

## Yolk-shell $\text{Co}_3\text{O}_4$ -CoO/Carbon Composites for Lithium-Ion Batteries with Enhanced Electrochemical Properties

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A novel yolk-shell structured  $\text{Co}_3\text{O}_4$ -CoO/Carbon ( $\text{Co}_3\text{O}_4$ -CoO/C) composite was prepared and applied as cathode active materials in lithium ion batteries. The composites were prepared through a facile copolymerization of ionic liquid and acrylonitrile, which were used as carbon sources, followed by carbonization. The yolk-shell structure of the composite endows the electrodes with excellent cycling performance due to improved electronic conductivity, reduced strain and depressed aggregation of active materials compared with bare  $\text{Co}_3\text{O}_4$ . Thus, the  $\text{Co}_3\text{O}_4$ -CoO/C composite showed a high reversible capacity and cycling life of  $590 \text{ mAh}\cdot\text{g}^{-1}$  after 50 charge/discharge cycles at a current rate of  $890 \text{ mAh}\cdot\text{g}^{-1}$ .

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**Keywords:**  $\text{Co}_3\text{O}_4$ -CoO/Carbon Composites; Yolk-shell structure; Lithium-ion batteries

### 1. INTRODUCTION

Lithium-ion batteries (LIBs) have been identified as one of the most important energy storage device because of their many beneficial properties such as low self-discharge rate, high voltage platform, environmental friendliness and portability [1-4]. However, as their applications expand from small-scale portable electronics to largescale electronic vehicles, graphite, the common used anode material with a limited theoretical capacity ( $372 \text{ mAh}\cdot\text{g}^{-1}$ ), can no longer meet the requirements for high density energy devices [5]. Therefore, plenty of researches have been taken to find a substitution

for conventional carbon materials as new anode materials to improve the electrical energy density for LIBs[6].

Transition metal oxides, such as  $\text{Fe}_3\text{O}_4$ ,  $\text{Fe}_2\text{O}_3$ ,  $\text{Co}_3\text{O}_4$ , have attract numerous attentions due to their high theoretical capacity which is several times higher than that of traditional electrodes [7-10]. The spinel crystal structured  $\text{Co}_3\text{O}_4$  in particularly possess high theoretical capacity ( $890 \text{ mAh}\cdot\text{g}^{-1}$ ), high electro-activity, natural abundance, environmental friendliness and easy processing, which makes it an efficient anode electrode material for LIBs[10-13]. Nevertheless, the wide application of the  $\text{Co}_3\text{O}_4$  in electrode is hindered by its low electron transfer ability and huge volume change during lithiation and dislithiation process. [14, 15].

To overcome this bottleneck, many researches have been carried out and can be divided into four groups: (1) Nanocrystallization of  $\text{Co}_3\text{O}_4$  particles, which introduces many void among  $\text{Co}_3\text{O}_4$  particles, can increase the contacting area between the active materials and electrolyteand, therefore, shorten the diffusion distance of ions and electrons[16]. In addition, the nanocrystallization can also moderate the stress existing in the  $\text{Co}_3\text{O}_4$  electrode during the lithiation and dislithiation process [17]. (2) Specialization morphologies of  $\text{Co}_3\text{O}_4$  nano-materials can prevent the aggregation of nanoparticles and maintain the cycling stability of its nanostructure[18, 19], such as flower-like morphologies[20], hollow microspheres[21], nanorods[18], nanotubes[22, 23], nanocages[10], nanowires[24], network-like morphologies[25] and etc.. (3) Combination of  $\text{Co}_3\text{O}_4$  with other materials can improve its electrical conductivity and ameliorate its volume change effectively during charge-discharge process[26, 27]. Carbon materials (such as graphite[28], carbon nanotubes[29-31], graphene[32, 33] and etc.) and metallic oxide materials (such as  $\text{Fe}_2\text{O}_3$ [27], NiO[34] and etc.) are the mainly used integrating materials. (4) Cladding carbon film on  $\text{Co}_3\text{O}_4$  particles to form mechanical protection layer, which can also promote to form uniform conductive network and stable solid electrolyte interface (SEI) during cycling process [35].

