

Short Communication

Hydrothermal Synthesis of Spindle-Like Nanostructured LiFePO₄ Powders Mediated by Organic Acid

Yatang Dai, Li Ma, Huan Zhang, Yuhua Chen, Yulei Chen, Wei Wang *

School of Materials Science and Engineering, Southwest University of Science and Technology, Mianyang 621010, China.

*E-mail: w.wang@swust.edu.cn

Received: 2 November 2015 / Accepted: 15 December 2015 / Published: 1 January 2016

Well-crystallized spindle-like narrow size distribution (100~140 nm) LiFePO₄ nanoparticles for high rate lithium-ion battery cathode is synthesized via hydrothermal route employing amino tris(methylene phosphonic acid) (ATMP) as a surfactant. The samples are characterized by powder X-ray diffraction (XRD), dynamic Light Scattering (DLS), galvanostatic charge and discharge tests, scanning electron microscopy (SEM) and electrochemical impedance spectroscopy (EIS) measurements. Both the pure (F-blank) and with AMPT (F-AMPT) composites exhibit comparable capacities of 160.57 mAh g⁻¹ and 132.53 mAh g⁻¹ (0.1 C), 152.36 mAh g⁻¹ and 121.36 mAh g⁻¹ (0.5 C), and 126.57 mAh g⁻¹ and 95.57 mAh g⁻¹ (5 C), respectively. The capacity retention rates of F-AMPT sample over 50 cycles at 0.1 C is 95.57 % while F-blank yields 85.26 %. EIS measurements confirm that F-AMPT sample has an increased lithium ion diffusion coefficient of $1.62 \times 10^{-12} \text{ cm}^2 \text{ s}^{-1}$ due to its higher surface area.

Keywords: Hydrothermal synthesis; Surfactant; Nanoparticles; Lithium ion batteries;

1. INTRODUCTION

Among a series of Li ion battery materials, olivine-type LiFePO₄ has been considered as one of the most important, dynamic, and promising materials for electric or hybrid carriers, due to its outstanding high-rate theoretical capacity, superior capacity retention, excellent thermal stability, highly reversible and repeatable property, nontoxicity, safety, environmental benignity and potentially low cost [1-3]. However, the extremely poor electronic conductivity and low lithium diffusion constant restrict its commercial application.

Recently, a lot of researchers have been devoted to ameliorate the intrinsic character of bulk LiFePO₄ or with extrinsic modifications to applicable level. Numerous effective approaches include reducing particle size [4], surface coating the active nanoparticles with ultra-thin carbon, generating homogeneous particle size distribution nanocomposites with compositing of polymers or doping metal

cationic ions, to enhance the electronic transport between the nanocomposites and LiFePO_4 [5, 6]. These efforts have been testified useful to short the diffusion length of electrons and lithium ions transport [7].

Zheng *et al.* [8] have prepared hollow LiFePO_4 nanoparticles by solvothermal method, which show excellent cycling capability. Tian *et al.* [9] synthesized LiFePO_4/C nanorods with carbon coating by hydrothermal method. At high rates, the LiFePO_4/C nanorods exhibit higher power densities, and better cycling performance. Cho *et al.* [10] have prepared core-shell $\text{LiFe}(\text{PO}_4)(\text{OH})$ microspheres with the size distribution from 700 nm to more than 7 μm . However, these methods are too complicated or require expensive reagents to prepare the uniform LiFePO_4 nanoparticles or nanocomposites.

In this paper, we report a simple one-pot synthesis method to prepare spindle-Like LiFePO_4 nanocrystals with narrow size distribution by a facile hydrothermal process mediated by inexpensive amino tris (methylene phosphonic acid) (ATMP) surfactant. The synthesized LiFePO_4 nanocrystals possess considerably high electronic conductivity, remarkable large surface area, and an improved lithium ion diffusion coefficient.

