# **Enhancing Performance of a Lithium Ion Battery by Optimizing** the Surface Properties of the Current Collector

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A thin uniform and compact oxide layer (COL) was synthesized a on an Al foil as a current collector for Li-ion battery cathode by an oxidation process using KMnO<sub>4</sub> as an oxidant. The oxide composite mainly consists of manganese and aluminium oxides which were determined by EDX technique. The morphologies of the as-prepared COL modified Al current collector were characterized by AFM and SEM. Electrochemical measurements were performed to evaluate the corrosion behaviour of the current collectors in Li-ion battery (LIB) electrolyte. And the adhesion property of current collectors' surface to battery electrode materials layer was examined using a peel test technique. It was found the COL on an Al current collector can effectively protect it from corrosion and also exhibits good adhesion properties to the electrode materials layer. Such good properties are quite beneficial to the electrochemical performance and cyclability of Li-ion batteries for extensive cycle life test evaluations which were confirmed by electrochemical characterizations and cycle life tests.

Keywords: lithium ion battery; anti-corrosion; adhesion; current collector.

## **1. INTRODUCTION**

Extensive effort has been continuously devoted to the fast-growing technologies for rechargeable LIBs. The majority of interests have focused on designing and developing new and high performance materials for the anode, cathode, and electrolyte. [1-12] In comparison, far less attention has been paid on the current collectors. Typically the anode and cathode of lithium ion batteries are made of active materials which are supported on metal foil current collectors. The metal foils are normally copper or aluminium which serve as electrical conductors and mechanical supports.

Improved cycle life, gravimetric and volumetric energy densities, and power densities of the Li-ion batteries require not only high performance cathode and anode materials, but high anti-corrosion ability of electrode and good adhesion properties within the electrode composite and at its interface with the current collector. Only with these optimized components, the Li-ion batteries can maintain to deliver the designed energy and power output under operation conditions.

Typically good adhesion properties between the electrode composite and current collectors' interface should be achieved to deliver the designed power output and cyclability throughout the service period. In order to enhance the surface adhesion properties of the current collectors, surface roughing process, roll pressure, and addition of special binders are commonly employed to maintain the mechanical durability of the electrodes. Such mechanical enhancements have proven beneficial to the cyclability and performance of batteries. [13-15] Besides, in practice, commercial current collectors are often not 100% pure in their metal content. Trace of impurity content may result in continued corrosion of current collectors which can lead to a significant contribution to the Li-ion batteries' internal resistance of cells, which further results in power and capacity fade. [16] Therefore, special treatment on current collectors to form protect layer on their surface is conducive to maintain the battery performance. Normally, a metal can passivate to enhance its anti-corrosion ability via different ways such as solvent chemisorptions, air-formed layer, alloying additions of transition metals, hydroxide/oxyhydrogxide formation, and a salt film consisting of metal cations and anions from the electrolyte, etc. [17-22] However, to obtain a uniform and effective protect layer on current collectors is still so far challenging.

Herein we demonstrates in this study a simple and low cost strategy which was employed to prepare a protective oxide thin layer coated Al foil as a cathode current collector for Li-ion batteries. In this work, we also found the formation of the COL on Al foil not only improves the anti-corrosion ability of the current collector in Li-ion battery electrolyte, but enhances its adhesion property to LiFePO<sub>4</sub> cathode laminate. The as-prepared current collectors were characterized physically and the related corrosion behavior was investigated by electrochemical technique. The corresponding current collectors were employed to fabricate into Li-ion full cell batteries and the related cycle life evaluations were performed.

## 2. EXPERIMENTAL

#### 2.1 Materials

Potassium Permanganate (KMnO<sub>4</sub>), Sodium Hydrooxide (NaOH), Sodium Carbonate (Na<sub>2</sub>CO<sub>3</sub>), Sodium dodecyl sulfate (SDS), Potassium fluozirconate (K<sub>2</sub>ZrF<sub>6</sub>), and Nitrate Acid (HNO<sub>3</sub>) were purchased from Sinopharm Chemical Reagent Co., Ltd, China and used as received. The Iron phosphate (LiFePO<sub>4</sub>) was commercial product from Aleees Eco Ark Co., Ltd., Taiwan. The electrolyte used throughout the electrochemical and cycle life testing was BLE-8021 commercial product from BICR.