Among all these methods, combining  $\text{Co}_3\text{O}_4$  with carbon materials is the most promising method because it can not only facilitate the transmission of electron but also accommodate the volume changes[15, 36]. Jun Liu et al.[37] used glucose as carbon source to synthesized carbon-coated mono-dispersed  $\text{Co}_3\text{O}_4$  porous and hollow nanoparticles by using an *in-situ* carbon-coating method. The as-prepared nano-composites exhibited high reversible capacity ( $1150 \text{ mAh}\cdot\text{g}^{-1}$  for the initial 20 cycles at the current of 0.25C), stable cycle life, and high energy efficiency ( $1078 \text{ mAh}\cdot\text{g}^{-1}$  at the current of 0.5C,  $700 \text{ mAh}\cdot\text{g}^{-1}$  at the current of 2C). Baojun Li et al.[38] successfully prepared a  $\text{Co}_3\text{O}_4$ @graphene composites through a chemical deposition method, with  $\text{Co}_3\text{O}_4$  nanoparticles homogeneously dispersed onto graphene sheets, displaying a superior cyclic performance in LIBs.

Among all the precursors of carbon films, ionic liquids (ILs) have been the focus of an ever increasing amount of research due to its extraordinary properties, such as negligible vapor pressure and structure designability. The negligible vapor of ILs enable them simple processing during the carbonization process without evaporation. In addition, as the ILs are designable, it is feasible to synthesis carbon materials with controllable structure and a certain amount of heteroatoms by adjusting the cation or anion components of ILs. Due to these unique properties, the integrating carbon film prepared from ILs show superior electronic conductivity, and excellent bonding strength between carbon substrate and active materials[39, 40].

In this study,  $\text{Co}_3\text{O}_4\text{-CoO/C}$  composites have been successfully synthesized by the polymerization of the hydroxide radicals on the surface of cobalt oxide, using IL and acrylonitrile as monomers, followed by carbonization of the above copolymers. The as-obtained composites exhibit a novel yolk-shell structure and an enhanced electrochemical performance when served as electrode materials. The electrodes made of the composites possess a superior cycling capacity of  $580 \text{ mAh}\cdot\text{g}^{-1}$ . Due to the facile preparation method, favorable capacity and outstanding cycling performance, the yolk-shell structured  $\text{Co}_3\text{O}_4\text{-CoO/C}$  composite is an promising candidate for electrode material of lithium ion batteries.

## 2. EXPERIMENTAL SECTION

### 2.1 Materials synthesis

$\text{Co}_3\text{O}_4$  (99%) was purchased from Aladdin. 1-vinyl-3-methyl tetrafluoroborate was purchased from Lanzhou Institute of Chemical Physics. All the other reagents were analytical grade and used without further purify. In a representative synthesis, 1g  $\text{Co}_3\text{O}_4$  was added into the solution(84ml) of deionized water and absolute ethyl alcohol (1:1 by vol.%). Subsequently, 9 ml silane coupling agent (KH570) is dissolved dropwise into the above mixture with continuous stirring at ambient environment followed by ultrasonication for half an hour. The PH value of the mixture was then adjusted to 5 by acetic acid at room temperature. The mixture was mechanical stirred for an hour, heating in water bath at 30 °C. The modified  $\text{Co}_3\text{O}_4$  was separated by suction filtration and washed with deionized water for three times. The modified  $\text{Co}_3\text{O}_4$  (0.5g) was mixed with the solution (40ml) of deionized water and absolute ethyl alcohol (1:1 by vol.%). 0.6g acrylonitrile (AN) , 0.4g ionic liquid (IL),0.07g 2,2-azobisisobutyronitrile (AIBN) and 0.1g poly(ethylene glycol) diacrylate (PEGDA) were added into the above mixture in sequence followed by mechanical stirring for 20 hours at ambient conditions. The obtained sample was washed with deionized water for 3 times, and finally, calcined for 2 hours at the temperature of 500 °C with the protection of argon atmosphere.

