Short Communication

# Facile Synthesis of High-quality N-doped Graphene Anchored with Fe<sub>2</sub>O<sub>3</sub> for Use As Lithium-ion Battery Anode Materials

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A hybrid material composed of nitrogen-doped graphene anchored with  $Fe_2O_3$  was readily obtained by a facile and one-step hydrothermal method followed by assisting with microwave under mild conditions. Scanning electron microscopy and transmission electron microscopy images show that  $Fe_2O_3$  with average size of 30-50 nm are homogeneously distributed and anchored on the surface of nitrogen-doped graphene sheets. The N-doped graphene/ $Fe_2O_3$  hybrid electrodes exhibit excellent electrochemical performance with the charge/discharge capacity maintain stable at around 844.3 mAhg<sup>-1</sup> and 937.7 mAhg<sup>-1</sup> after the 40 th cycle. The enhanced lithium-storage performance is ascribed to the rational design of the hybrid and has a great potential as anode material for lithium ion batteries.

Keywords: lithium-ion battery; anode materials; graphene; Fe<sub>2</sub>O<sub>3</sub>

## **1. INTRODUCTION**

For now, rechargeable lithium ion batteries (LIBs) are widely utilized in a range of secondary energy storage devices of portable electronic devices due to their high power density, high safety and long cycle life [1-3]. However, the conventional graphite electrode shows a relatively low specific capacity (372 mAH g<sup>-1</sup>) calculated by forming the compound of LiC<sub>6</sub> which cannot meet the current demands for high capacity lithium ion batteries [4]. Transition metal oxides (MO) such as Fe<sub>3</sub>O<sub>4</sub>[5], SnO<sub>2</sub> [6], Co<sub>3</sub>O<sub>4</sub> [7], MoO<sub>2</sub> [8] and NiO [9] with enhanced electrochemical properties have been extensively investigated as anode materials for LIBs. AS one of the most promising anode materials, Fe<sub>2</sub>O<sub>3</sub> has received much attention owing to its high theoretical capacity, low cost and plenty [10,11]. However, hindered by a large volume expansion during Li<sup>+</sup> insertion /extraction, Fe<sub>2</sub>O<sub>3</sub> shows poor electrochemical property as anode material for LIBs [12,13]. To overcome the drawback caused by mentioned above, graphene is the most promising matrix to anchor nanoparticles and alleviate the large volume change due to its inherent outstanding properties such as large specific surface area, high electrical conductivity, good flexibility, and high chemical stability [14]. Furthermore, nitrogen-doped grapheme (NGr) is one typical form of chemically-modified graphene compounds which has been aroused the attention attributing to its superior specific capacity for the introduction of nitrogen atoms with numerous outstanding properties such as the increased disorder on the surface, excellent hydrophobicity, improved electrode electrolyte wettability, and enhanced electrochemical activity[15-17].

Herein, we demonstrate a facile and reliable approach to fabricate high-quality N-doped graphene anchored with  $Fe_2O_3$  for lithium ion battery anode with excellent electrochemical performance. Moreover, no toxic materials have been used in the whole process of fabrication. The morphology, structure, electrochemical properties and the first charge discharge capacity of N-doped graphene anchored with  $Fe_2O_3$  were also investigated. It is showed that the nanocomposite can fairly meet the current demand for high capacity and quality as the anode material of LIBs.

### **2. EXPERIMENT**

### 2.1 Preparation of Material

### 2.1.1 Preparation of nitrogen doped graphene anchored with $Fe_2O_3$

All reagents used in the experiment were analytical pure without any further purification. Graphene oxide(GO) was fabricated from natural graphite flake(size $\approx$ 2 µm) by using a modified Hummers method which has been described detailed in our previous work[18]. The nitrogen doped graphene anchored with Fe<sub>2</sub>O<sub>3</sub> (N-doped graphene/Fe<sub>2</sub>O<sub>3</sub>) was synthesized by a facile method and onestep of simultaneous hydrothermal synthesis and assembly procedure. A typical synthesis of the N-doped graphene/Fe<sub>2</sub>O<sub>3</sub> hybrids was as follows: 10.8 g urea was added into the 60 ml 2 mg ml<sup>-1</sup> GO solution by ultrasonication stirring technology for 30 min. Then 6 m mol FeSO<sub>4</sub>·6H<sub>2</sub>O was introduced into the above solution followed by ultrasonication for another 30 min. The whole mixture as the precursor was transferred to a 100 ml polytetrafluoroethylene Teflon-lined stainless steel autoclave, sealed and maintained at 160 °C for 16 h. After the reaction completed, the N-doped graphene/Fe<sub>2</sub>O<sub>3</sub> collected by centrifugation, filtration, and washed several times with distilled water and anhydrous ethanol followed by vacuum drying for 12 h. Finally, the production was prepared by microwave treating for 2 min using a household microwave oven (700 W, Galanz) to eliminate the oxygen functional groups on the surface of graphene sheets. Fig.1 shows the shcematic procedure for preparation of N-doped graphene/Fe<sub>2</sub>O<sub>3</sub> hybrid.