2. EXPERIMENTAL

2.1. Materials and preparation

In our experiment, $\text{LiOH}\cdot\text{H}_2\text{O}$, $(\text{NH}_4)_2\text{FeSO}_4\cdot 7\text{H}_2\text{O}$ and H_3PO_4 were utilized as Li, Fe and P sources in the synthesis, respectively. In a typical experiment, 120 mmol $\text{LiOH}\cdot\text{H}_2\text{O}$ and 40 mmol $(\text{NH}_4)_2\text{FeSO}_4$ were dissolved in 3 mL H_3PO_4 solution (14 mol L^{-1}) and 20 mL ascorbic acid solution (0.1 mol L^{-1}), respectively. After that, the $\text{LiOH}/\text{H}_3\text{PO}_4$ solution was slowly dropwisely into the mixture. Subsequently, 0.0024 mol ATMP was injected into the solution with stirring. Then the mixture was stirred at $25\text{ }^\circ\text{C}$ for 5 min, and control the solution $\text{pH}=8$, by adding ammonia solution. For each time, 150 mL of the mixture was quickly transferred into the hydrothermal system (volume 250 mL). The hydrothermal reaction for preparing LiFePO_4 was carried out at $150\text{ }^\circ\text{C}$ overnight. Finally, the LiFePO_4 precipitates were washed with Milli-Q water and ethanol at least 3 times. After dried in vacuum at $110\text{ }^\circ\text{C}$ for 1 h, the precipitates were carefully collected. For the sake of comparison, phase-pure LiFePO_4 without ATMP was prepared in the parameters constant.

2.2. Characterization

The composition and phase of as-prepared spindle-Like LiFePO_4 nanoparticles were analyzed by X-ray diffraction (XRD) using Japan Neo-Confucianism Company. The morphologies of the samples were characterized by scanning electron microscope (SEM, Hitachi S-4800).

2.3. Electrochemical measurement

Electrochemical properties of LiFePO_4 samples were characterized by coin-type CR2025 lithium half-cells. The working cathode was prepared by 80 wt% LiFePO_4 , 15 wt% acetylene black

and 5 wt% poly(vinylidene fluoride) (PVDF) binder in N-methyl-2-pyrrolidone to form a homogeneous slurry. Then, the slurry was coated on an Al foil current collector and dried at 120 °C for 10 h in a vacuum oven. After that, CR2025 coin-type cells were assembled in an argon-filled glove box by utilizing the metal lithium foils as the anode, the Celgard 2400 microporous membrane as the separator and 1M LiPF₆ solution in a mixture of ethylene carbonate (EC) and dimethyl carbonate (DMC) with a volumetric ratio of 1:1 as the electrolyte.

The galvanostatic charge and discharge tests were conducted in a potential range 2.5–4.2 V (vs. Li/Li⁺) using a Land BT2001A automatic battery system at room temperature (RT, 25 °C). The electrochemical impedance spectroscopy (EIS) tests were performed by the IM6 electrochemical work station with the frequency ranging from 0.1 Hz to 100 kHz at a scanning rate of 5 mV s⁻¹.

3. RESULTS AND DISCUSSION

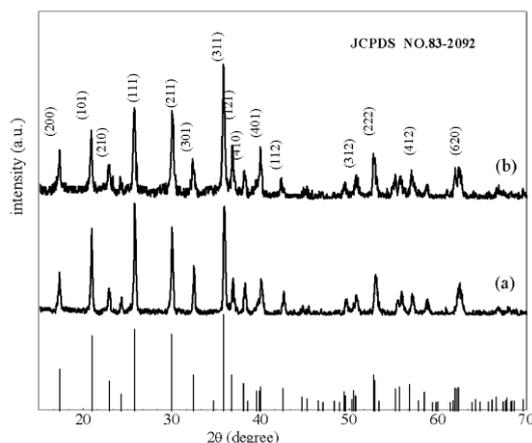


Figure 1. XRD patterns of as-prepared LiFePO₄ samples: (a) F-blank, (b) F-ATMP

Fig. 1 illustrates XRD results of as-synthesized LiFePO₄ nanoparticles with or without ATMP present. The LiFePO₄ nanoparticles show single phase, and the results are indexed well to the typical peaks of standard orthorhombic structure (JCPDS No. 83-2092). No impurities such as Li₃PO₄, trivalent Fe₂O₃ and Li₃Fe₂(PO₄)₃, which can be often found in the LiFePO₄ product prepared by traditional routes [11-13].

