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Short Communication

Synthesis and Electrochemical Properties of Intermediate Temperature $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}$ -molten Carbonate Composite Electrolyte

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An erbium-doped SrCeO₃/SrZrO₃ electrolyte was synthesized via a sol-gel method using zirconium nitrate, cerium nitrate, strontium nitrate, nitric acid, citric acid and erbium oxide as raw materials. And SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α}-molten carbonate composite electrolyte was obtained at a low synthesis temperature. The thermal gravimetry analysis-differential scanning calorimetry (TGA-DSC) was tested to determine the formation temperature of SrCeO₃/SrZrO₃ solid solution phase. The structure (X–ray diffractometer, XRD), morphology (scanning electron microscope, SEM), conductivities and H₂/O₂ fuel cells of SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α} and SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α}-Na₂CO₃-Li₂CO₃ were studied at 400-600 °C. The highest power densities of SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α} and SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α}-Na₂CO₃-Li₂CO₃ were 16.3 mW·cm⁻² at 600 °C, respectively.

Keywords: Composite; SrCeO₃; Electrolyte; Conductivity; Fuel cell

1. INTRODUCTION

Inorganic SrCeO₃-based solid electrolytes have excellent protonic conduction in watersaturated hydrogen atmospheres at 600-1000 °C. They are extensively used in fuel cells, solid sensors, organic reactors and hydrogen pumps etc. [1–10]. Normally, Ce⁴⁺ was doped with the trivalent elements such as Y^{3+} , Eu³⁺, Yb³⁺ and Tm³⁺ can improve the conductivities of SrCeO₃ [11–16]. Tsuji et al. synthesized SrCe_{0.9}Eu_{0.1}O_{3- $\alpha}$} by using the solid-state reaction after calcining at 1550 °C for 10 h, respectively [11]. Li et al. studied an SrCe_{0.9}Eu_{0.1}O_{3- α} protonic solid electrolytes to assemble a organic reactor for H₂ production [12]. However, the chemical stabilities of SrCeO₃-based solid electrolytes are poor. In contrast, SrZrO₃-based solid electrolytes are stable. Therefore, SrCeO₃/SrZrO₃ solid solutions which have high chemical stabilities and conductivities were investigated [13–16]. For example, Bredesen et al. [13] studied thulium or ytterbium doped SrCeO₃/SrZrO₃ solid solution which applied to hydrogen permeability. The electro-negativity and the ionic radii of Er^{3+} are 1.24 and 0.89 Å, which are much closer in electro-negativity and ionic radius to Y³⁺ (1.22, 0.90 Å), Yb³⁺ (1.3, 0.87 Å) and Tm³⁺ (1.25, 0.88 Å). In our opinion, if these trivalent elements have almost the same electro-negativity and ionic radius, it is logical to assume that the Er^{3+} -doped SrCeO₃ should also exhibit high proton conductivity. Therefore, erbium-doped SrCeO₃/SrZrO₃ is the object in the study.

Over the past decade, composite electrolytes such as BaCeO₃, LaGaO₃, CeO₂-based solid ceramic-carbonates, have been extensively investigated for intermediate temperature (400–600 °C) fuel cells [17–23]. Wang et al. synthesized a La_{0.9}Sr_{0.1}Ga_{0.8}Mg_{0.2}O_{2.85}-(Li/Na)₂CO₃ and it had a stable conductivity of $7-9\times10^{-2}$ S·cm⁻¹ at 600 °C [20]. Marques et al. found that the conductivities of Ce_{0.5}Yb_{0.5}O_{1.75}-Li₂CO₃-Na₂CO₃ composite electrolyte exceeded 0.1 S·cm⁻¹ above 500 °C [22]. Our previous studies have shown that SrCeO₃-NaCl-KCl composite electrolytes had excellent intermediate temperature electrochemical properties [24–25]. However, there are few studies about SrCeO₃/SrZrO₃ solid solution-carbonate composite electrolytes.

In this study, an erbium-doped $SrCeO_3/SrZrO_3$ electrolyte was synthesized via a sol-gel method. And $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}$ -molten carbonate composite electrolyte was obtained at a low temperature. The structure (XRD), morphology (SEM), conductivities and H_2/O_2 fuel cells of $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}$ and $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}$ -Na₂CO₃-Li₂CO₃ were studied at 400-600 °C.

2. EXPERIMENTAL

 $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}$ was synthesized via a sol-gel method using zirconium nitrate, cerium nitrate, strontium nitrate, nitric acid, citric acid and erbium oxide as raw materials. Firstly, erbium oxide was dissolved in nitric acid and zirconium nitrate, cerium nitrate and strontium nitrate were mixed in distilled water. Then, the *p*H was adjusted to 8-9 with an ammonia solution using citric acid as complexing agent. Finally, $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}$ was prepared after being calcined at 1200 °C and 1540 °C for 5 h, respectively.

The molten carbonate salt (mole ratio of $Li_2CO_3:Na_2CO_3 = 1:1$) was obtained after being heated at 580 °C [26]. The mixtures (weight ratio of the molten carbonate salt and $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha} = 1:4$) were calcined at 620 °C for 1 h to synthesize $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}-Na_2CO_3-Li_2CO_3$.