## 2.1.2 Characterization of samples

The transmission electron microscopy (TEM) and high-resolution TEM (HRTEM) images were obtained by using a JEOL, JEM-2100F, with the acceleration voltage of 200 kV. Preparation of

the testing samples were ultrasonically dispersed in ethanol and deposited onto copper grids covered with a holey carbon film. The morphologies of the samples were investigated by scanning electron microscopy (SEM), JEOL, JSM-7001F, with the acceleration voltage of 15 kV. The X-ray diffraction (XRD) patterns of the synthesized samples were studied by an X'Pert Pro diffractometer (PANalytical Co., Holland) with Cu K $\alpha$  radiation (40 kV and 40 mA) at room temperature.

## 2.1.3 Electrochemical measurement

Electrochemical performance of the composite electrodes assembled in a coin cell configuration (CR 2032-type), and were tested in a LAND battery testing system which conducted between 0.01-3 V vs Li<sup>+</sup>/Li with the lithium metal foil working as the counter electrode. A slurry coating procedure was utilized to prepare the working electrode which consist of active materials, carbon black and PVDF with a weight ratio of 80:10:10 followed by mixing sufficiently in N-methylpyrrolldinone (NMP), after that the samples coated in the Cu foil and dried under at 80 °C overnight in vacuum. Before the whole assembly procedure, all the samples were needed to dry at 75 °C for 3 h to remove the residual moisture and impurities.

The 2032 coin cell were assembled in an argon-filled glove box with the electrolyte consisting of 1M LiPF6 dissolved in the solution with 1:1 ethylene carbonate and diethyl carbonate by volume, and the separator (Celgard 2340, USA) was added between the working electrode and the counter electrode. All the electrochmical measurements were carried out at room temperature.



Figure 1. Schematic illustration for the synthesis mechanism of the N-doped graphene/Fe<sub>2</sub>O<sub>3.</sub>

# **3. RESULTS AND DISCUSSION**

The phase structure of the testing material was characterized by XRD measurement. The XRD pattern of the N-doped graphene/Fe<sub>2</sub>O<sub>3</sub> hybrid nanocomposite can be observed in Fig.2. There are two

different Fe<sub>2</sub>O<sub>3</sub> crystal structures coexisting in the hybrids which are hematite( $\alpha$ ) and maghemite ( $\gamma$ ) Fe<sub>2</sub>O<sub>3</sub>.For the samples, the diffraction peaks appear at 20=20.04°, 24.17°, 30.11°, 33.05°, 35.51°, 40.89°, 43.13°, 49.35°, 53.90°, 57.47°, 62.44° and 63.98° corresponding to the (200), (012), (411), (104), (110), (113), (202), (024), (116), (122), (214) and (300) planes of  $\alpha$  and  $\gamma$ crystal phase Fe<sub>2</sub>O<sub>3</sub>, respectively. No excess peaks of impurities were shown in the XRD which indicating that Fe<sub>2</sub>O<sub>3</sub> particles are absolutely formed on the surface of the nitrogen doped graphene without blending any impurity matters. Besides, the characteristic diffraction peak at about 20=25° of graphene cannot be observed which suggesting no aggregation of nitrogen doped graphene happened during the hydrothermal process. The strong interfacial bonding between Fe<sub>2</sub>O<sub>3</sub> nanoparticles and nitrogen doped graphene and prevent the aggregation of nitrogen doped graphene and prevent the aggregation of nitrogen doped graphene [19,20].



**Figure 2.** X-ray diffraction (XRD) pattern of the N-doped graphene/Fe<sub>2</sub>O<sub>3</sub> with different Fe<sub>2</sub>O<sub>3</sub> crystal structures Fe<sub>2</sub>O<sub>3</sub>.

The surface morphology and structure of the N-doped graphene/Fe<sub>2</sub>O<sub>3</sub> were investigated by SEM as shown in Fig.3. It can be clearly observed in the Fig.3(a-c) that a great number of Fe<sub>2</sub>O<sub>3</sub> nanosize particles are uniformly and homogeneously distributed on the nitrogen doped graphene sheets which are crumpled to a curly and wavy shape. In addition, the high magnification SEM image (Fig.2(c)) suggests that the nitrogen doped graphene sheets well permeated in Fe<sub>2</sub>O<sub>3</sub> nano particles, indicating the successful formation of N-doped graphene/Fe<sub>2</sub>O<sub>3</sub> nanocomposite. For further research of the detailed microstructure of the N-doped graphene/Fe<sub>2</sub>O<sub>3</sub> are clearly revealed in Fig.3(d-f). It can also be obviously observed that the Fe<sub>2</sub>O<sub>3</sub> nano particles are about 30-50 nm in size and are homogeneously distributed on the surface of graphene sheets which are wrinkled transparent flakes. The agglomeration of Fe<sub>2</sub>O<sub>3</sub> nano particles with graphene sheets can promote electron transfer during charge/discharge. The NH<sub>3</sub> released from urea boost the formation of pores on the surface of iron oxides during a long hydrothermal process. The HR-TEM image (Fig. 3f) shows the Fe<sub>2</sub>O<sub>3</sub> nano particles were surrounded by curly graphene sheets. The lattice fringe spacing of the 0.252 and 0.208 nm correspond in sequence to the inter planar spacing between (110) and (202) planes of  $\alpha$  and  $\gamma$ Fe<sub>2</sub>O<sub>3</sub>, respectively, which is identical to the previous XRD curve.