Fig. 2 shows SEM images of the as-synthesized spindle-Like LiFePO₄ nanoparticles. When ATMP is not applied, the obtained LiFePO₄ powders were aggregated with each other, which were 400 nm in wide and 200 nm in thickness. In contrast, the ATMP was proved to change the pattern of LiFePO₄ powders dramatically. As displays in Fig.2b, all of the individual crystals took the shape of discrete spindle-like particles with about 250 nm in length and 100 nm in diameter, indicating the useful of the appearance of the ATMP [14].

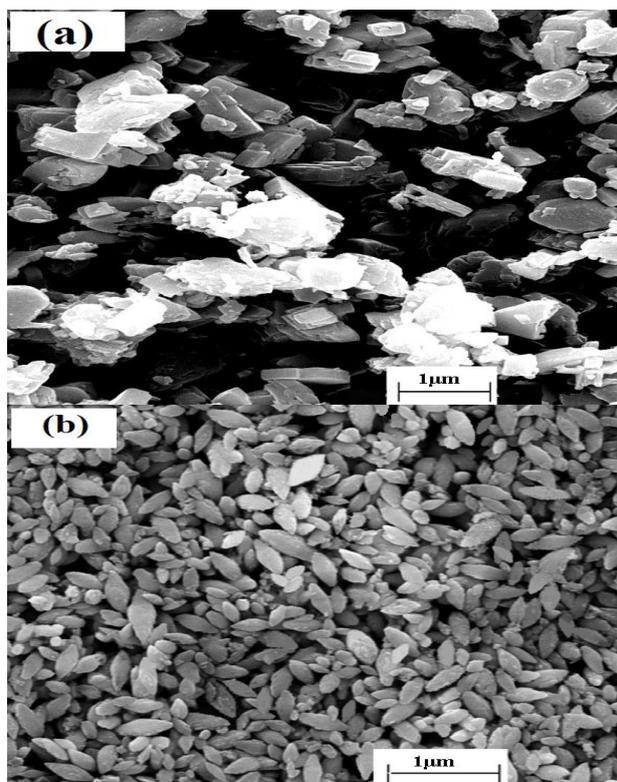


Figure 2. SEM images of the LiFePO_4 nanoparticles: (a) F-blank, (b) F-ATMP

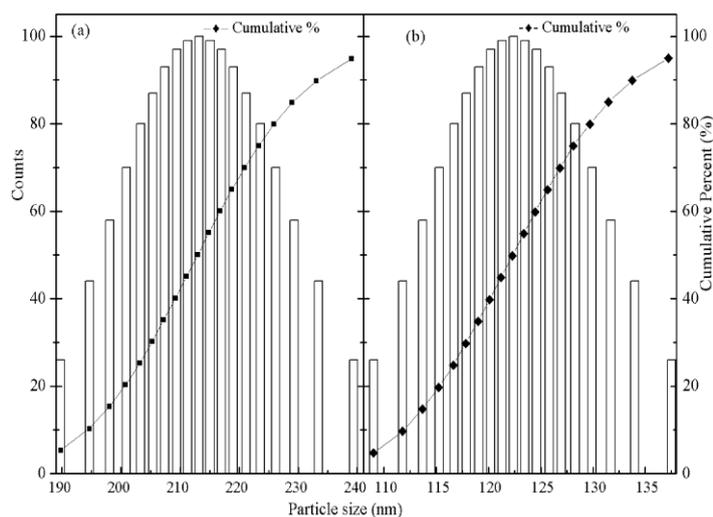


Figure 3. Size distribution of LiFePO_4 powders synthesized (a) without and (b) with ATMP