### 2.2 Materials characterization

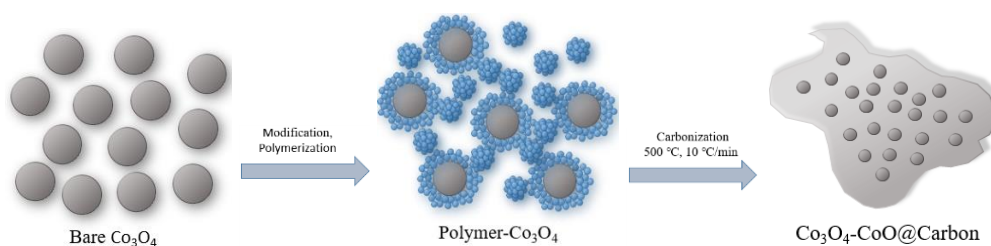
The phase structure of as-prepared samples were characterized with X-ray diffraction (XRD, D/max-2400, Cu K $\alpha$ ).The morphology was studied by transmission electron microscope (TEM, JEM-1200EX, 100kV) and scanning electron microscope (SEM, Quanta450FEG). The presence of organic groups on the sample surface was analyzed by Fourier transform infrared spectroscopy (FTIR, Nexus 670). Raman spectra was conducted with excitation from the 532 nm line of an Ar ion laser on LabRAM HR800.

### 2.3 Electrochemical measurements

The electrochemical properties of as-synthesized materials were evaluated by using CR2032 type coin cells assembled in argon-filled glove box. The working electrodes were fabricated by

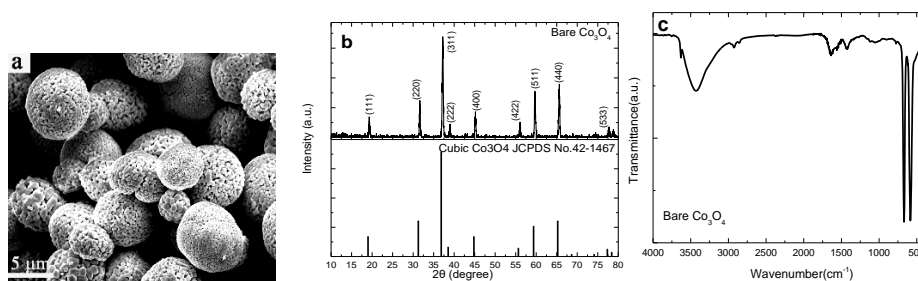
spreading the slurry on copper foils current collector, which was a mixture of 85 wt.% active materials ( $\text{Co}_3\text{O}_4\text{-CoO/C}$  or bare  $\text{Co}_3\text{O}_4$ ), 8 wt.% acetylene black as conductive material, 7 wt.% polyvinylidene fluoride (PVDF). After that, the working electrodes were dried in vacuum at 60 °C for 6 hours and pressed subsequently. Lithium foil was employed as the counter electrode, and Celgard 2300 served as a separator. The electrolyte was a solution of 1.0M  $\text{LiPF}_6$  in ethylene carbonate (EC) and 1,2-dimethoxyethane (DME) (1:1 by vol.%). The electrode capacity and cycling properties were measured by a galvanostatic discharge-charge method in the voltage range between 0.005 and 3.0V (vs.  $\text{Li/Li}^+$ ) at a current density of  $89 \text{ mAh}\cdot\text{g}^{-1}$  on a battery test system (LAND CT2001, China).