**Figure 3.** Microstructure and morphology of the N-doped graphene/Fe<sub>2</sub>O<sub>3</sub> hybrid electrodes. SEM(ac) and TEM(d-f) images of the N-doped graphene/Fe<sub>2</sub>O<sub>3</sub> composite.

The charge-discharge profiles of the N-doped graphene/Fe<sub>2</sub>O<sub>3</sub> hybrid electrodes for the 1st, 2 nd, 10 th and 50 th cycles at a current density of 100 mAg<sup>-1</sup> were shown in Fig.4. Two obvious plateaus can be observed in the first discharge profiles. The discharge capacities of the hybrid in the 1 st, 2 nd, 10 th and 50 th cycle are 742.6, 678.6, 704.5, 937.7 mAhg<sup>-1</sup>, respectively. The charge capacities of the hybrid in the 2 nd, 10 th and 50 th cycle are 555.9, 610.7 and 844.3 mAhg<sup>-1</sup>, respectively. It is no doubt that the whole cycles have a relatively high Coulombic efficiency except the first cycle which suffered a great capacity loss owing to the formation of solid electrolyte interface (SEI) layer [4,21-23]. It is suggesting that the hybrids electrodes own high electrochemical reversibility as anode materials for lithium ion batteries. The cyclic properties of the N-doped graphene/Fe<sub>2</sub>O<sub>3</sub> hybrid electrodes at current density of 100 mAg<sup>-1</sup> are revealed at Fig.5. The initial discharge/charge capacity of the N-doped graphene/Fe<sub>2</sub>O<sub>3</sub> hybrid electrodes is 742.6 mAhg<sup>-1</sup> and 555.9 mAhg<sup>-1</sup>, respectively, at the current density of 100 mAg<sup>-1</sup>. The irreversible loss of capacity in the first cycle may be attributed to the formation of SEI layer. After that cycle, the charge and discharge capacity gradually raise from 555.9, 678.6 mAhg<sup>-1</sup> to 843.8 and 936.7 mAhg<sup>-1</sup>, respectively, in the 40 th cycle. It is generally considered as the activation process of the material [24]. Then the charge/discharge capacity maintain stable at around 844.3 mAhg<sup>-1</sup> and 937.7 mAhg<sup>-1</sup> after the 40 th cycle. This phenomenon may be due to the N-doped graphene/Fe2O3 composite lose its crystallinity transform to an amorphous structure, thus enhancing the Li+ diffusion kinetics so that more Li+ can insert into or extract from the material. This excellent capacities might be due to the unique structure of hybrid and the microwave treatment which enhancing the conductivity of hybrid anode by removing the excess oxygen functional groups of graphene sheets.



**Figure 4.** Charge/discharge profiles of the N-doped graphene/ $Fe_2O_3$  hybrid electrodes for the 1st, 2 nd, 10 th and 50 th cycles at a current density of 100 mAg<sup>-1</sup>.



**Figure 5.** Cyclic performance of the N-doped graphene/Fe<sub>2</sub>O<sub>3</sub> hybrid electrodes in the voltage range of 0.01 and 3.0 V at current density of 100 mAg<sup>-1</sup>.

## 4. CONCLUSIONS

In summary, a facile and one-step fabrication method was invited to fabricate a nitrogen-doped graphene and  $Fe_2O_3$  with a average size 30-50 nm, were homogeneously distributed and anchored on the surface of nitrogen-doped graphene sheets. The obtained N-doped graphene/Fe<sub>2</sub>O<sub>3</sub> hybrid shows excellent nanostructure morphology and electrochemical properties as the anode material for lithium ion batteries. This is due to the unique architecture of the hybrid in which graphene can be treated as the buffer that not only avoid the Fe<sub>2</sub>O<sub>3</sub> nanoparticles aggregating on the surface of the nitrogen-doped graphene but also ease the severe volume change during the charge/discharge process. The enhanced lithium-storage performance is ascribed to the rational design of the hybrid and has a great potential as anode material for LIBs.

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