Fig. 3 shows the SEM images of the size distribution of LiFePO_4 nanoparticles. The images show that approximately 95% of F-AMPT is within 190–240 nm range with an average particle size (D_{50}) of ≈ 122 nm compared to that of ≈ 213 nm corresponding to the LiFePO_4 synthesized without the chelating agent. The result proves that the appearance of ATMP has effectively on the disperse of particles. The spindle-like LiFePO_4 nanoparticles show much smaller size and narrower size distribution.

3.2 Electrochemical measurements

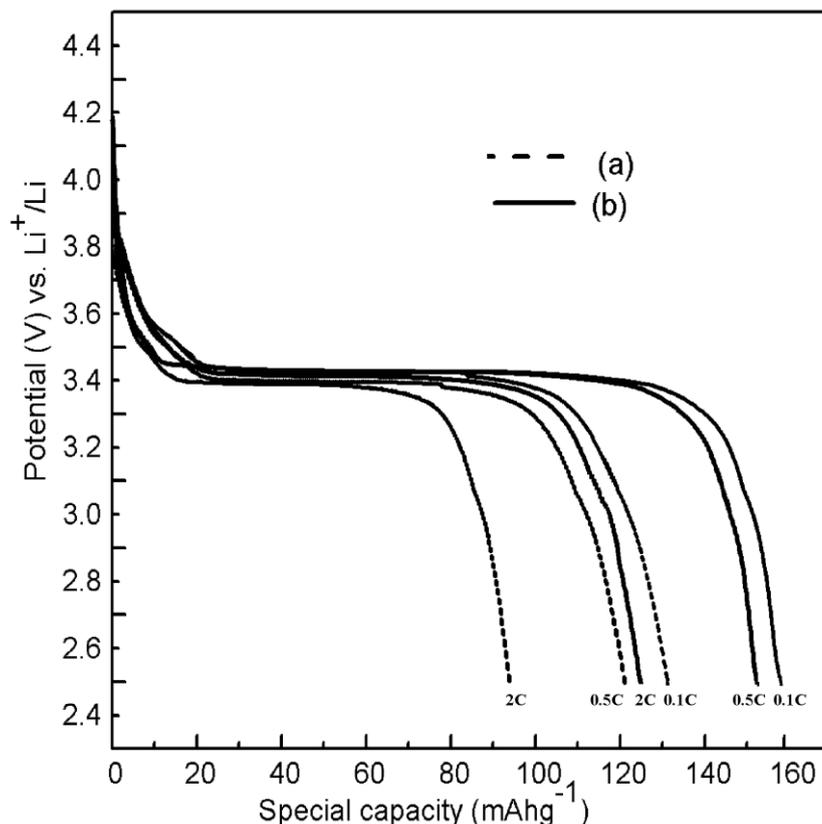


Figure 4. The initial discharge profiles of the LiFePO_4 synthesized (a) without ATMP and (b) with ATMP at various current rates (from 0.1 C to 2 C) between 2.5 V and 4.2 V at 25 °C.

Fig.4 shows the initial galvanostatic discharge curves of the LiFePO_4 cathodes at different current rates between 2.5 V and 4.2 V versus Li/Li^+ at room temperature. Both samples have similar discharge curves with flat plateaus corresponding to the lithium intercalation. The LiFePO_4 nanoparticles exhibit a flat plateau around 3.4 V for discharge voltage, corresponding well to the two-phase redox reaction between FePO_4 and LiFePO_4 system [15].