### 3. RESULTS AND DISCUSSION



**Figure 1.** Schematic of the synthesis process for  $\text{Co}_3\text{O}_4\text{-CoO/C}$  composites.

The preparation process of  $\text{Co}_3\text{O}_4\text{-CoO/C}$  composites was depicted in Fig.1. In order to obtain the modified  $\text{Co}_3\text{O}_4$ , KH570 and acetic acid were added into a water-ethanol suspension of bare  $\text{Co}_3\text{O}_4$  under uninterrupted stirring. To synthesis polymer- $\text{Co}_3\text{O}_4$  composites, the ionic liquid and acrylonitrile were mixed with the modified  $\text{Co}_3\text{O}_4$  and water-ethanol solution followed by adding the other reagents with the existence of water bath, and after that, the mixture was stirred for twenty hours. The as-obtained polymer- $\text{Co}_3\text{O}_4$  were washed for three times, and then calcined at the argon atmosphere to obtain the  $\text{Co}_3\text{O}_4\text{-CoO/C}$  composites promoted by carbonization and reduction actions.

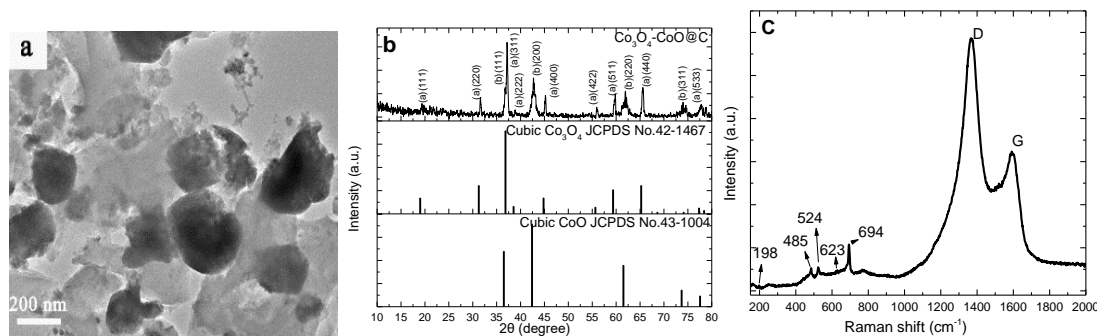


**Figure 2.** SEM image (a), X-ray diffraction patterns (b), and FTIR spectrum (c) of bare  $\text{Co}_3\text{O}_4$

The surface morphology of bare  $\text{Co}_3\text{O}_4$  micro-particles has been investigated by using Scanning Electron microscope (SEM), as seen in Fig.2a. The bare  $\text{Co}_3\text{O}_4$  are uniform spherical mono-

particles composed of nano-grains with numerous pores. Fig.2b shows the representative x-ray diffraction (XRD) patterns of spinel structured  $\text{Co}_3\text{O}_4$ . The diffraction reflections of bare  $\text{Co}_3\text{O}_4$  at  $2\theta=19.1, 31.6^\circ, 37.2^\circ, 38.8^\circ, 45.3^\circ, 56.1^\circ, 59.8^\circ, 65.6^\circ$  and  $77.8^\circ$  are attributed to (111), (220), (311), (222), (400), (422), (511), (440) and (533) planes, respectively, implying the well crystallization of the as-mentioned bare  $\text{Co}_3\text{O}_4$  with spinel structure. The strong and sharp peaks of spherical  $\text{Co}_3\text{O}_4$  proved its well formation at the same time.