#### 2.2 Synthetic procedures

The compact oxide layer on aluminium current collector (denoted as COL-Al) was prepared as follows. A commercial pristine Al current collector was used and cleaned up with acetone for several times to remove organic species on the surface, and then subjected to alkaline (an aqueous solution containing 0.75 M NaOH and 0.125 M Na<sub>2</sub>CO<sub>3</sub>) treatment to enhance its adhesion with the oxide composite layer. The resulted hydroxides and oxides by NaOH and Na<sub>2</sub>CO<sub>3</sub> were washed out from the aluminium collector with diluted HNO<sub>3</sub> solution and rinsed with deionized water to remove the residual acid. The cleaned current collector was immersed in an aqueous solution containing 0.03 M KMnO<sub>4</sub>, 0.15 mM SDS, and 0.15 mM K<sub>2</sub>ZrF<sub>6</sub> for 20s. The pH value of the solution was maintained at 2 by adjusting the concentration of HNO<sub>3</sub> in the solution for the reaction. The modified Al foil was washed with deionized water to remove the excess reactants and dried in a vacuum oven at 60 °C for 3 hours.

#### 2.3 Characterizations

The surface properties of the aluminium foils before and after treatment were characterized with an atomic force microscope (AFM, Multimode Nanoscope). The morphologies and EDX analysis of the Al foils surface were investigated using a FEI Nova NanoSEMTM 450 scanning electron microscope (SEM) equipped with an EDAX genesis 2000 (Amtek GmbH, Wiesbaden, Germany). Powder X-ray diffraction (XRD) measurements were performed using a Bruker D8 diffractometer with Cu K $\alpha$  radiation. The room-temperature conductivity of the current collectors was measured by a Jandel standard four-probe conductivity meter (Jandel Engineering Limited, U.K.). The peel tests were conducted using a CMT8501 Electromechanical Universal Testing Machine (MTS Systems (China) Co., LTD) to evaluate the adhesion properties of electrode. The procedure was similar to the reported elsewhere [13].

#### 2.4 Electrochemical measurements

The chronoamperometric curves were recorded using a homemade air-proof three-electrode set up connected with a CHI 760E bipotentiostat (CH Instruments, USA). The working electrodes were COL-A1 and commercial pristine Al current collectors. A lithium foil and a lithium ring were served as counter and reference electrode respectively. The assembly was conducted in an argon filled glove box. The electrolyte was 1.0 M LiPF<sub>6</sub> dissolved in a 50:50 w/w mixture of ethylene carbonate and diethyl carbonate. Electrochemical impedance spectra were measured using the same three-electrode cell while the working electrodes were Al current collectors with LiFePO<sub>4</sub> cathodic materials which were prepared same as the electrochemical interface (Solartron SI 1287) was used to perform the electrochemical impedance spectra (EIS) recording of the cells. The applied amplitude of the AC signal was 5 mV over the frequency range between 1 M Hz and 0.01 Hz. The charge/discharge performance of batteries was characterized by using CR2016-type coin cells which were assembled in an argon-filled glove box. The working electrodes were made of 91.7 wt% LiFePO<sub>4</sub>, 3.7 wt% carbon black, and 4.6 wt% LA133 binder which were supported on Al current collectors. Lithium foil was used as both reference and counter electrode. The Celgard 2000 microporous membrane was used as separator. For the full cell assembling, the anode was a thin film containing 95 wt% graphite, 1 wt% carbon black, and 4 wt% LA133 binder which were supported on a Cu current collector. The cycle life tests of the full cells were conducted using our commercially designed 25 Ah single full cell battery device which were assembled and activated in our factory (Sinopoly Battery Limited, Jilin, China). The long-term cycle life tests were conducted with a potential window between 2.0 and 3.8 V using a BTS 5V/50A Battery Testing System (Neware Electronic Co., China) at a 1 C charge-discharge rate.

## **3. RESULTS AND DISCUSSION**

Normally commercial pristine Al foil current collector has a very smooth and white surface with a low effective interfacial area. In air a thin and dense oxide  $(Al_2O_3)$  passive layer often forms on its surface due to the activity of Al element. To enhance the surface property which is conducive to form a robust electrode, we synthesized a high surface area oxide passive layer which was expected to replace the smooth  $Al_2O_3$  layer on the Al foil by a simple and quick dip-soak method. In the treatment process by KMnO<sub>4</sub>, the presence of  $K_2ZrF_6$  is critical for the formation of high quality and robust compact oxide film on the Al foil. The SDS used here serves as a surfactant which helps form uniform oxide layer. After the oxidation by KMnO<sub>4</sub> solution, a thin and light yellow film was formed on the surface of Al foil.



**Figure 1.** AFM images of the surfaces of pristine commercial Al foil (a) and COL-Al current collector (b).

As shown in the AFM image in Figure 1 (b), the resulted surface of Al foil shows rough and compact structure of a uniform bump pattern. Obviously, the rough surface will provide more contact area for the electrode laminate and the current collector. This process typically yields amorphous  $MnO_2$  and  $Al_2O_3$  mixture products which can be explained by EDX shown in Figure 2. [23, 24]



Figure 2. EDX spectrum of COL-Al and XRD patterns of the two Al current collectors.



Figure 3. SEM image and elemental mapping images of the COL-Al surface.

