>O<sub>3</sub> content dependence of hole concentration ( $p$ ),  $T_{c-onset}$  and  $T_{c-zero}$  for  $x = 0, 0.01, 0.05, 0.10,$  and  $0.15$  wt. %



**Figure 5.**  $J_c$  of (Bi,Pb)-2223( $\text{Bi}_2\text{O}_3$ ) $_x$  as a function of temperature for  $x = 0, 0.01, 0.05, 0.10,$  and  $0.15$  wt. %



**Figure 6.**  $J_c$  of (Bi,Pb)-2223( $\text{Bi}_2\text{O}_3$ ) $_x$  as a function of different adding concentrations for  $x = 0, 0.01, 0.05, 0.10,$  and  $0.15$  wt. %

$J_c$  for all samples as a function of temperature is shown in Figure 5. In general,  $J_c$  decreased with the increased of temperature from 30 to 77 K due to the thermally activated flux creep. All of the  $\text{Bi}_2\text{O}_3$  added samples showed much higher  $J_c$  compared with the non-added sample. For the non-added sample,  $J_c$  at 30 and 77 K was 344 mA/cm<sup>2</sup> and 59 mA/cm<sup>2</sup>, respectively. The  $x = 0.01$  wt. % sample showed the highest  $J_c$ , which was 4615 mA/cm<sup>2</sup> and 2500 mA/cm<sup>2</sup> at 30 and 77 K, respectively. The improved  $J_c$  may be due to the enhancement of flux pinning strength as a result of nanosized  $\text{Bi}_2\text{O}_3$  at the grain boundaries.

$J_c$  showed a sudden decrease when the amount of  $\text{Bi}_2\text{O}_3$  was increased to more than  $x = 0.01$  wt. % (Figure 6). The lattice parameters, hole concentration,  $p$ ,  $T_{c\text{-onset}}$ ,  $T_{c\text{-zero}}$ , and  $J_c$  at 30 and 77 K of the samples are summarized in Table 1. It is interesting to note that nano PbO addition also increased  $J_c$  of the BiPb-2223 superconductor [5]. Hence, although these two elements are already in the parent compound, addition of nanosized  $\text{Bi}_2\text{O}_3$  and PbO led to the enhancement of  $J_c$  without suppressing the transition temperature. The enhancement of  $J_c$  is possibly due to enhanced grain connectivity as a result of nanosized  $\text{Bi}_2\text{O}_3$  at the grain boundaries. Smaller  $\text{Bi}_2\text{O}_3$  (50-80 nm) addition showed  $x = 0.06$  wt. % as the highest  $J_c$  value [14]. In this work, 150 nm  $\text{Bi}_2\text{O}_3$  was used and a smaller amount ( $x = 0.01$  wt. %) was sufficient to optimize  $J_c$ .

In conclusion, the effects of nanosized  $\text{Bi}_2\text{O}_3$  addition on the superconducting properties of bulk  $\text{Bi}_{1.6}\text{Pb}_{0.4}\text{Sr}_2\text{Ca}_2\text{Cu}_3\text{O}_{10}(\text{Bi}_2\text{O}_3)_x$  samples were investigated. The optimal amount which showed the highest  $J_c$ ,  $T_{c\text{-zero}}$ , and hole concentration,  $p$  was  $x = 0.01$  wt. %.  $J_c$  of  $x = 0.01$  wt. % was about 13 and 42 times larger than the non-added sample at 30 and 77 K, respectively. The enhancement of  $J_c$  may be due to the fact that the particles size of  $\text{Bi}_2\text{O}_3$  is larger than  $\xi$  and smaller than  $\lambda$  of (Bi, Pb)-2223, which can increase the flux pinning ability in the samples.

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