FTIR spectrum of bare  $\text{Co}_3\text{O}_4$  (fig.2c) showed two characteristic absorption peaks at  $566\text{cm}^{-1}$  and  $671\text{cm}^{-1}$  verify the existence of spinel structured  $\text{Co}_3\text{O}_4$ [41]. The band at  $566\text{cm}^{-1}$  indicates that the  $\text{OB}_3$  composition of the spinel structure, where B refers to  $\text{Co}^{3+}$  in the octahedral interstice, and the band at  $671\text{cm}^{-1}$  implies the  $\text{ABO}_3$  composition of the spinel structure, where A denotes  $\text{Co}^{2+}$  in the tetrahedron interstice and B refers to  $\text{Co}^{3+}$  in the octahedral interstice[42, 43]. The absorption peaks at  $3430\text{ cm}^{-1}$  and  $1637\text{ cm}^{-1}$  are ascribed to  $-\text{OH}$  of water existed in samples or  $\text{KBr}$ [44]. The unsharp peak at  $3632\text{cm}^{-1}$  is attributed to the non-hydrogen bond  $\text{O}-\text{H}$ [45], which confirms the hydroxide radicals are on the cobalt oxide micro-particles.



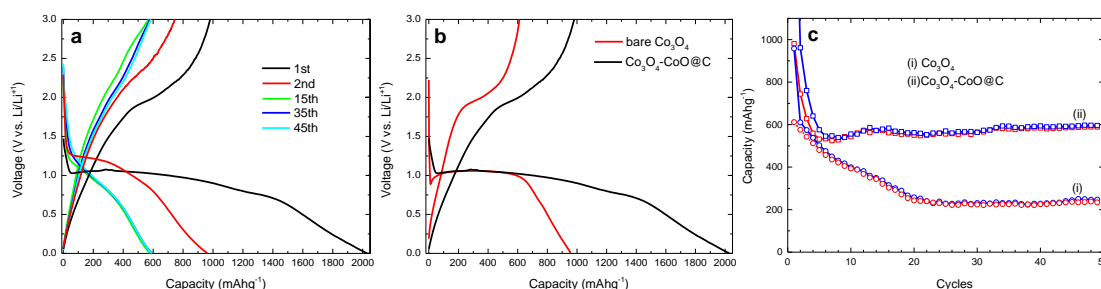
**Figure 3.** TEM image (a), X-Ray diffraction pattern (b), and Raman spectrum (c) of  $\text{Co}_3\text{O}_4\text{-CoO/C}$

Fig.3a displays the morphology and microstructure of as-obtained  $\text{Co}_3\text{O}_4\text{-CoO/C}$  composites by using TEM analysis. We can observe cobalt oxide particles seeding in the carbonous substrate uniformly. The interface between cobalt oxide and nitrogen doped carbon bonds well, which is beneficial for the transmission of electrons. The morphology of cobalt oxide particles, after annealing, have changed into irregular spherical particles by comparison with bare  $\text{Co}_3\text{O}_4$  (fig.2a). There are grains incorporated with each other on account of calcinating and agglomerating effects. It is distinctive that the particle size is smaller than bare  $\text{Co}_3\text{O}_4$ , from which it can be deduced that it is aroused by the emitting of oxygen atoms from  $\text{Co}_3\text{O}_4$  and the collapsing of the  $\text{Co}_3\text{O}_4$  particles.

Apparently, a crucial transformation of the as-synthesized composites has taken place compared with bare  $\text{Co}_3\text{O}_4$  (fig.2b) as revealed by XRD diffraction patterns (fig.3b), demonstrating that the initial bare  $\text{Co}_3\text{O}_4$  with spinel structure has partially changed into face-centered cubic structured  $\text{CoO}$  aroused by the reduction action of polyacrylonitrile (PAN) and poly ionic liquid (PIL). The diffraction peaks at  $2\theta=19.1^\circ, 31.6^\circ, 37.2^\circ, 38.8^\circ, 45.3^\circ, 56.1^\circ, 59.8^\circ, 65.6^\circ$  and  $77.8^\circ$  are corresponded to (111), (220), (311), (222), (400), (422), (511), (440) and (533) planes of spinel structured  $\text{Co}_3\text{O}_4$ , respectively. The diffraction peaks at  $2\theta=36.8, 42.6, 61.6$  and  $74.3$  are assigned to (111), (200),

(220) and (311) planes of face-centered cubic structured CoO, respectively. Meanwhile, it is worthy to take note of that the main phase is spinel structured  $\text{Co}_3\text{O}_4$ , mixed with relatively less CoO phase as designed.