Figure 3 shows the SEM image of the COL-Al surface and the corresponding energy dispersive X-ray spectroscopy (EDS) mappings over a relatively large area for the Al, Mn, and O, which indicates

all the elements were distributed throughout the surface and also the surface was functioned with a mixture of manganese and aluminum oxides. EDX measurements were carried out to determine the components content for the oxide layer. The corresponding EDX result shown in Figure 2 presents that the atomic ratio of Mn to O is 1.44:3.74. It is clear that the atomic proportion of O is excess in all the stoichiometric ratios of Mn to O in various manganese oxides. That suggests that the surface composite mainly consists of not only manganese oxide, but aluminium oxide. Assuming all the O is from the structure of oxides, the molar ratio of manganese oxide to aluminium oxide is 1.44 to 0.57 on the current collector surface.

The adhesion strength plays a critical role for keeping contact of the electrode materials coating and the current collector, which will greatly influence mechanical stability of the electrodes and further the cyclability of the battery. In order to evaluate the adhesion property difference for the different surfaces, we casted a very thin cathode laminate layer (50  $\mu$ m) on the Al foils and then performed 180°peel test to evaluate the adhesion of the Al current collectors with LiFePO<sub>4</sub> /binder/carbon black layer in a LIB cathode.



**Figure 4.** Force-displacement curves of the LiFePO4/binder/carbon black on pristine Al (a), and COL-Al current collector (b) in peel tests.

From the force-displacement curves shown in Figure 4, the average peel forces are 3.93 N and 3.04 N for LiFePO<sub>4</sub>/binder/carbon black layer on COL-Al and the pristine Al foil current collector, respectively, which indicates the presence of rough oxide layer could enhance the adhesion property of electrode materials layer to the current collector. Such enhancement is expected that the coated Al current collector can hold thicker electrode materials layer than the pristine one and keep stronger interconnection for the cathodic materials layer with the electronic conductor. We believe that the enhanced interfacial property is ascribed to the roughen surface which could enforce the bonding force between the electrode materials layer and the current collector. Obviously the superior adhesion property can further effectively protect the electrode from dusting during extensive cycling which is expected to enhance the batteries' performance.



Figure 5. Chronoamperometric curves of Al current collectors in LIB electrolyte at 4.2 V vs Li<sup>+</sup>/Li.

Except aluminium element, pristine commercial Al current collector normally consists of trace of other elements such as Cu, Fe, and Si, etc. Exposing to electrolyte, these elements connected with Al component will form corrosion cells even without applied potential which will lead to gradual etching of Al. Typically, a thin and dense oxide passive coating on Al surface can prevent these cells from exposing to electrolyte and thus protect the current collector from corrosion. [22] This oxide layer is very thin and will not bring impairment to the conductivity. Figure 5 show chronoamperometric curves of Al current collectors in LIB electrolyte at 4.2 V vs Li<sup>+</sup>/Li. It can be seen that the anodic current of pristine commercial Al is higher than that of COL-Al. That implies the presence of oxide layer on the current collector can effectively slow down the corrosion process. Actually, the corrosion of current collectors contributes adverse consequences to the performance of Li-ion batteries. Such behavior not only decreases the effective interfacial area of the current collector but increases the resistance of the interface between the cathode and electrode materials layer. Server corrosion is likely to lead to the cracking of electrodes and also the aluminum ions in the electrolyte which etched out from the corrosion process may impair the performances of both the cathode and anode[16]. Therefore the protective layer on the current collector is expected to enhance its anticorrosion property and help maintain energy output as well as cyclability in a long-term operating.

We performed EIS measurements to examine effect of the presence of the thin COL on the interface between the current collector and active materials layer on the bulky properties of the cathode such as resistance, capacitance and reaction kinetics of the cathode in a LIB. As shown Figure 6, both electrodes show normal EIS characteristics for a cathode in a LIB with a depressed semicycle corresponding to charge-transfer resistance at the high-middle frequency region, and a sloping line related to mass-transport resistance at the low frequency region [25-27]. The lower charge-transfer resistance of cathode with COL-Al current collector is quite evident which may be ascribed to that the introduction of COL can reinforce the interaction between the active materials and the current collector, which facilitates the charge and lithium ion delivery in the electrochemical reaction. Although the oxide layer has a semiconductor property, as shown in the EIS profiles, the conductivity

of the electrode was not changed with its presence. Also, no additional arc loop was observed at high frequency region in the EIS of COL-Al electrode which shows the COL on the Al foil surface did not change the bulky properties of the Al current collector. Therefore we can conclude that the conductivity and capacitance of the electrode did not undergo clear impairment after the oxidation treatment process.



**Figure 6.** Electrochemical impedance spectra of electrodes obtained at open circuit potential (a. electrode with COL-Al current collector; b. electrode with commercial Al current collector).