The spindle-like LiFePO_4 nanoparticles exhibit capacity of $160.57 \text{ mAh g}^{-1}$ at 0.1 C, $152.36 \text{ mAh g}^{-1}$ at 0.5 C, and $126.57 \text{ mAh g}^{-1}$ at 2 C, whereas the LiFePO_4 synthesized without ATMP yields $132.53 \text{ mAh g}^{-1}$ at 0.1 C, $121.36 \text{ mAh g}^{-1}$ at 0.5 C and 95.57 mAh g^{-1} at 2 C, respectively. These results are comparable with the capacities values of other reports [12-16].

In addition, the spindle-like LiFePO_4 nanoparticles electrode operated at 2 C exhibit 78.83% capacity retentions compared to its capacities at 0.1 C, which are higher than that of 72.11% for the LiFePO_4 synthesized without ATMP electrode. It is obviously that the spindle-like LiFePO_4 nanoparticles show much higher discharge capacity values and capacity retentions than the LiFePO_4 synthesized without ATMP.

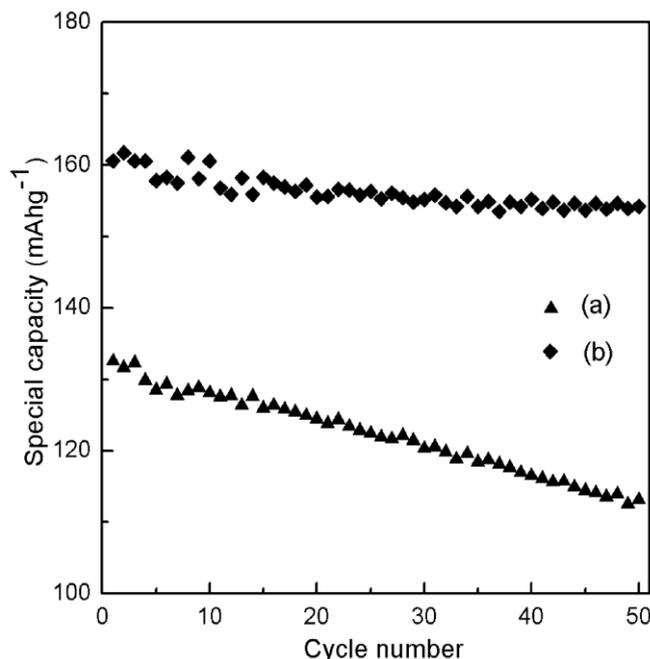


Figure 5. Cycle performances of LiFePO₄ (a) without and (b) with ATMP.

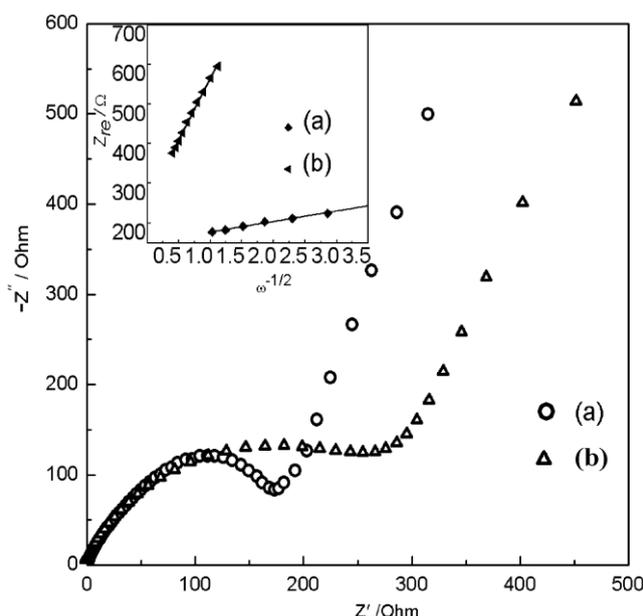


Figure 6. Electrochemical impedance spectroscopy (EIS) of LiFePO₄: (a) F-AMPT (b) F-blank and Inset graph of Z' plotted against $\omega^{-1/2}$.