TGA-DSC was tested using the SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- $\alpha}$} precursor in nitrogen from 25 °C to 1000 °C with a heating rate of 15 °C·min⁻¹. The electrolyte structures and morphologies of SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- $\alpha}$} and SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α}-Na₂CO₃-Li₂CO₃ were measured using XRD and SEM. For conductivities measurements, SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α}-Na₂CO₃-Li₂CO₃ and SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α} were processed into thin wafers (electrode area = 0.50 cm², thickness = 1.0-1.1 mm). The resistance values were obtained using an electrochemical analyzer (CHI660E, made in China) in the frequency range of 1 Hz–1 MHz from 400 °C to 600 °C in air. H₂/O₂ fuel cells were also tested using SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α} as electrolyte at 600 °C.

3. RESULTS AND DISCUSSION

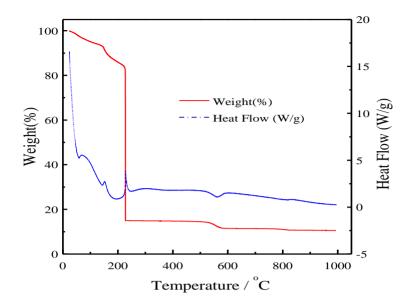


Figure 1. TGA-DSC plot of the SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-α} precursor.

The TGA-DSC curve of the SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α} precursor is shown in Fig. 1. The plot shows a 16.5% weight loss from 25 °C to 225 °C which is attributed to the evaporation of water and ammonia from the dry gel. There was a severe weight loss accompanied by an exothermic peak in a very narrow temperature range from 225 to 230 °C which is ascribed to the decomposition of ammonium salt and citric acid. A gentle declining weight loss was found from 530 to 580 °C which is attributed to the incomplete nitrate [27–28]. TGA and DSC plots are almost horizontal after 980 °C which means that the temperature is close to the formation of SrCeO₃/SrZrO₃ solid solution phase. Therefore, the first calcined temperature is 1200 °C.

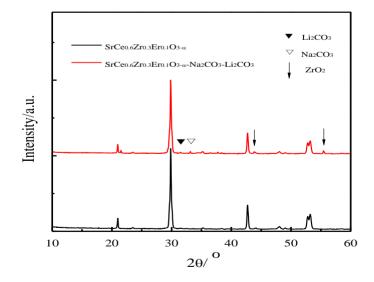


Figure 2. XRD patterns of $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}$ and $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}-Na_2CO_3-Li_2CO_3$.

The XRD patterns of $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}$ and $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}-Na_2CO_3-Li_2CO_3$ are shown in Fig. 2. The diffraction peaks confirm that $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}$ is consistent with the standard $SrCeO_3$ crystal plane. There are diffraction peaks of Li_2CO_3, Na_2CO_3 and ZrO_2 besides $SrCeO_3$ phase in the $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}-Na_2CO_3-Li_2CO_3$ pattern.

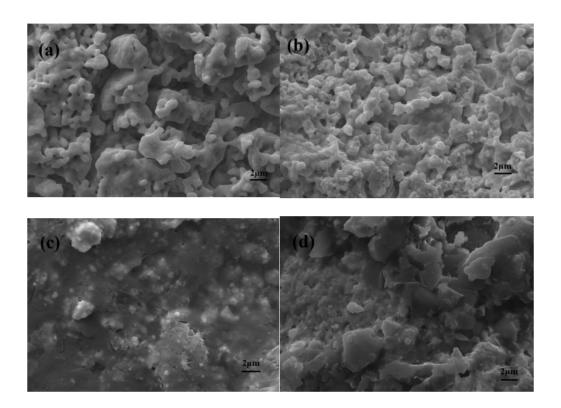


Figure 3. (a,b) SEM photos of $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}$ pellet calcined at 1540 °C for 5 h; (c,d) external and cross-sectional surfaces of $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}-Na_2CO_3-Li_2CO_3$.

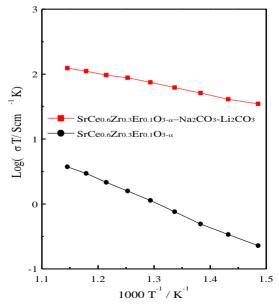


Figure 4. The log (σ T) ~ 1000 T⁻¹ plots of SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α} and SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α}-Na₂CO₃-Li₂CO₃ in air from 400 °C to 600 °C.

The result illustrates that Na₂CO₃-Li₂CO₃ melts entry into SrCeO₃/SrZrO₃ solid solution to reduce the limitation of ZrO₂. Marques et al. revealed that the introduction of Li₂CO₃-Na₂CO₃ promoted a slight decomposition of Ce_{0.5}Yb_{0.5}O_{1.75} [22]. And combined with previous studies [22, 23, 29], Na₂CO₃-Li₂CO₃ coexists in two forms of crystalline and amorphous phases in SrCe_{0.6}Zr_{0.3}Er_{0.1}O₃- α -Na₂CO₃-Li₂CO₃.