The Raman spectra of the composites was shown in Fig.3c. The peaks at  $1365\text{ cm}^{-1}$  and  $1592\text{ cm}^{-1}$  are attributed to D band and G band respectively[46, 47]. The D band corresponds to the vibration of  $\text{sp}^3$ -hybridization carbon originating from the defects of the carbon substrates, and the G band is ascribed to the vibration of  $\text{sp}^2$ -hybridization carbon which is related with graphitic carbon[48, 49]. The intensity ratio of D and G bands ( $I_D/I_G$ ) is widely applied to characterize the graphitization degree of carbon materials or the carbon disorder degree[50, 51]. The  $I_D/I_G$  value was found to be 1.30, indicated that the carbon substrates have a high disorder degree. The peaks around  $198\text{ cm}^{-1}$ ,  $485\text{ cm}^{-1}$ ,  $524\text{ cm}^{-1}$ ,  $623\text{ cm}^{-1}$ ,  $694\text{ cm}^{-1}$  correspond to the  $B_{1g}$ ,  $E_g$ ,  $F_{1g}$ ,  $F_{2g}$  and  $A_{1g}$  modes of  $\text{Co}_3\text{O}_4$ , respectively[50, 52]. Whereas, due to the limited amount of CoO, its characteristic peak is barely observed[47].



**Figure 4.** Electrochemical performance. a) the first charge-discharge voltage profiles of bare  $\text{Co}_3\text{O}_4$  and  $\text{Co}_3\text{O}_4\text{-CoO/C}$  composites, respectively. b) the discharge-charge voltage profiles of  $\text{Co}_3\text{O}_4\text{-CoO/C}$  composites. c) cycling performance of bare  $\text{Co}_3\text{O}_4$  and  $\text{Co}_3\text{O}_4\text{-CoO/C}$  composites, respectively.

The discharge-charge voltage profiles, at a current density of  $89\text{ mAh}\cdot\text{g}^{-1}$ , of the electrodes made of as-obtained  $\text{Co}_3\text{O}_4\text{-CoO/C}$  composites are exhibited in fig.4a. The first long discharge voltage plateau, with a classic feature, at about  $1.05\text{V}$  can be witnessed, assigning to the formation of solid electrolyte interfaces (SEI) and the transformation of cobalt oxide to cobalt[53]. The first charge voltage plateau at about  $1.9\text{V}$  can be attributed to the oxidation reaction of cobalt to cobalt oxide[30]. When it comes to the second discharge process, the voltage plateau becomes more gradient, which is imputed to the heterogeneity of the active materials during lithiation and dislithiation[1]. Satisfactorily, the 35<sup>th</sup> and 45<sup>th</sup> are in line with each other, implying the favorable reversibility of the composites.

Fig.4b shows the first discharge-charge voltage profiles of the electrode made of bare  $\text{Co}_3\text{O}_4$  and the as-obtained composites, respectively. Unexpectedly, the first discharge capacity of the both materials have exceeded theoretical capacities, especially for the as-obtained composites, which are due to the decomposition of electrolyte and the formation of SEI layers[37]. As for the as-obtained composites, the especially large first discharge capacities might originate from the lower internal resistance (proved by a little lower discharge voltage plateau) [54] and the shortened the lithium ion

diffusion length of the smaller cobalt oxide particles (fig.3a) [31, 55, 56]. Moreover, the smaller particle size can also attribute in seriously decomposition of electrolyte.

As shown in fig.4c, The cycling performances of both electrodes have been measured at the current density of  $89 \text{ mAh}\cdot\text{g}^{-1}$ , indicating that the  $\text{Co}_3\text{O}_4\text{-CoO/C}$  electrode possesses a better performance. With regard to the  $\text{Co}_3\text{O}_4\text{-CoO/C}$  electrode, the reversible capacity declined up to the 10<sup>th</sup> cycle, and increased slowly to about  $580 \text{ mAh}\cdot\text{g}^{-1}$  due to the activation process of electrode materials or an ameliorative surface wetting performance of electrolyte[57, 58], subsequently showed a good capacity retention. For comparison, the bare  $\text{Co}_3\text{O}_4$  electrode reversible capacity continues to decline until the 50<sup>th</sup> cycle.