Fig. 5 compares the cycling performance of as-prepared two samples over limited cycle at 0.1 C. Compared with the F-blank, the F-ATMP sample displays enhancement electrochemical performance. The F-ATMP discharges 153 mAh g⁻¹ with capacity maintaining ratios of 95.57 %. In contrast, F-blank yields 113 mAh g⁻¹ with capacity retention ratios of 85.26 %. It is obviously that the F-ATMP shows enhancing cycle performance of LiFePO₄, corresponding to its narrow and homogeneous particle size.

Subsequently, we characterized the electrochemical impedance spectroscopy in fully charged condition. Fig.6. presents Nyquist plots of LiFePO₄ electrodes with and without ATMP in the frequency range from 100 kHz to 10 mHz. As illustrated in Fig. 6, the impedance spectra can be explained according to an equivalent circuit with electrolyte resistance (R_e), charge transfer resistance (R_{ct}), double layer capacitance and passivation film capacitance (CPE) and Warburg Impedance (Z_w) [17]. Both profiles have a semicircle in the high-frequency region and a straight line in the low-frequency region, which are attributed to the charge-transfer resistance of electrochemical reaction and diffusion-controlled Warburg impedance, respectively [18]. The straight line is ascribed to the Warburg diffusion [15], and the lithium ion diffusion coefficient is calculated by the following equations [19, 20]:

$$D = R^2 T^2 / 2 A^2 n^4 F^4 c^2 \sigma^2 \quad (1)$$

$$Z_{re} = R_D + R_L + \sigma \omega^{-1/2} \quad (2)$$

where A is the electrode surface area, n is electrons per molecule during oxidization number, F is the Faraday constant, R is the gas constant, T is the absolute temperature, σ is the Warburg factor which can be obtained from the slope of the straight line of $Z_{re} - \omega^{-1/2}$ (Inset plot shown in Fig. 8), it corresponding to Eq. (2), C is the lithium ion concentration in LiFePO₄ electrode which is relative with.

$$c = n/V = m/MV = \rho V / MV = \rho / M \quad (3)$$

where ρ is the density, M is the molecular weight, C is calculated to be 5.67×10^{-3} mol cm⁻³ by Eq. (3). The lithium ion diffusion coefficient of F-ATMP sample is 1.62×10^{-12} cm²s⁻¹, while the value of F-blank sample is 1.85×10^{-14} . It is apparently that the F-ATMP sample generates a higher lithium ion diffusion coefficient, due to the smaller spindle-like structure. The pathway for Li ion transfer of F-ATMP is short, which contribute to better electrochemical performance [21], corresponding to the charge and discharge results.

4. CONCLUSION

A spindle-like LiFePO₄ particle was synthesized by hydrothermal method in the presence of ATMP. SEM illustrates that the LiFePO₄ mediated by ATMP consists of nearly homogenous nanostructures with size of 100 nm; in contrast, the F-blank comprises large agglomerates.

Both the F-blank and F-AMPT composites exhibit comparable specific capacities of 160.57 mAh g⁻¹ vs. 132.53 mAh g⁻¹ at 0.1 C, 152.36 mAh g⁻¹ vs. 121.36 mAh g⁻¹ at 0.5 C, and 126.57 mAh g⁻¹ vs. 95.57 mAh g⁻¹ at 5 C, respectively. The capacity retention rates of F-AMPT samples over 50 cycles at 0.1 C is 95.57% (vs. the first-cycle corresponding C-rate capacity) while F-blank yields 85.26%. The spindle-like LiFePO₄ nanoparticles with small size and narrow size distribution produced excellent electrochemical performance, due to their rough surfaces. It is believed that the spindle-like morphology increases the contact area between the spindle-like LiFePO₄ nanoparticles and the electrolyte and reduces the Li⁺ diffusion length.

ACKNOWLEDGEMENTS

We acknowledge the financial supports of the National Natural Science Foundation of China (Grant No. 31400811) and Doctoral Research Fund of Southwest University of Science and Technology (No. 12zx7131).