The SEM photos of $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}$ (a,b) and $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}$ -Na₂CO₃-Li₂CO₃ (c,d) are shown in Fig. 3. From Fig. 3(a,b), the sintered $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}$ has clear agglomeration with a grain size is 2-6 µm. From Fig. 3(c,d), there is an interlacing composition between $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}$ and Li_2CO_3 -Na₂CO₃. It can be observed that the $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}$ particles are nestled inside molten carbonate salt which makes the composite electrolyte into a continuous 3-D reticulated structure [17].

Fig. 4 shows the log (σ T) ~ 1000 T⁻¹ plots of SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- $\alpha}$ and SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α}-Na₂CO₃-Li₂CO₃ in air from 400 °C to 600 °C. From Fig. 4, the slope of SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α}-Na₂CO₃-Li₂CO₃ is lower than that of SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- $\alpha}$ which means the conduction ion migration ability of the former is strong. The activation energies of SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- $\alpha}$ and SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α}-Na₂CO₃-Li₂CO₃ are 31.6 ± 0.7 kJ·mol⁻¹ and 69.8 ± 0.8 kJ·mol⁻¹, respectively. The conductivities of SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- $\alpha}$ and SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- $\alpha}$ and SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- $\alpha}$ -Na₂CO₃-Li₂CO₃ vary from 3.4×10⁻⁴ S·cm⁻¹ to 4.1×10⁻³ S·cm⁻¹ and 5.2×10⁻² S·cm⁻¹ to 1.4×10⁻¹ S·cm⁻¹ at 400–600 °C, correspondingly. The lower activation energy and higher conductivities of SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α}-Na₂CO₃-Li₂CO₃ indicate that the molten carbonate salt provide an additional transport function between the interfaces [17,23,30].}}}}}}

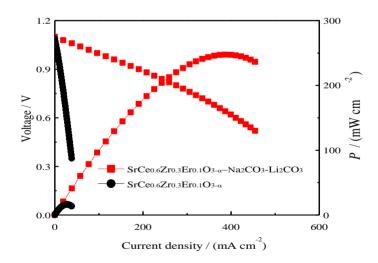


Figure 5. The *I-V-P* curves of $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}$ and $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}-Na_2CO_3-Li_2CO_3$ at 600 °C.

Table 1. The highest power densities of $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}$ -Na₂CO₃-Li₂CO₃ and similar electrolytes in the literatures.

Electrolytes	Highest power densities
SrCe _{0.6} Zr _{0.3} Er _{0.1} O _{3-α} -Na ₂ CO ₃ -Li ₂ CO ₃ (80: 20)	247.5 mW·cm ⁻² , 600 °C, in this work
Ce _{0.8} Sm _{0.2} O _{1.9} - (Li/Na) ₂ CO ₃ (80: 20)	605 mW·cm ⁻² , 575 °C, [31]
Ce _{0.8} Sm _{0.2} O _{1.9} - (Li/Na) ₂ CO ₃ (60: 40)	240 mW·cm ⁻² , 575 °C, [31]
$Ce_{0.9}Gd_{0.1}O_{1.95}$ - LiCl-SrCl ₂ (12: 5)	245 mW·cm ⁻² , 550 °C, 320 mW·cm ⁻² , 590 °C, [18]

Fig. 5 shows the *I-V-P* curves of $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}$ and $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}-Na_2CO_3-Li_2CO_3$ at 600 °C. The highest power density obtained with using $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}-Na_2CO_3-Li_2CO_3$ as the electrolyte is 247.5 mW·cm⁻², which was estimated to be about 15 times higher than that of the $SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3-\alpha}$ with 16.3 mW·cm⁻². This may be attributed to the homogeneous 3-D reticulated structure of the composite electrolyte. The *P*_h value of our result is close to the fuel cell performance of 60 wt% Ce_{0.8}Sm_{0.2}O_{1.9}-40 wt% (Li/Na)₂CO₃ (575 °C) and Ce_{0.9}Gd_{0.1}O_{1.95}- LiCl-SrCl₂ (550 °C), however, lower than 80 wt% Ce_{0.8}Sm_{0.2}O_{1.9}-20 wt% (Li/Na)₂CO₃ (575 °C) and Ce_{0.9}Gd_{0.1}O_{1.95}- LiCl-SrCl₂ (550 °C), shown in Table 1 [18, 31]. This may be due to the different electrolyte types and fuel cell construction.

4. CONCLUSIONS

An erbium-doped SrCeO₃/SrZrO₃ electrolyte was synthesized via a sol-gel method and SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α}-molten carbonate composite electrolyte was obtained at a low synthesis temperature. The XRD patterns indicate that Na₂CO₃-Li₂CO₃ coexists in two forms of crystalline and amorphous phases in SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α}-Na₂CO₃-Li₂CO₃. The SEM photos show that the SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α} particles are nestled inside molten carbonate salt to form a continuous 3-D reticulated structure. The highest power density and conductivity of SrCe_{0.6}Zr_{0.3}Er_{0.1}O_{3- α}-Na₂CO₃-Li₂CO₃ are 247.5 mW·cm⁻² and 1.4×10⁻¹ S·cm⁻¹ at 600 °C, respectively.

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