It has been reported that bare  $\text{Co}_3\text{O}_4$  suffers from extremely large volume changes and particle agglomeration during the reduplicative lithiation and delithiation which can also lead to the pulverization and exfoliation of the electrode due to the huge volume change, resulting in the on-going deterioration of the cell[43]. As for  $\text{Co}_3\text{O}_4\text{-CoO/C}$  electrode, the yolk-shell composites have improved electronic conductivity, with carbonous substrate facilitating the transmission of electron[59, 60]. Carbonous substrate can also avoid the aggregation of active materials and relieve the strain during lithium insertion and extraction[55], which is favorable for the prevention of the pulverization and exfoliation of the electrode, and beneficial for the maintenance of its cycling stability. All these reasons contribute to the excellent cycling performance.

The electrochemical performance of the  $\text{Co}_3\text{O}_4\text{/C}$  electrode is compared with previous studies as shown in Table 1. The as prepared composite displays competitive comprehensive electrochemical properties, implying that the composite has a promising prospect in the future.

**Table 1.** Comparison of the  $\text{Co}_3\text{O}_4\text{/C}$  electrode with similar electrodes reported in the literature

Material	Grain size	Morphology	Specific capacity ( $\text{mAh}\cdot\text{g}^{-1}$ )	Current density ( $\text{mA}\cdot\text{g}^{-1}$ )	Upgrade rate (%)	Cycle number	Ref.
$\text{Co}_3\text{O}_4\text{/C}$	Length 100nm Diameter 25nm	dendrite-like	500	89	238	40	35
$\text{Co}_3\text{O}_4\text{/C}$	10-20 nm	$\text{Co}_3\text{O}_4\text{/Carbon}$ matrix	822	100	40.7	30	36
$\text{Co}_3\text{O}_4\text{/C}$	250 nm	quasi-sphere	1150	222	29.2	20	37
$\text{Co}_3\text{O}_4\text{/graphene}$	10-30 nm	CGC	740	178	270	60	38
$\text{Co}_3\text{O}_4\text{-CoO/C}$	$\sim 5 \mu\text{m}$	Solid sphere	580	89	190	50	this work

#### 4. CONCLUSION

In summary, the present work demonstrated a novel preparation approach of  $\text{Co}_3\text{O}_4\text{-CoO/C}$  composites via a copolymerization of ionic liquid and acrylonitrile followed by carbonization method. The structure and morphology analysis revealed that the composites have an excellent yolk-shell structure and the interface between cobalt oxide and carbonous substrate bonds well, which is responsible for the superior cycling performance. In order to manifest the tremendous superiority, bare  $\text{Co}_3\text{O}_4$  electrodes were also fabricated, of which the capacity continuous to decline up to 50 cycles. The excellent cycling performance of  $\text{Co}_3\text{O}_4\text{-CoO/C}$  composites is due to improved electronic conductivity, reduced strain and depressed aggregation of active materials. According to these results, we can conclude that the  $\text{Co}_3\text{O}_4\text{-CoO/C}$  composites have a promising prospect as a new-style electrode materials of lithium ion batteries.

#### CONFLICTS OF INTEREST STATEMENT

The authors certify that they have NO affiliations with or involvement in any organization or entity with any financial interest (such as honoraria; educational grants; participation in speakers' bureaus; membership, employment, consultancies, stock ownership, or other equity interest; and expert testimony or patent-licensing arrangements), or non-financial interest (such as personal or professional relationships, affiliations, knowledge or beliefs) in the subject matter or materials discussed in this manuscript.

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