References

1. N. Yabuuchi, K. Yamamoto, K. Yoshii, I. Nakai, T. Nishizawa, A. Omaru, T. Toyooka and S. Komaba, *J. Electrochem. Soc.*, 160 (2013) A39-A45.
2. B. Jin and H. B. Gu, *Solid State Ionics*, 178 (2008), 1907-1914.
3. Z. G. Lu, H. L. Chen, R. Robert, B. Y. X. Zhu, J. Q. Deng, L. J. Wu, C. Y. Chung and C. P. Grey, *Chem. Mater.*, 23 (2011) 2848-2859.
4. K. Dokko, S. Koizumi, K. Sharaishi and K. Kanamura, *J. Power Sources*, 165 (2007) 656-659.
5. J. Z. Li, S. Y. Wang, L. H. Wei, Z. Liu and Y. W. Tian, *Int. J. Electrochem. Sci.*, 10 (2015) 6135-6145.
6. B. Pei, Q. Wang, W. X. Zhang, Z. H. Yang and M. Chen, *Electrochim. Acta*, 56 (2011) 5667-5672.
7. X. Y. Zhang, M. van Hulzen, D. P. Singh, A. Brownrigg, J. P. Wright, N. H. van Dijk and M. Wagemaker, *Nat. Commun.*, 6 (2015).
8. Z. M. Zheng, W. K. Pang, X. C. Tang, D. Z. Jia, Y. D. Huang and Z. P. Guo, *J. Alloy. Compd.*, 640 (2015) 95-100.
9. R. Y. Tian, G. Y. Liu, H. Q. Liu, L. N. Zhang, X. H. Gu, Y. J. Guo, H. F. Wang, L. F. Sun and W. G. Chu, *RSC Adv.*, 5 (2015) 1859-1866.
10. M. Y. Cho, Y. S. Lim, S. M. Park, K. B. Kim and K. C. Roh, *Crystengcomm.*, 17 (2015) 6149-6154.
11. L. Wang, W. T. Sun, X. Y. Tang, X. K. Huang, X. M. He, J. J. Li, Q. W. Zhang, J. Gao, G. Y. Tian and S. S. Fan, *J. Power Sources*, 244 (2013), 94-100.
12. R. Shahid and S. Murugavel, *Phys. Chem. Chem. Phys.*, 15 (2013) 18809-18814.
13. Y. Xia, W. K. Zhang, H. Huang, Y. P. Gan, J. Tian and X. Y. Tao, *J. Power Sources*, 196 (2011) 5651-5658.
14. G. Meligrana, C. Gerbaldi, A. Tuel, S. Bodoardo and N. Penazzi, *J. Power Sources*, 160 (2006) 516-522.
15. J. L. Liu, R. R. Jiang, X. Y. Wang, T. Huang and A. S. Yu, *J. Power Sources*, 194 (2009) 536-540.
16. D. Choi and P. N. Kumta, *J. Power Sources*, 163 (2007) 1064-1069.
17. G. H. Qin, S. Xue, Q. Q. Ma and C. Y. Wang, *Crystengcomm*, 16 (2014) 260-269.
18. L. Wang, L. B. Yang, L. Gong, X. Q. Jiang, K. Yuan and Z. B. Hu, *Electrochim. Acta*, 56 (2011) 6906-6911.
19. Y. D. Liu, Q. Liu, Z. F. Li, Y. Ren, J. Xie, H. He and F. Xu, *J. Electrochem. Soc.*, 161 (2014) A620-A632.
20. O. A. Drozhzhin, M. A. Vorotyntsev, S. R. Maduar, N. R. Khasanova, A. M. Abakumov and E. V. Antipov, *Electrochim. Acta*, 89 (2013) 262-269.
21. X. Y. Wang, H. Hao, J. L. Liu, T. Huang and A. S. Yu, *Electrochim. Acta*, 56 (2011) 4065-4069.