**Figure 7.** Galvanostatic charge-discharge curves, discharge rate capabilities and coulombic efficiency profiles of LIBs with COL-Al (a, b) and pristine commercial Al current (c, d) collectors.

To verify the effect of superior anti-corrosion and adhesive properties of the current collector on lithium ion battery performance and durability, we carried out charge/discharge measurements at various rates ranging from 0.1 C to 5 C for the LIBs with different current collectors which are shown in Figure 7. The  $\Delta E$  values evaluated from the difference between charge and discharge potential plateaus were 0.061 V and 0.150 V at a rate of 0.1C for the LIBs using COL-Al and pristine Al current collector, respectively. The corresponding values increased with the increased rates and were respectively boosted up to 0.648 V and 1.069 V at the rate of 5 C. It is quite clear that the electrochemical polarization of the electrode was decreased with the introduction of COL at all the rates. The reduced electrochemical polarization can be ascribed the enhanced adhesion and anticorrosion ability of the current collector surface [28-29]. These results are well consistent with the above EIS measurements. We can also observe that the specific capacities of the COL-Al LIB increased at all the rates. As can be also seen from Figure 7a and 7c, the specific capacities of the LIB with COL-Al are 4 ~ 8 mAh g<sup>-1</sup> higher than that with pristine commercial Al current collector for all the rates. That further confirms the low electrochemical polarization of the electrodes which benefits the fast electrode kinetics and thus deliveries higher energy for the LIBs.

The merits of higher anti-corrosion ability and better interfacial contact property of the COL-Al were highlighted by long-term cycle life testing. We carried out life tests for the LIB full cells using pristine commercial and OCL-Al foils as current collectors respectively. The design capacity of the batteries is 25 Ah which is same as our commercial products volume, for we expected to examine the possibility for realizing the achievement in real commercial applications. The test results are shown in Figure 8. The discharge capacity of the cell with pristine commercial Al foil as cathode current collector experienced fading at No. 250 cycle which decreased to 85% of the initial capacity after 2000 cycles. It illustrates that the texturing surface of the Al current influences the capacity retention of the cells mainly on an extended time scale, because we can observed that the cell with bared Al current collector as well did not have a significant capacity fading when the cycle number was less than 250 [30].



**Figure 8.** Cycle life test of LiFePO<sub>4</sub>/graphite full cells (LIB using COL-Al current collector (a) and pristine commercial Al current collector (b)).



Figure 9. SEM images of pristine commercial Al current collector before (a) and after (c) long-term cycle life test in full cells, and COL-Al current collector before (b) and after (d) cycle life test.

While the battery with COL-Al foil current collector for cathode showed much better cyclability, with no significant discharge capacity drop was observed until 825 cycles. Even after 2000 charge-discharge cycles, the capacity still presented with 95% retention of the initial capacity. We performed morphology investigation for the surface of the current collector after cycle tests. The electrode materials layer was carefully removed from the current collector. The SEM investigation was performed and the images of the surfaces of the current collector are shown in Figure 9 (c) and Figure 9 (d), respectively. We observed that some big corrosion pins together with many small ones on the surface of the pristine commercial Al current collector after 2000 cycles. In comparison, no corrosion pin was observed on the surface of COL-Al current collector while a few areas became smooth with bumps disappeared after the cycle life test. The results showed that the presence of COL cannot completely prevent the Al current from corrosion as other coatings [28, 31]. However, due to the protection by the COL, the corrosion inhibitor than other coatings in lithium ion batteries scale. Especially the oxide layer can effectively protect the Al current collector from corrosion in the extensive cycling. This result is in good consistence with the electrochemical measurement above.

# 4. CONCLUSIONS

In summary, a thin COL consisted of manganese and aluminium oxides was synthesized on an Al current collector for Li-ion battery cathode to reinforce its adhesive and anti-corrosion properties by

a simple oxidation process. The smooth surface of Al foil became rough with bump pattern structure after the modification. The electrochemical characterizations showed the presence of COL on the Al current collector can greatly enhance its anti-corrosion ability. The peel tests showed the COL-Al has a better adhesion property than the pristine commercial Al current collector. Electrochemical characterizations showed that the COL effectively low the electrode polarization and enforce the LIB's performance. The Al current collectors were also fabricated into a 25 Ah Li-ion battery which is identical to our commercial cell as a cathode current collector to evaluate the cycle performance. Extensive cycle life tests showed the Li-ion battery still had a discharge capacity of 95% retention of the initial capacity after 2000 cycles, in comparison with that with pristine commercial Al current collector for high cycle life Li-ion batteries.

Moreover, such a preparation approach for Al current collector modified with the COL is simple, cost-effective and can be utilized for large-scale lithium ion batteries devices.

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